

# Improving Long-term Capacity Retention of NMC811 via Lithium Aluminate Coatings Using Mixed-metal Alkoxides

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# Contents

S1. Materials .....	3
S2. Experimental Procedures.....	3
S2.1. Synthetic Procedures.....	3
S2.2. Materials Characterization.....	5
S3. Characterization of the single-source precursors .....	9
S3.1. Solution NMR spectra .....	9
S3.2. Solid-state NMR spectra.....	18
S4. Precursor thermolysis .....	20
S4.1. Thermogravimetric analysis .....	20
S4.2. Powder X-ray diffraction .....	22
S4.2.2. Rietveld refinements results for the thermolysis products of 1-3.....	22
S4.3. Solid-state NMR.....	24
S5. Characterization of the coated NMC811 materials .....	24
S5.1. Electron microscopy.....	24
S5.1.1. Scanning electron microscopy.....	24
S5.1.2. Energy-dispersive X-ray spectroscopy .....	31
S5.2. Powder X-ray diffraction .....	35
S5.2.1. Laboratory Powder X-ray Diffraction Patterns .....	35
S5.2.2. Refinement Results for the Laboratory PXRD data .....	38
S5.2.3. Synchrotron Powder X-ray Diffraction Patterns .....	41
S5.2.4. Refinement Results for the Synchrotron PXRD data .....	45
S5.3. X-ray Photoelectron Spectroscopy.....	46
S5.3.1. O 1s and C 1s spectra .....	46
S5.3.2. Peak parameters in the fittings of the C 1s, O 1s and Al 2p spectra .....	47
S5.3.3. Method for fitting the O 1s spectra.....	48
S5.3.4. Peak constraints used to fit of the O 1s, C 1s and Al 2p spectra.....	48
S5.4. Solid-state NMR spectra.....	50
S5.4.1. $^1\text{H}$ and $^7\text{Li}$ SSNMR study of the effect PC-NMC811 water treatment .....	50
S5.4.2. $^1\text{H}$ SSNMR Measurements .....	50
S5.4.3. $^7\text{Li}$ SSNMR Measurements.....	53
S5.4.4. $^{27}\text{Al}$ SSNMR Measurements .....	56
S5.5. Electrochemical cycling .....	59
S5.5.1. Specific Capacity vs. Cycle Number Data .....	59
S5.5.2. Voltage Profiles.....	61
S5.5.3. dQ/dV Profiles for Single-crystal NMC811 Electrochemical Cycling data.....	64
S5.5.4. Comparison with Literature .....	65
References .....	67

## S1. Materials

Polycrystalline NMC811 (Targray), Single-crystal NMC811 (Li-FUN), 1.0 M LiAlH<sub>4</sub> solution in THF (Sigma-Aldrich). Benzyl alcohol (Sigma-Aldrich ≥ 99 %), isopropanol (Sigma-Aldrich ≥ 99 %), tert-butanol (Alfa Aesar ≥ 99 %).

## S2. Experimental Procedures

### S2.1. Synthetic Procedures

*General synthetic considerations:* All the solvents were dried over activated molecular sieves for 48 h before use and all the operations were carried out under strict inert atmosphere.

*Synthesis of Li[Al(OCH<sub>2</sub>Ph)<sub>4</sub>] (1):* Addition of 4 mL of a 1M LiAlH<sub>4</sub> solution in THF over 1.66 mL of a benzyl alcohol solution in 10 mL of THF (4 equivalents) led to evolution of H<sub>2</sub>(g) and the formation of a white suspension. After 1h, a clear solution is obtained by removing the solids by filtration and the product (white powder) is obtained by removing the solvent under vacuum. The product is then dried under vacuum at 40 °C for 1h. Yield: 500 mg, 1.08 mmol, 27 %.

**<sup>1</sup>H NMR** (400 MHz, d<sub>8</sub>-THF, 295K) δ/ppm: 7.34 (s, 1H, Ph-H *ortho*), 7.19 (s, 1H, Ph-H *meta*), 7.09 (s, 1H, Ph-H, *para*), 4.81 (s, 2H, -CH<sub>2</sub>Ph). **<sup>13</sup>C{<sup>1</sup>H} NMR** (400 MHz, d<sup>8</sup>-THF, 295K) δ/ppm: 146.6 (1C, Ph(C)-C), 128.6 (2C, Ph(C) *meta*), 128.2 (2C, Ph(C) *para*), 127.1 (2C, Ph(C) *ortho*), 65.6 (1C, -CH<sub>2</sub>Ph). **<sup>7</sup>Li NMR** (400 MHz, d<sub>8</sub>-THF, 295K) δ/ppm: 0.35 (s, b), -0.23 (s). **<sup>27</sup>Al NMR** (400 MHz, d<sub>8</sub>-THF, 295K) δ/ppm: 69.76 (s). **CHN analysis** LiAlC<sub>28</sub>H<sub>28</sub>O<sub>4</sub>: 72.73; H, 6.06; N, 0.0. Found: C, 71.93; H, 6.08; N, 0.0.

*Synthesis of Li[Al(O<sup>i</sup>Pr)<sub>4</sub>] (2):* Addition of 4 mL of a 1M LiAlH<sub>4</sub> solution in THF over 1.22 mL of isopropanol solution in 10 mL of THF (4 equivalents) leads to H<sub>2</sub>(g) formation and the precipitation of a white solid. After 1h, the solid is filtered and the solvent is removed from the resulting white suspension under vacuum. A white powder is obtained and dried under vacuum at 40 °C for 1h. Yield: 300 mg, 1.11 mmol, 12.3 %.

**CHN analysis** LiAlC<sub>12</sub>H<sub>28</sub>O<sub>4</sub>: C, 50.06; H, 10.01; N, 0.0. Found: C, 50.32; H, 9.91; N, 0.0.

*Synthesis of Li[Al(O<sup>t</sup>Bu)<sub>4</sub>] (3):* Addition of 4 mL of a 1M LiAlH<sub>4</sub> solution in THF over 1.52 mL of tert-butanol solution in 10 mL of THF (4 equivalents) leads to H<sub>2</sub>(g) formation and a clear solution (no precipitate is observed). After 1h, the solvent is removed under vacuum and the resulting white powder is dried under vacuum at 40 °C for 1h. Yield: 1.08 g, 3.31 mmol, 83 %.

**<sup>1</sup>H NMR** (400 MHz, d<sub>8</sub>-THF, 295K) δ/ppm: 1.23 (s, -C(CH<sub>3</sub>)<sub>3</sub>, 9H), 1.21 (s, -C(CH<sub>3</sub>)<sub>3</sub>, 9H). **<sup>13</sup>C{<sup>1</sup>H} NMR** (400 MHz, d<sub>8</sub>-THF, 295K) δ/ppm: 68.32 (s, -C(CH<sub>3</sub>)<sub>3</sub>, 1C), 34.16 (s, -C(CH<sub>3</sub>)<sub>3</sub>, 3C), 34.05 (s, -C(CH<sub>3</sub>)<sub>3</sub>, 3C). **<sup>7</sup>Li NMR** (400 MHz, d<sub>8</sub>-THF, 295K) δ/ppm: -0.26 (s). **<sup>27</sup>Al NMR** (400 MHz, d<sub>8</sub>-THF, 295K) δ/ppm: 77.86 (s), 60.26 (s), 53.1 (s). **CHN analysis** LiAlC<sub>16</sub>H<sub>36</sub>O<sub>4</sub>: C, 58.91; H, 11.04; N, 0.0. Found: C, 58.1; H, 11.74; N, 0.0. **HRMS** (ASAP<sup>-</sup>) m/z calculated for [M]<sup>-</sup> [C<sub>16</sub>H<sub>36</sub>Al<sub>1</sub>O<sub>4</sub>]<sup>-</sup>: 319.240 found: 319.2422. [C<sub>12</sub>H<sub>28</sub>Al<sub>1</sub>O<sub>3</sub>]<sup>-</sup>: 247.190, found: 247.1855.

*Solution NMR measurements of compounds 1-3.* <sup>27</sup>Al NMR spectra were background-subtracted to eliminate the contribution from the probe to the observed spectra (Figure 4.1). The measurements were performed in d<sub>8</sub>-THF at room temperature using a 9.4 T magnet (400 MHz <sup>1</sup>H Larmor frequency) and a solution of Al(NO<sub>3</sub>)<sub>3</sub> in D<sub>2</sub>O at 0 ppm as external reference.

*Thermolysis of compounds 1-3.* The precursors were loaded in alumina crucibles and heated under air with a heating rate of 10 °C / min to 300, 400, 500 or 800 °C (**1**) or 400, 500 or 800 °C (**2** and **3**) for 4 h and then allowed to cool down to room temperature. The measurements, shown in Figure S4.1.1., were performed under synthetic air flow as the annealing of the coated NMC811 samples after deposition is performed under air. To minimize precursor hydrolysis, a small amount of sample was quickly transferred from the glovebox to the TGA instrument where it was kept under synthetic (dry) air. The precursors were heated from 25 to 800 °C at a rate of 10 °C / min and the changes in sample masses were recorded.

*Water soaking step.* NMC811 is loaded in a Schlenk flask inside the glovebox and transferred to a Schlenk line and put under nitrogen. Deionized water is then added (10 mL water / g of NMC) and stirred for 2h at room temperature. The water is then removed by syringe and the sample is dried under vacuum at 50 °C for 2h. The resulting product is kept under inert atmosphere for the coating deposition step. Part of this sample is annealed under air at 400 °C to study the effect of annealing on a water soaked NMC without a coating (control experiment).



*Coating deposition.* The coating was deposited onto NMC811 using this same procedure regardless of the precursor used (compounds **1** – **3**) or the NMC sample ( $\text{Al}_2\text{O}_3$  coated single-crystal NMC811 or uncoated polycrystalline NMC811). The precursor and NMC811 are loaded in a Schlenk flask inside the glovebox (precursor wt. is 1 % of the total solid wt. %). The reaction vessel is connected to a Schlenk line and put under nitrogen. Dry THF (10 mL / g of NMC) is added, and the reaction proceeds at 60 °C for 48 h stirring. After that, the solvent is removed by syringe and washed three times with 10 mL of THF each. Finally, the product is dried under vacuum at 100 °C for 2 h. The same procedure was employed for the NMC811 that was pre-soaked in water (water soaking step) and for the control samples but without the addition of precursor for the control experiments.

## S2.2. Materials Characterization

*X-ray photoelectron spectroscopy (XPS).* The XPS spectra were collected using a Phi Versaprobe instrument with Al K $\alpha$  radiation using a facility from the Henry Royce institute. A spot size of 100  $\mu\text{m}$ , a step size of 0.1 eV and a pass energy of 55 eV were used. Due to the relatively short amount of time for the measurements available in the previous facility, the XPS data of the pre-soaked PC-NMC811 coated with **3** was measured in a HarwellXPS facility using a Kratos SUPRA instrument with a pass energy of 40 eV, a spot size of 700 x 300  $\mu\text{m}$  and a step size of 0.1 eV. The XPS spectra were calibrated from the C 1s aliphatic peak which was set to 284.8 eV.

*Fitting of the C 1s XPS spectra.* The C 1s spectra were fitted to five components corresponding to aliphatic (C-C, C-H), ethers and alcohols (C-OH, C-O-C), ketones (C=O), esters (O-C=O) and carbonates ( $\text{CO}_3^{2-}$ ) and the aliphatic peak was set to 284.8 eV.

*Fitting of the Al 2p XPS spectra.* The Al 2p region was fitted to 8 components. Al 2p, Mn 3p, Li 1s regions were fitted to a single peak each. The Co 3p was fitted to two peaks corresponding to the main peak and the satellite. The area of the Co 3p satellite was constrained to be 10% of that of the Co 3p main peak. The Ni 3p region was fitted to three peaks corresponding to Ni 3p $_{3/2}$ , Ni 3p $_{1/2}$  and Ni 3p satellite. The Ni 3p $_{1/2}$  was constrained to be half of the area of the Ni 3p $_{3/2}$ , same width and 2.2 eV above the position of the Ni 3p $_{3/2}$  peak. The Ni 3p satellite was constrained to be 6eV above the Ni 3p $_{3/2}$  peak and with a FWHM of 3 to 5 eV.<sup>1,2</sup>

*Scanning Electron Microscopy.* Images were taken with a TESCAN MIRA3 Field emission gun (FEG) – SEM in secondary electron (SE) mode with beam voltages ranging from 2 kV for the pristine NMC and 5 kV for the coated samples. The working distance was 6 mm for all images. All samples were coated with 10 nm of Cr before imaging using a Quorum Technologies Q150T ES Turbo-Pumped Sputter Coater/Carbon Coater.

*SSNMR Measurements of the Precursors.* Most of the spectra were recorded on a 11.7 T Bruker Avance III spectrometer (500 MHz  $^1\text{H}$  Larmor frequency), using a Bruker 2.5 mm magic-angle spinning (MAS) NMR probe and spinning at 30 kHz MAS frequency. Only the pristine compound **3** annealed at 400 and 500 °C samples were recorded using a 16.4 T Bruker Avance III spectrometer (700 MHz  $^1\text{H}$  Larmor frequency), with a Bruker 1.3 mm magic-angle spinning (MAS) probe and spinning at 50 kHz MAS frequency. The spin lattice relaxation time constant ( $T_1$ ) was determined for each sample using a saturation recovery experiment. The spectra shown in this work were acquired using a rotor-synchronized Hahn-echo pulse sequence. The recycle delay ( $d_1$ ) was determined directly from the  $T_1$  values ( $d_1 = T_1 * 5$ ). Sample quantities ranged between 1.15 and 14.15 mg for the 500 MHz measurements and between 1.84 and 8.53 mg for the 700 MHz measurements.  *$^{27}\text{Al}$ -NMR measurements.*  $\text{AlF}_3$  (-17.0 ppm) was used as an external reference to calibrate the  $^{27}\text{Al}$  chemical shift, between 256 and 376,320 scans were recorded.  *$^1\text{H}$ -NMR measurements.* Adamantane ( $\text{C}_{10}\text{H}_{16}$ , 1.9 ppm) was used as an external reference to calibrate the  $^1\text{H}$  chemical shift, between 16 and 256 scans were recorded.  *$^7\text{Li}$ -NMR measurements.*  $\text{Li}_2\text{CO}_3$  (0 ppm) was used as an external reference to calibrate the  $^7\text{Li}$  chemical shift and between 16 and 32 scans were recorded.

*SSNMR Measurements of the NMC811 materials.* The spectra were acquired using a 16.4 T Bruker Avance III spectrometer (700 MHz  $^1\text{H}$  Larmor frequency), with a Bruker 1.3 mm magic-angle spinning (MAS) probe and spinning at 50 kHz MAS frequency. Sample quantities ranged between 4 and 8.53 mg. The spin lattice relaxation constant ( $T_1$ ) for the  $^1\text{H}$  measurements was determined for each sample using a saturation recovery experiment except for the  $^7\text{Li}$  experiments for which a recycle delay of 100 ms was used in all the measurements. The spectra were then acquired using a rotor-synchronized Hahn-echo pulse sequence. The recycle delay ( $d_1$ ) was determined directly from the  $T_1$  values ( $d_1 = T_1 * 5$ ) in the case of the  $^1\text{H}$  measurements. For the  $^{27}\text{Al}$  NMR measurements, the  $d_1$  was set to 25 ms and for the  $^7\text{Li}$  NMR to 100 ms. Same references were used for external calibration as in the previous section. To prepare the SSNMR samples of the cycled electrodes, the cycled cells were transferred to an argon-filled glovebox, opened and the electrode was washed one time with dimethyl carbonate (DMC) and dried under vacuum for 15 minutes. The material was then scrapped from

the aluminum current collector and packed in 1.3 mm SSNMR rotors. The spectra were normalized to the sample mass and the number of scans

*Powder X-ray Diffraction.* The laboratory diffraction measurements were performed using a diffractometer equipped with a Cu K $\alpha$  radiation source ( $\lambda = 1.541 \text{ \AA}$ ) and over the 10–100° 2 $\theta$  range of for 1 h. The synchrotron PXRD patterns were measured at I11 (Diamond Light Source, UK). For the synchrotron measurements, samples were packed in a quartz capillary and sealed with epoxy resin before shipping and the peak-shapes were obtained directly from a silicon refinement. All the refinements were performed using the TOPAS program (V6).

*Refinement of thermolysis products.* Li and Al atoms occupy the same site (4a) while the oxygen atoms are in a separate site (8b) in the  $\gamma$ -LiAlO<sub>2</sub> phase. Therefore, x and y coordinates were constrained to the same value and the z coordinate fixed to 0 for Li and Al, as demanded by the Wyckoff positions of the P4<sub>1</sub>2<sub>1</sub>2 space group. The coordinates of the oxygen atom (8b site) were allowed to vary freely. The  $\alpha$ -LiAlO<sub>2</sub> phase is described by a R<sup>3</sup>m space group and presents three different atomic positions 3a (Li), 3b (Al) and 6c (O). Only the z coordinate of the oxygen atom is refined, while the other atomic coordinates and site occupancies are fixed. All the occupancies were fixed to 1 in these refinements.

*Electrode fabrication.* Electrodes were prepared by mixing 90 wt % NMC811 powder with 5 wt% carbon black (Super-P Carbon, Timcal) and 5 wt % PVDF binder (Kynar, HSV-900). The total amounts of the coated and uncoated solids ranged between 250 and 400 mg. The solid mixture was ground for 5 minutes under air using a pestle and mortar. Then 0.5-0.8 mL of N-methyl-2-pyrrolidone (NMP) was added, the slurry was mixed in a planetary centrifugal mixer (Thinky, ARM/310 CE) at 2000 rpm for at least 10 minutes and transferred to an argon-filled glovebox. Here, the slurry was cast on aluminium foil with a doctor-blade to 150  $\mu\text{m}$  thickness. The cast electrode was then dried under atmospheric pressure overnight at room temperature and then under vacuum for 4-5h at room temperature. The electrodes were then cut into disks of 12.7 mm diameter. These were dried under vacuum at 120 °C for 24 h before being transferred to an argon-filled glovebox with oxygen and water levels below 10 ppm. The resulting electrodes had active mass loadings of  $6.7 \pm 1.7 \text{ mg / cm}^2$ .

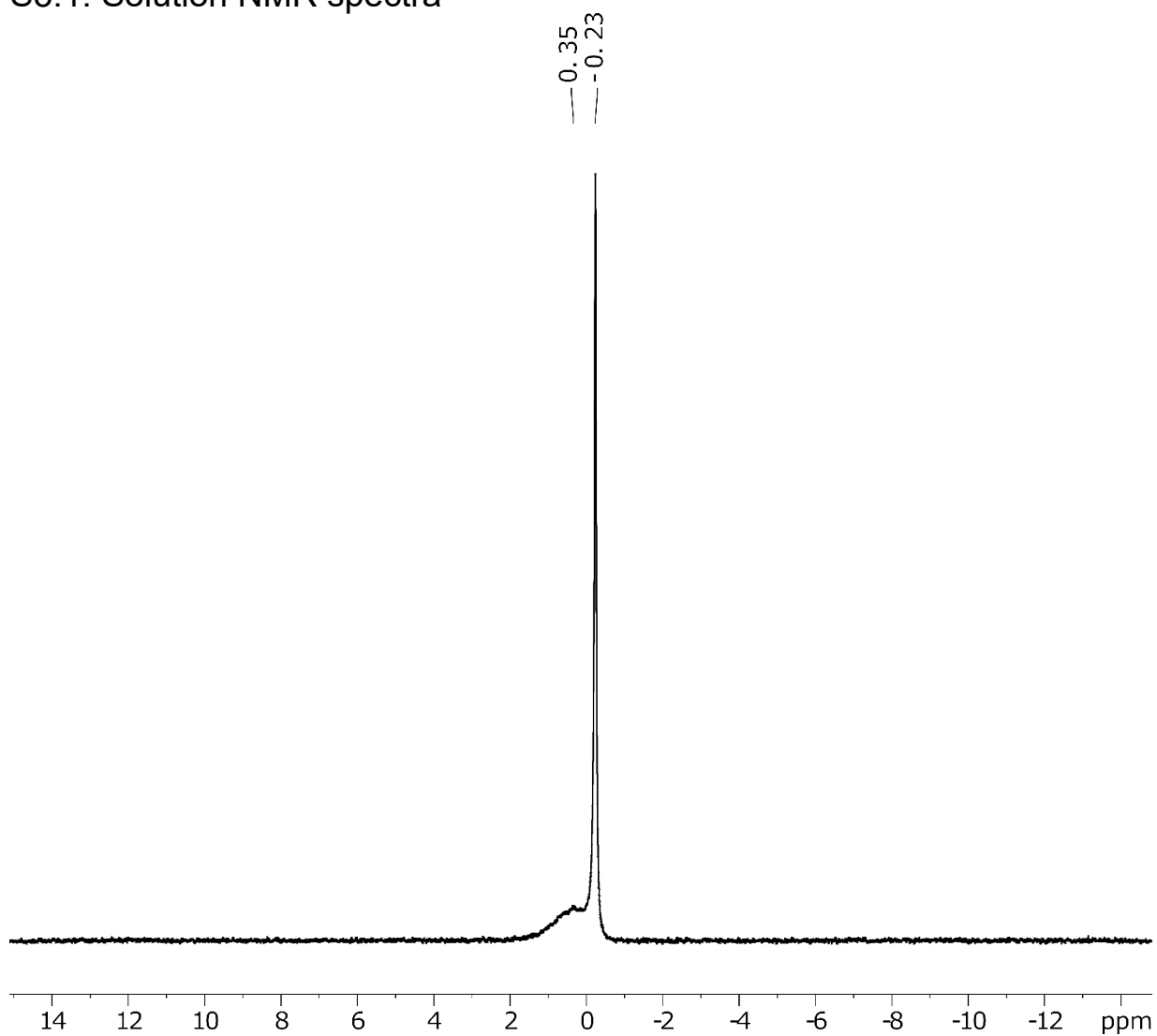
*Coin cell assembly.* Coin cells were assembled inside an argon filled glovebox with water and oxygen levels below 10 ppm. The cell components were a lithium disk of 16 mm

diameter, purchased from LTS Research Laboratories, Inc. as the counter electrode, one 0.5 mm thick spacer, a steel spring, a polypropylene separator (Celgard 3031) soaked with 60  $\mu\text{L}$  of electrolyte and the cathode disk. The Celgard separators had been previously cut into 16 mm disks, washed with ethanol, and dried under vacuum at 50  $^{\circ}\text{C}$  overnight. The electrolyte used was 1.0 M  $\text{LiPF}_6$  in ethylene carbonate (EC) and ethyl methyl carbonate (EMC) mixed in a 3:7 w/w ratio (SoulBrain, USA). The electrochemical cycling was performed using an Arbin LTB or a LAND CT2001A cycler.

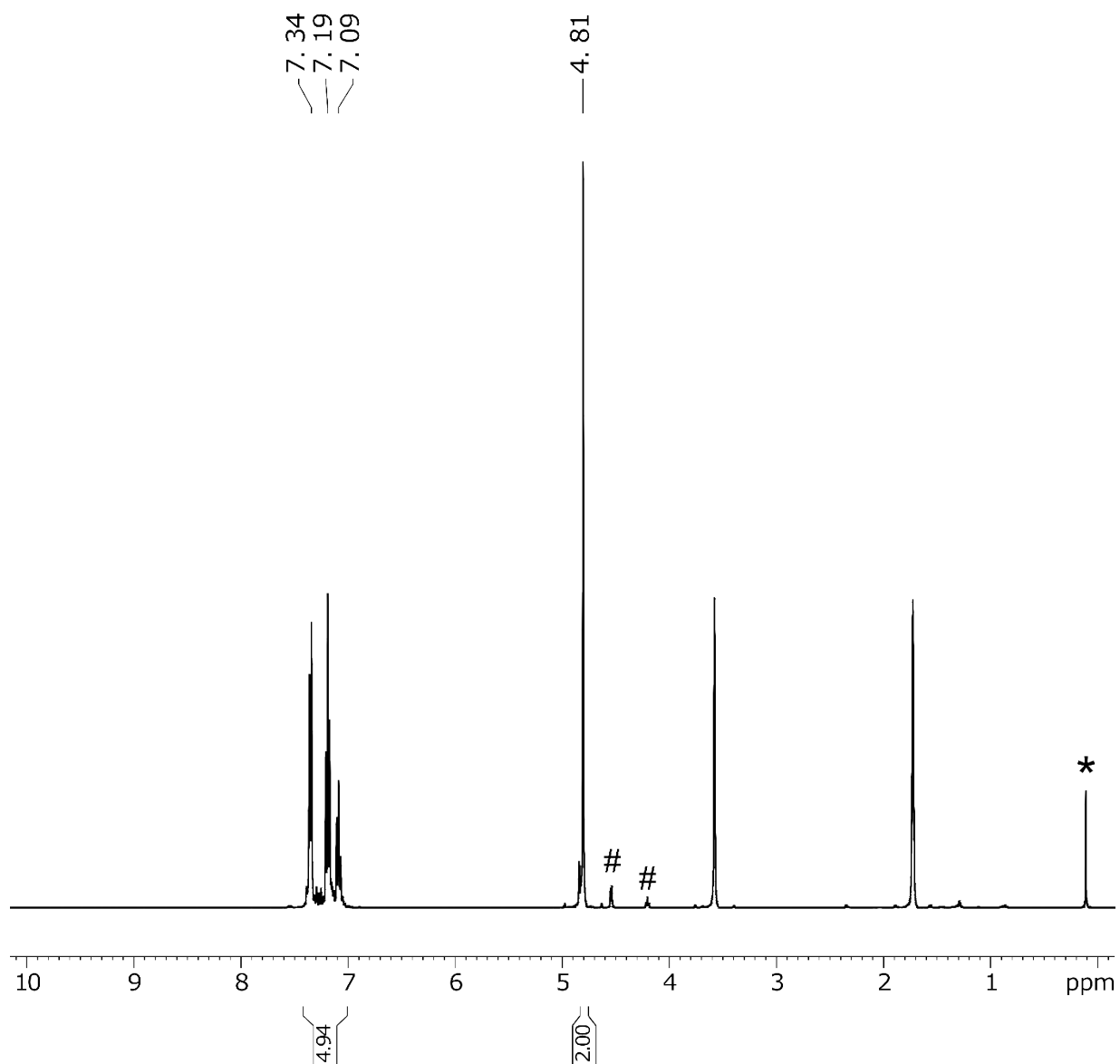
*Battery testing.* The voltage window was 3 to 4.3 V vs. Li with a constant current, constant voltage (CCCV) step at the end of charge. The testing was done at C/2 rate with two C/20 formation cycles and then two C/20 cycles every 50 cycles. In this work, 1C refers to the current needed to charge or discharge the cell in 1 h with an upper cut-off voltage of 4.3 V (corresponds to a cathode capacity of  $\sim 200$  mA h/g). The purpose of the slow cycles was to test if the observed decrease in specific capacity was caused by kinetic limitations which could be eliminated at slow rates.

### S3. Characterization of the single-source precursors

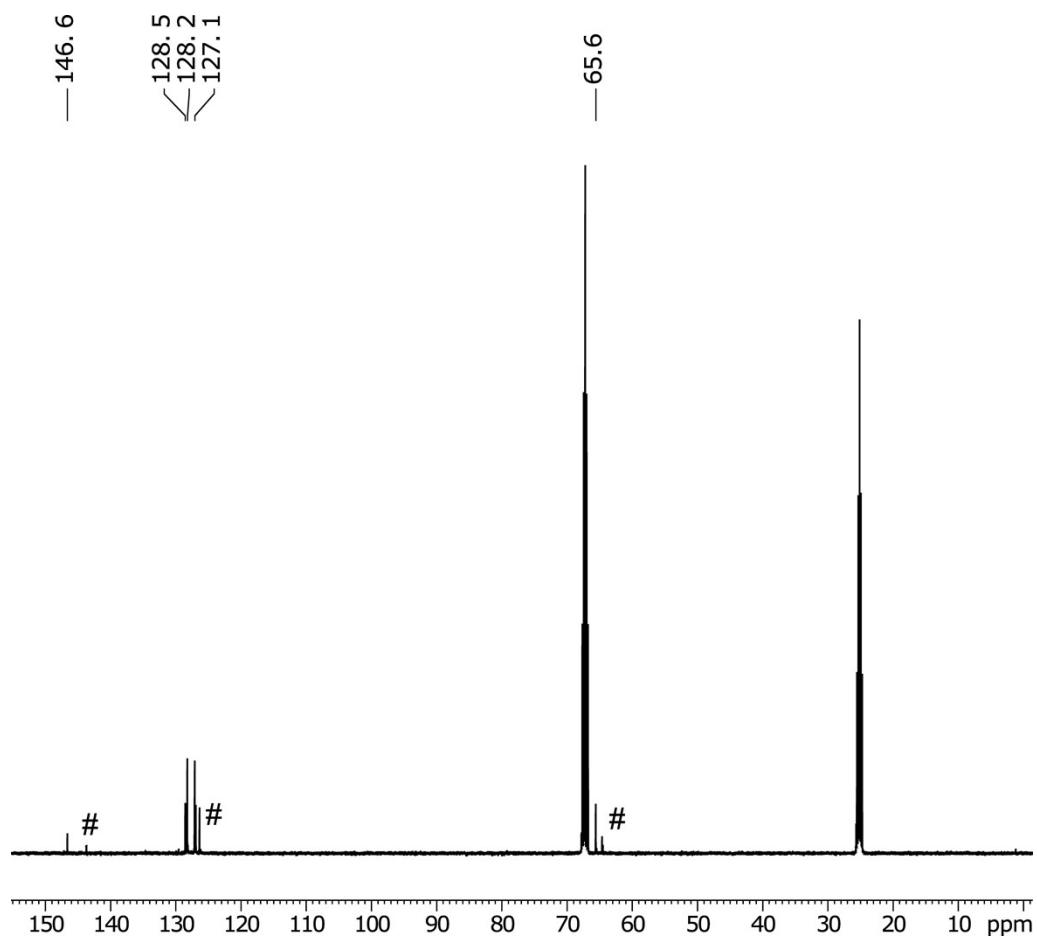
#### S3.1. Solution NMR spectra



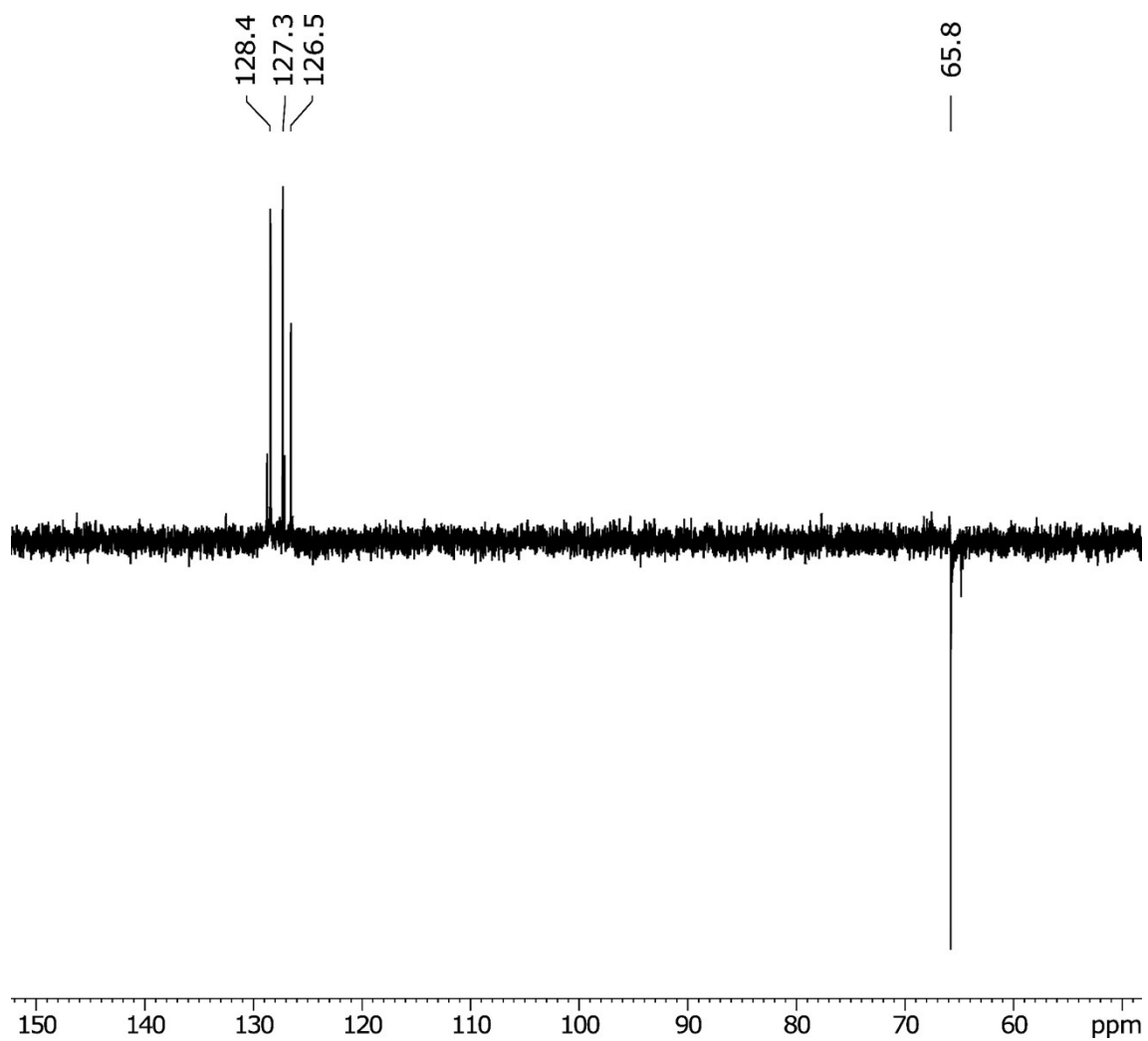
**Figure S3.1.1.**  ${}^7\text{Li}$  NMR (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[\text{OCH}_2\text{Ph}]_4$  (**1**).



**Figure S3.1.2.**  $^1\text{H}$  NMR (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[\text{OCH}_2\text{Ph}]_4$  (**1**). The peak due to silicone grease is marked with (\*). Peaks marked with an # are assigned to unreacted alcohol.

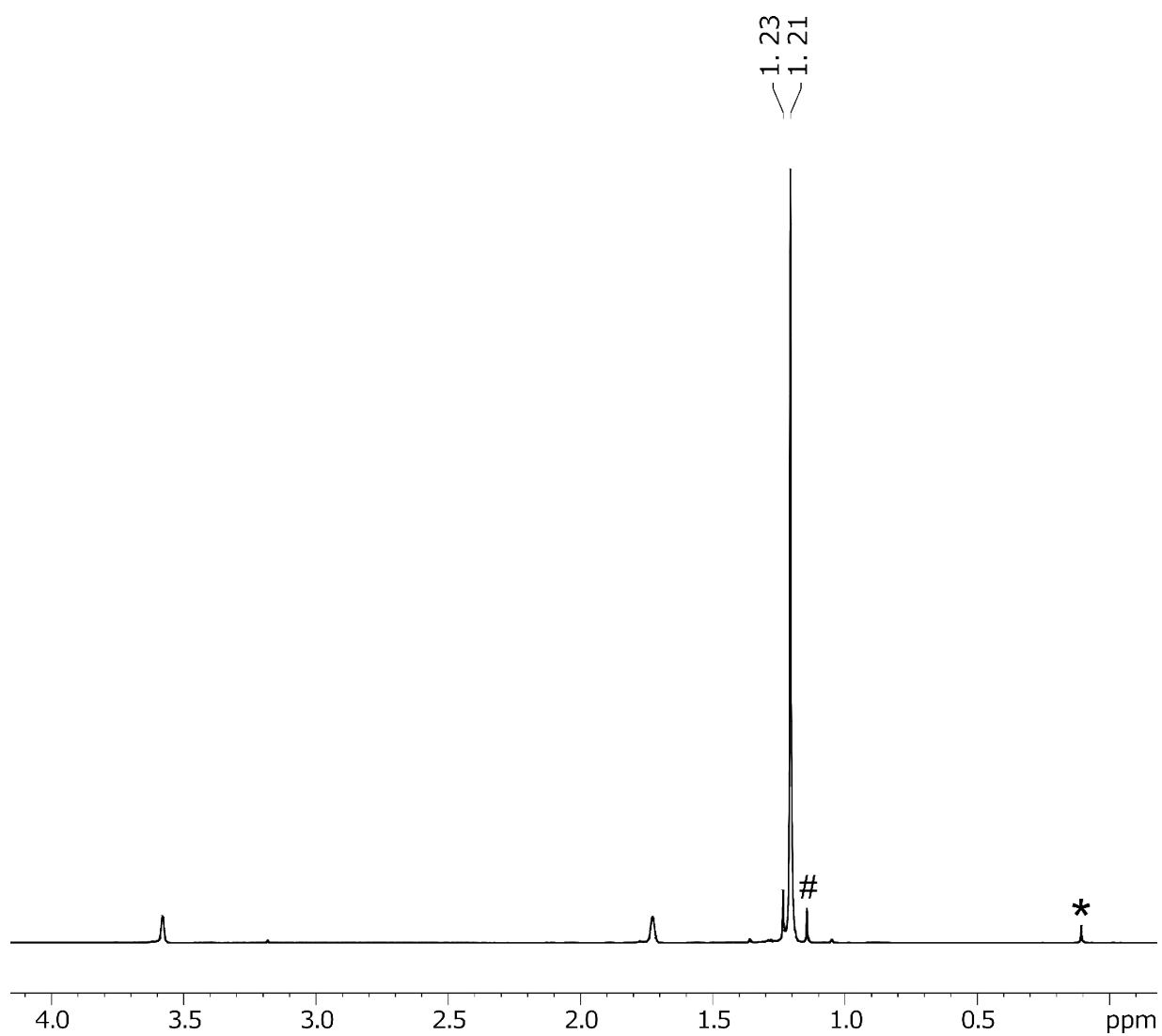


**Figure S3.1.3.**  $^{13}\text{C}$  NMR (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[\text{OCH}_2\text{Ph}]_4$  (**1**). Peaks marked with an # are assigned to unreacted alcohol.

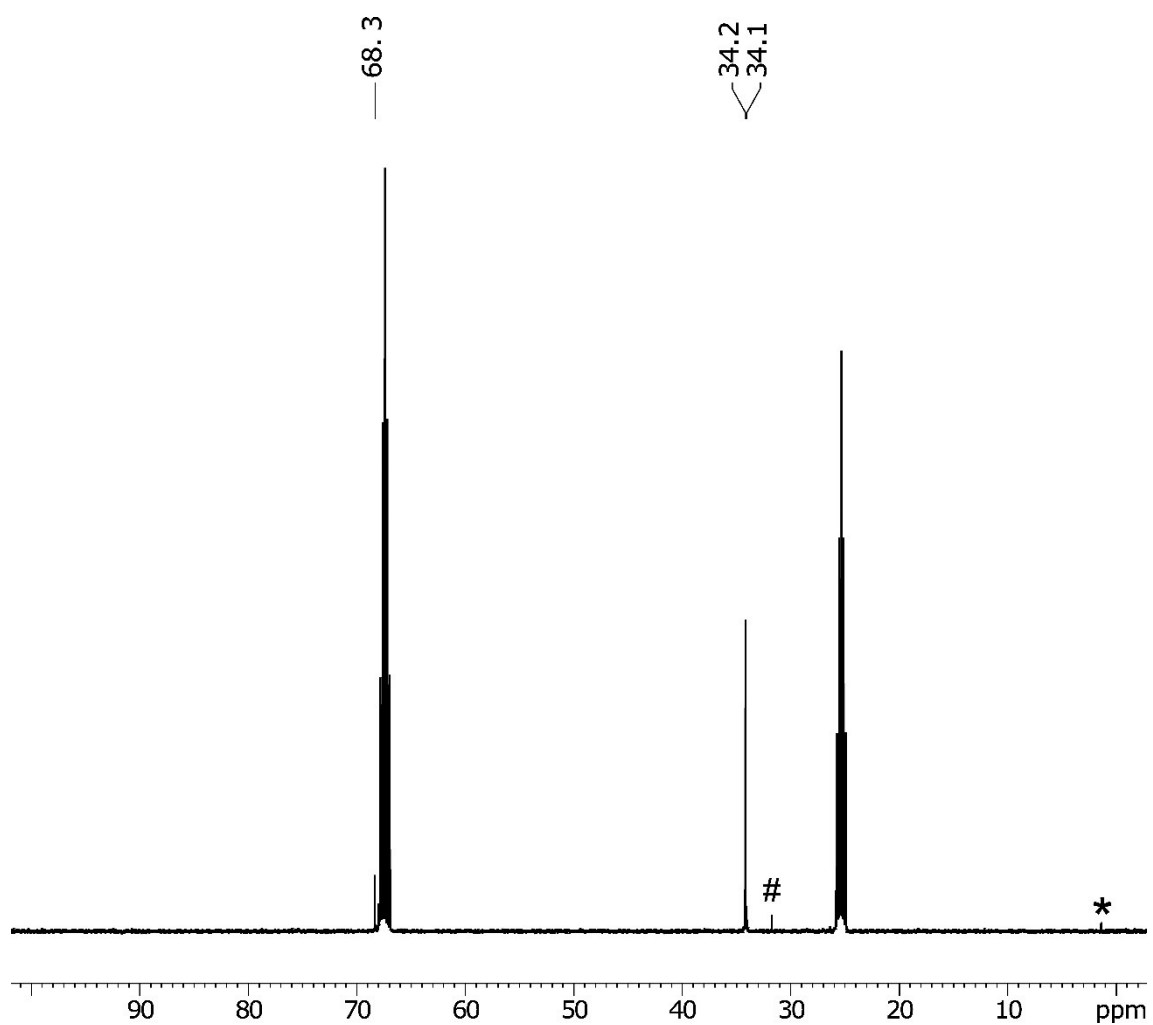


**Figure S3.1.4.**  $^{13}\text{C}$  NMR DEPT 135 (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[\text{OCH}_2\text{Ph}]_4$  (1).

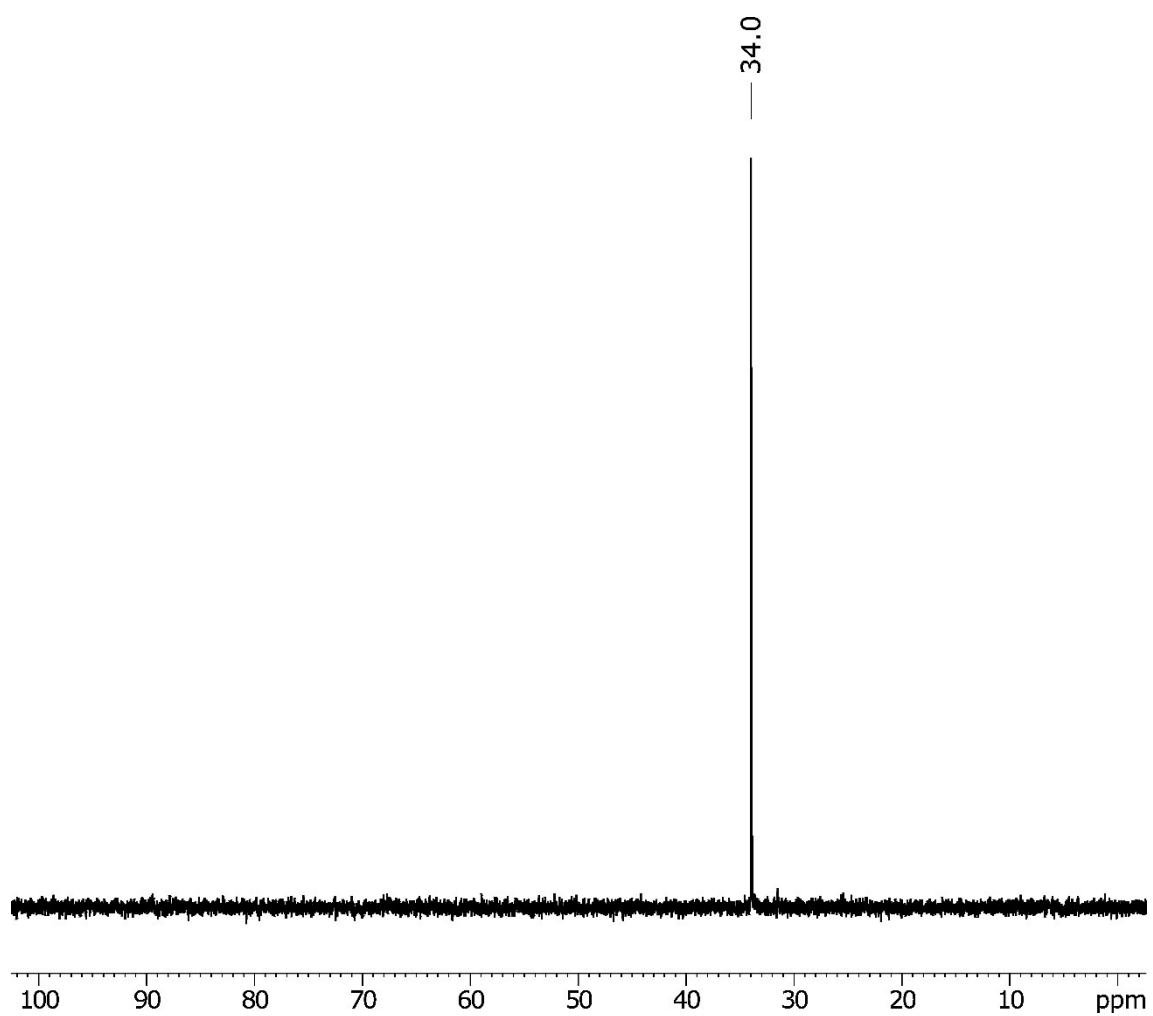




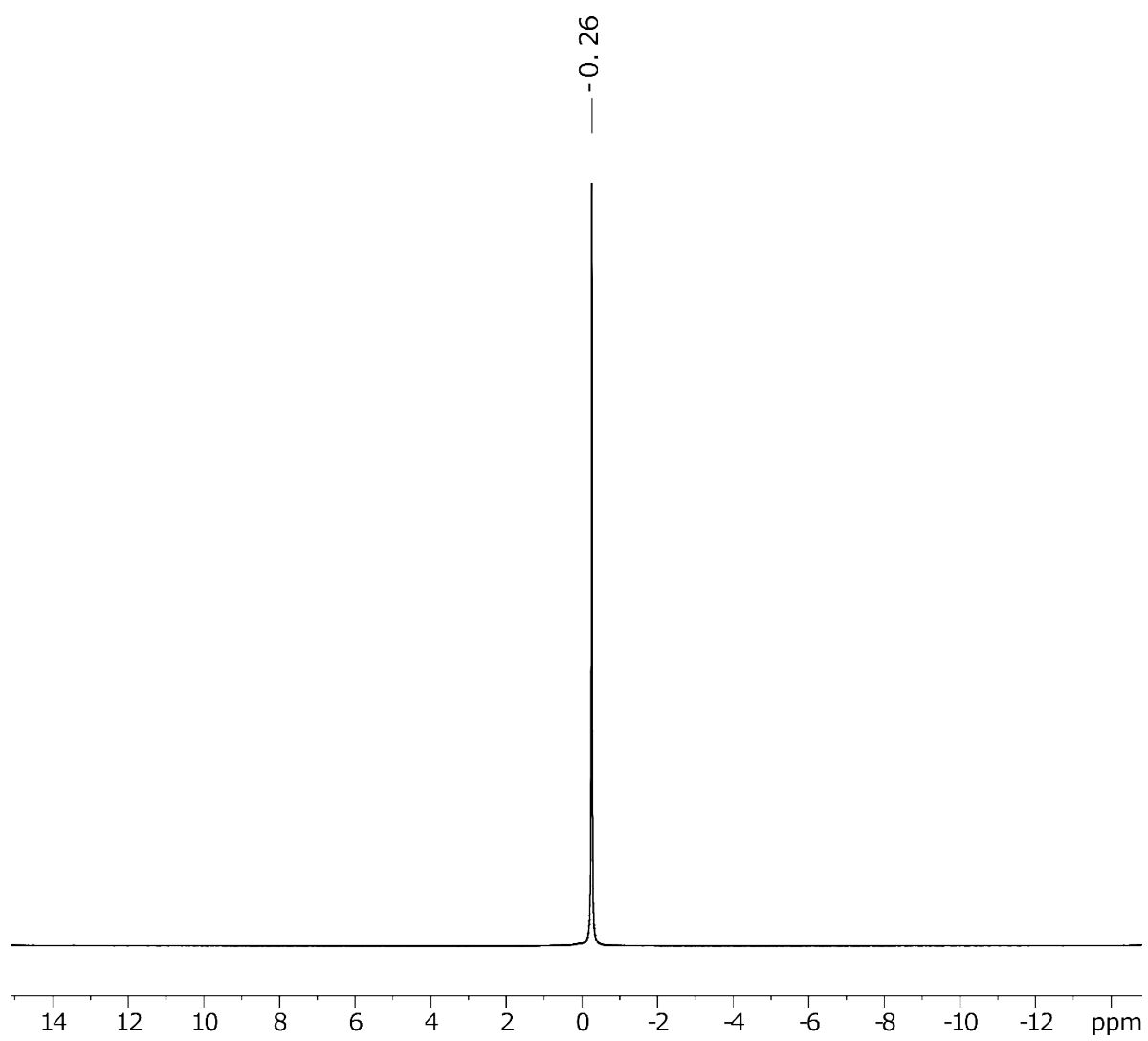
**Figure S3.1.5.**  $^1\text{H}$  NMR (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[(\text{O}^t\text{Bu})_4]$  (**3**). In addition of the THF solvent peaks, one signal arises due to the presence of silicone grease (\*) and one due to unreacted tert-butanol (#).



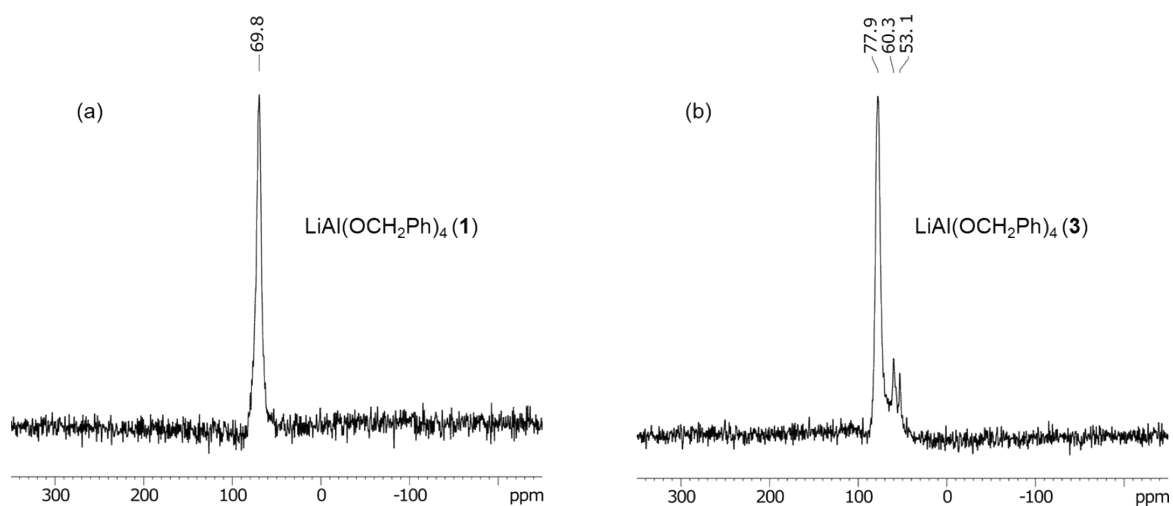
**Figure S3.1.6.**  $^{13}\text{C}$  NMR (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[\text{O}^t\text{Bu}]_4$  (**3**). Apart from the THF solvent peaks found at 67.21 and 25.12 ppm, one signals arise due to the presence of silicone grease (\*) and two due to unreacted tert-butanol (#).



**Figure S3.1.7.**  $^{13}\text{C}$  NMR DEPT 135 (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[\text{O}^t\text{Bu}]_4$  (**3**).

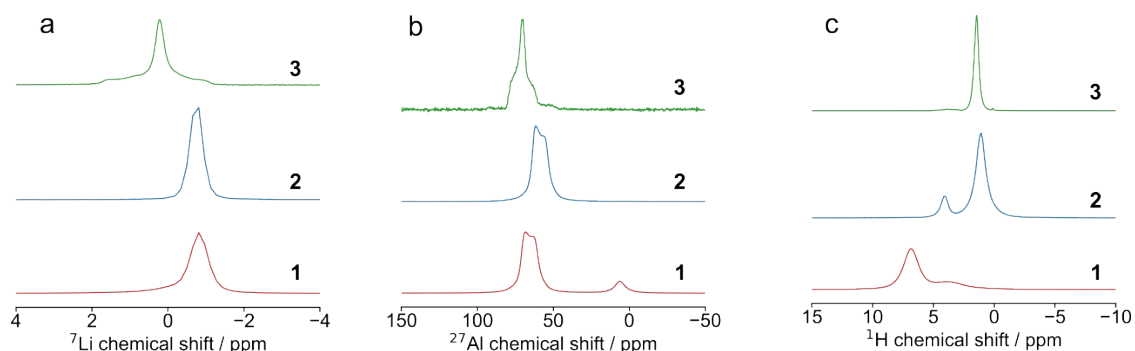


**Figure S3.1.8.**  ${}^7\text{Li}$  NMR (400 MHz, THF, 295 K) spectrum of  $\text{LiAl}[\text{O}^i\text{Bu}]_4$  (**3**).

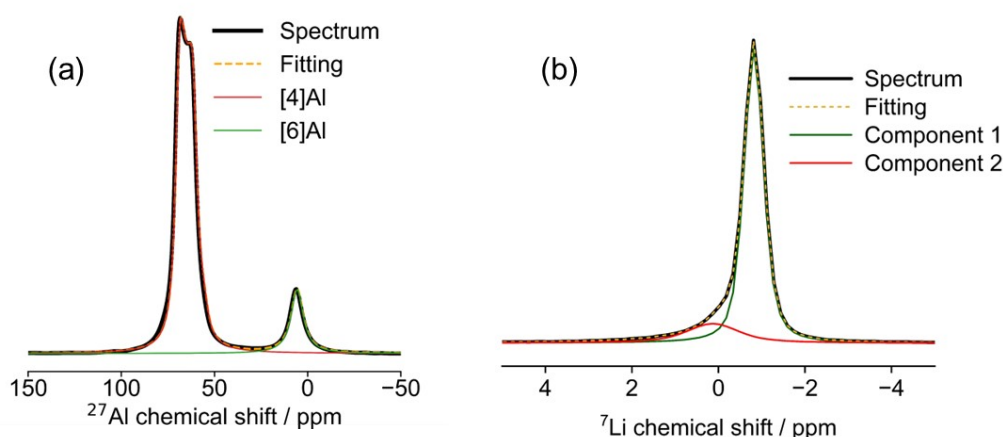


**Figure S3.1.9** .  $^{27}\text{Al}$  solution NMR of **1** and **3**. The compounds were dissolved in  $d_8$ -THF, and the NMR measurements were performed at room temperature using a 9.4 T magnet (400 MHz  $^1\text{H}$  Larmor frequency) and a solution of  $\text{Al}(\text{NO}_3)_3$  in  $\text{D}_2\text{O}$  at 0 ppm was used as an external reference. The lack of solubility of compound **2** prevented its characterization by solution NMR.

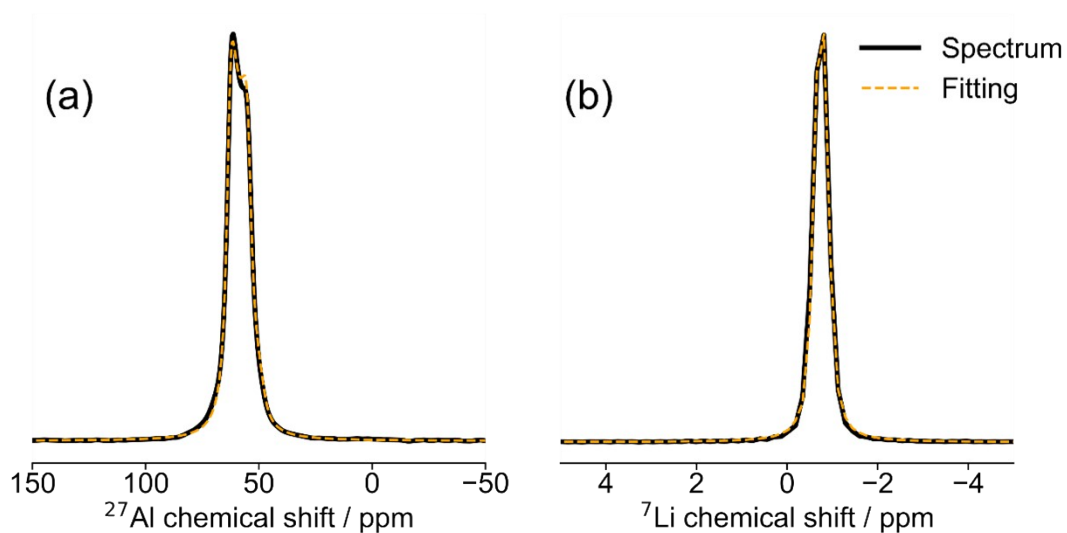
## S3.2. Solid-state NMR spectra



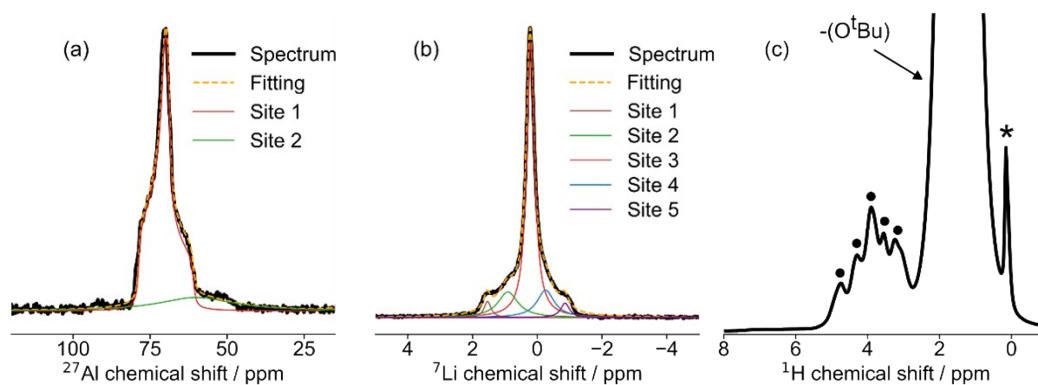
**Figure S3.2.1.** Solid-state NMR (SSNMR) spectra of the pristine precursors **1** (red), **2** (blue) and **3** (green). The  $^7\text{Li}$  (a),  $^{27}\text{Al}$  (b) and  $^1\text{H}$  (c) NMR spectra are shown. The rotors were packed inside a  $\text{N}_2$ -filled glovebox to avoid hydrolysis of the precursors under air. The spectra of **3** were recorded at a magnetic field strength of 16.4 T and a MAS frequency of 50 kHz. The spectra of **1** and **2** were collected at a field strength of 11.7 T and a MAS frequency of 30 kHz. Only the central (isotropic) peaks are shown here. The  $^1\text{H}$  SSNMR spectra are consistent with the presence of the expected alkoxy groups for each compound. Compound **1** shows two resonances at 3.9 (- $\text{OCH}_2$ -) and 6.9 ppm (-Ph). Meanwhile, **2** presents two resonances at 4.12 (-OCH-) and 1.23 ppm (- $\text{CH}_3$ ) as expected for the -O $^i$ Pr group and compound **3** showed only one proton peak at 1.45 ppm as there is only one proton environment in the -O $^i$ Bu ligand.



**Figure S3.2.2.** SSNMR spectra and fittings obtained for compound **1**. (a)  $^{27}\text{Al}$  SSNMR spectrum and fitting. (b)  $^7\text{Li}$  SSNMR spectrum and fitting.



**Figure S3.2.3.** SSNMR spectra and fittings obtained for compound **2**.  $^{27}\text{Al}$  SSNMR spectrum and fitting. (b)  $^7\text{Li}$  SSNMR spectrum and fitting.



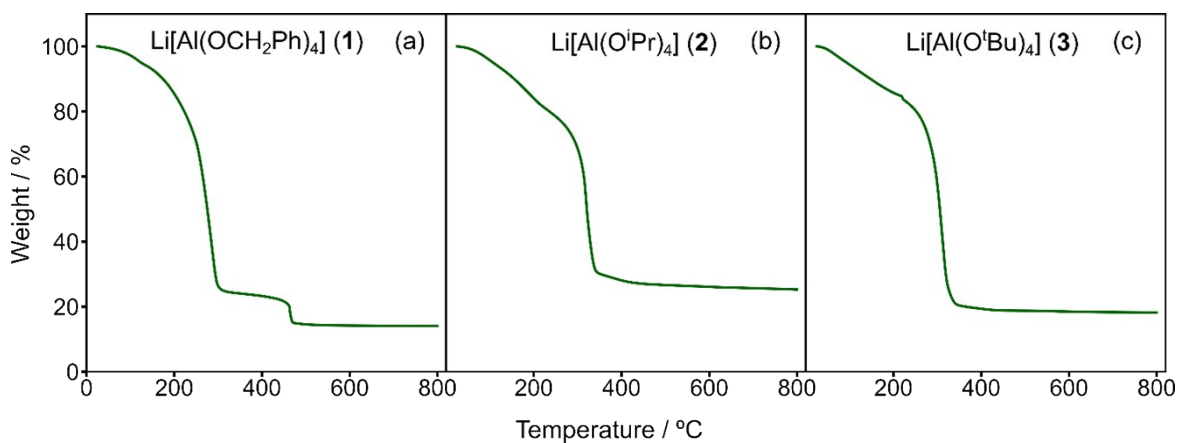
**Figure S3.2.4.** SSNMR and fittings for the  $^{27}\text{Al}$  (a),  $^7\text{Li}$  (b) and  $^1\text{H}$  (c) spectra of precursor **3**. The spectra were acquired at a magnetic field strength of 16.4 T and a MAS frequency of 50 kHz. The circles mark the peaks arising from the five hydric environments, and the asterisk marks the proton peak corresponding to vacuum grease.

**Table S3.2.1.** SSNMR fitting parameters

Nucleus	Compound	Site	$\delta_{\text{iso}}$	$C_Q$ / kHz	$\nu_Q$ / kHz	$\eta_Q$	
$^{27}\text{Al}$	1	1	72.9	4559	684	0.111	
		2	6.12	55	8	0.058	
		2	1	65.9	4471	671	-0.108
		3	1	79.0	5961	894	-0.912
		2	2	61.2	55	8	0.152
$^7\text{Li}$	1	1	-0.809	45	23	0.192	
		2	0.284	98	49	-0.051	
	2	1	-0.499	533	266	0.02	
		3	1	1.53	75	38	0.06
	2		0.926	73	37	0.019	
	3		0.232	0	0	0	
	4	-0.221	140	70	0.09		
5	-0.807	71	35	0.07			

## S4. Precursor thermolysis

### S4.1. Thermogravimetric analysis

**Figure S4.1.1.** TGA curves of precursors **1** (a), **2** (b) and **3** (c) measured in air atmosphere between 25 and 800 °C with a heating rate of 10 °C/min.

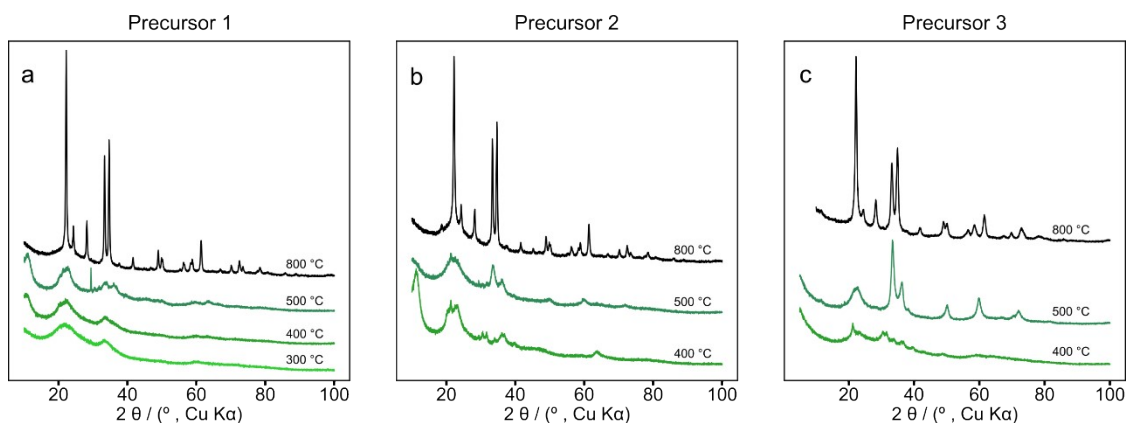


**Table S4.1.1.** TGA decomposition temperature, % mass loss and Al : Ni ratios after decomposition. The decomposition temperatures were calculated from the extrapolated onset temperatures obtained from the main decomposition step. ICP-OES measurements were carried out on samples annealed at  $T > T_d$ .

<b>Precursor</b>	<b><math>T_d</math> / °C</b>	<b>Expected mass loss (%)</b>	<b>Measured mass loss (%)</b>	<b>Li : Al ratio (ICP)</b>
Li[Al(OCH <sub>2</sub> Ph) <sub>4</sub> ] ( <b>1</b> )	236	86	86	0.876 (400 °C)
				0.914 (800 °C)
Li[Al(O <sup>i</sup> Pr) <sub>4</sub> ] ( <b>2</b> )	302	76	75	0.976 (400 °C)
Li[Al(O <sup>t</sup> Bu) <sub>4</sub> ] ( <b>3</b> )	283	80	82	1.003 (500 °C)

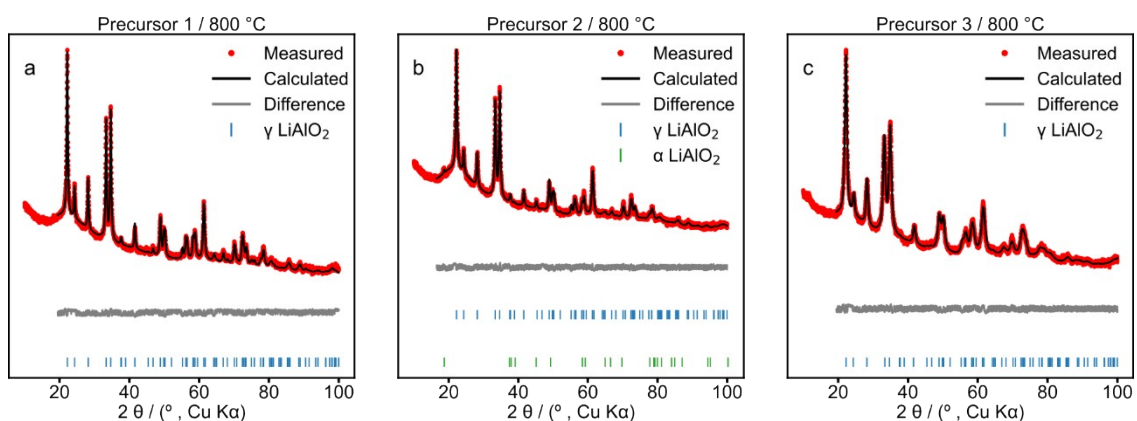
## S4.2. Powder X-ray diffraction

### S4.2.1. Diffraction patterns obtained for the thermolysis products of 1-3



**Figure S4.2.1.1.** PXRD patterns of compounds **1-3** decomposed by annealing under air for 4 h at 300, 400, 500 and 800 °C for compound **1** and 400, 500 and 800 °C for compound **2** and **3**.

### S4.2.2. Rietveld refinements results for the thermolysis products of 1-3



**Figure S4.2.2.1.** PXRD patterns of precursors **1-3** annealed under air at 800 °C for 4 h together with the fittings obtained from Rietveld refinement.

**Table S4.2.2.1.** Atomic coordinates and site occupancies of precursor **1** annealed at 800 °C.

Site	Atom	x	y	z	Occupancy
4a	Li1	0.795	0.795	0	1
	Al1	0.186	0.186	0	1
8b	O1	0.353	0.295	0.778	1

**Table S4.2.2.2.** Atomic coordinates and site occupancies of the  $\gamma$  – LiAlO<sub>2</sub> phase formed after decomposition of precursor **2** at 800 °C under air.

Site	Atom	x	y	z	Occupancy
4a	Li1	0.838	0.838	0	1
	Al1	0.178	0.178	0	1
8b	O1	0.338	0.286	0.779	1

**Table S4.2.2.3.** Atomic coordinates and site occupancies of the  $\alpha$  – LiAlO<sub>2</sub> phase formed after decomposition of precursor **2** at 800 °C under air.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	1
3b	Al1	0	0	0.5	1
6c	O1	0	0	0.249	1

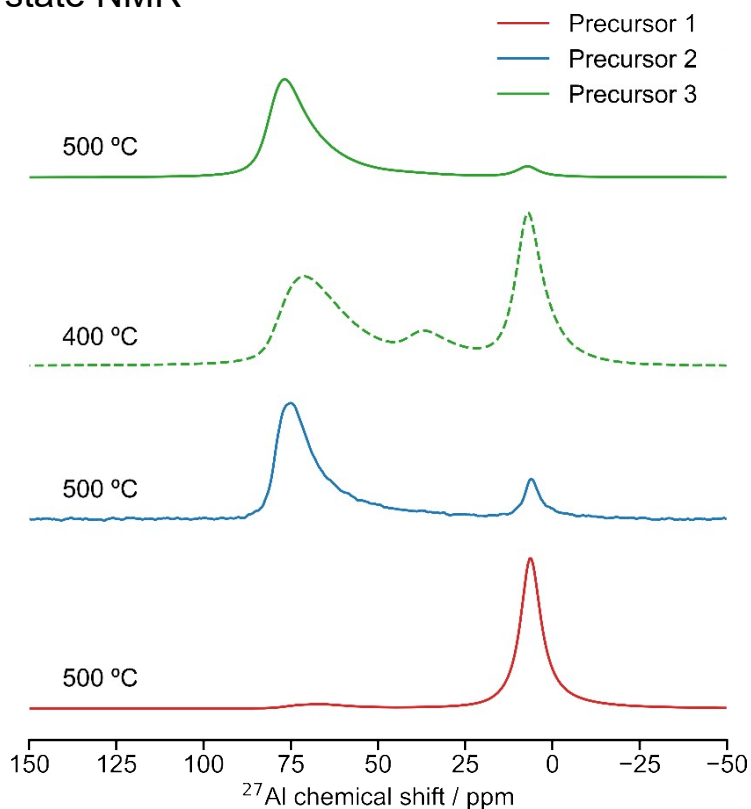
**Table S4.2.2.4.** Atomic coordinates and site occupancies of precursor **3** annealed at 800 °C.

Site	Atom	x	y	z	Occupancy
4a	Li1	0.815	0.815	0	1
	Al1	0.184	0.184	0	1
8b	O1	0.351	0.282	0.783	1

**Table S4.2.2.5.** Structure parameters obtained from Rietveld refinement of the PXRD patterns. The samples are compounds **1-3** annealed at 800 °C for 4 h.

Sample	R <sub>wp</sub> / %	Refined phases	Space group	Wt %	Space		
					a / Å	b / Å	c / Å
Precursor <b>1</b>	5.564	$\gamma$ -LiAlO <sub>2</sub>	P4 <sub>1</sub> 2 <sub>1</sub> 2	100	5.169	5.169	6.276
Precursor <b>2</b>	4.767	$\gamma$ -LiAlO <sub>2</sub>	P4 <sub>1</sub> 2 <sub>1</sub> 2	98.6	5.174	5.174	6.276
		$\alpha$ -LiAlO <sub>2</sub>	R <sup>3</sup> m	1.4	2.809	2.809	14.211
Precursor <b>3</b>	4.796	$\gamma$ -LiAlO <sub>2</sub>	P4 <sub>1</sub> 2 <sub>1</sub> 2	100	5.149	5.149	6.346

### S4.3. Solid-state NMR



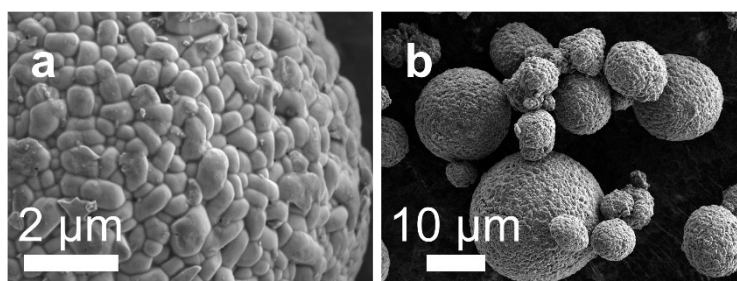
**Figure S4.3.1.** Solid-state  $^{27}\text{Al}$ -NMR spectra of the annealed compounds **1-3** at 500 °C and of **3** at 400 °C are shown. The spectra of **3** were recorded at a magnetic field strength of 16.4 T and a MAS frequency of 50 kHz. The spectra of **1** and **2** were collected at a field strength of 11.7 T and a MAS frequency of 30 kHz. Only the central (isotropic) peaks are shown.

## S5. Characterization of the coated NMC811 materials

### S5.1. Electron microscopy

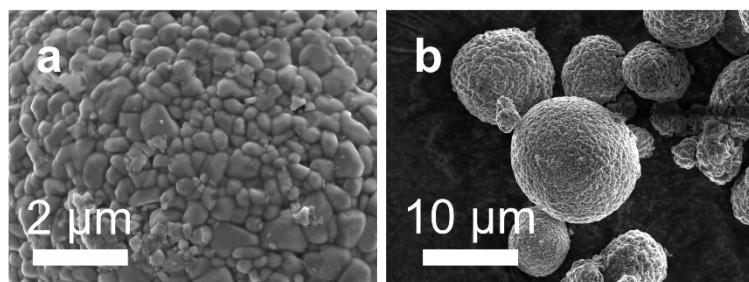
#### S5.1.1. Scanning electron microscopy

##### Pristine NMC811



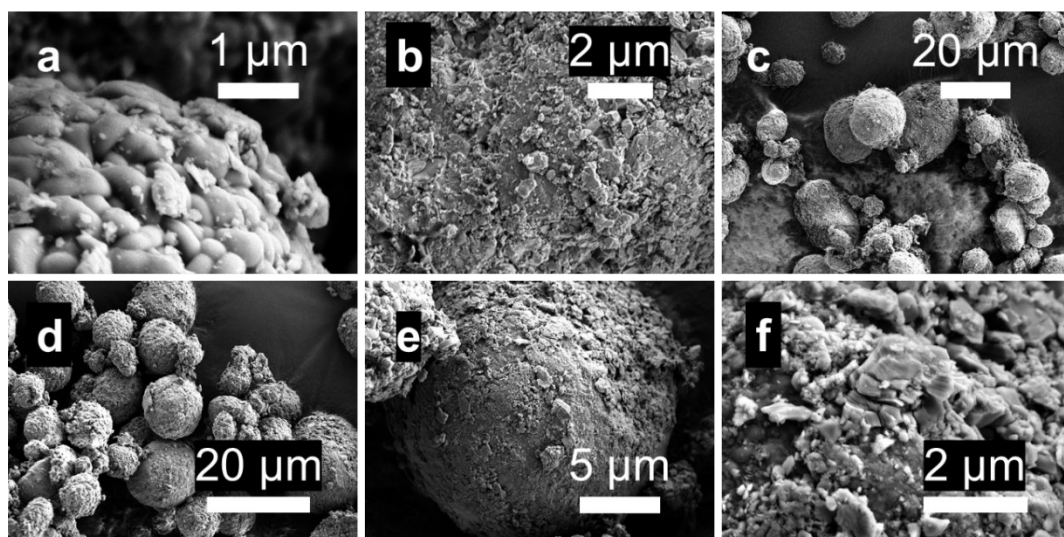
**Figure S5.1.1.1.** SEM images of pristine NMC811.

Coated with **1** / 400 °C - Air



**Figure S5.1.1.2.** SEM images of NMC811 coated using precursor **1** and then annealed at 400 °C under air.

Coated with **2** / 400 °C - Air



**Figure S5.1.1.3.** SEM images of NMC811 coated using precursor **2** and then annealed at 400 °C under air.

Coated with 3 / Not annealed

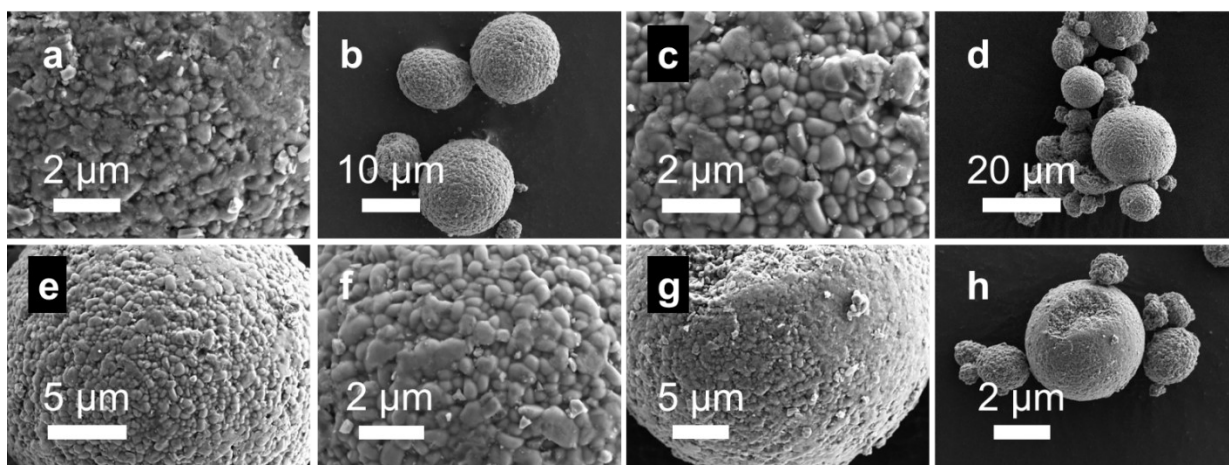


Figure S5.1.1.4. SEM images of NMC811 coated using precursor 3.

Coated with 3 / 400 °C - Air

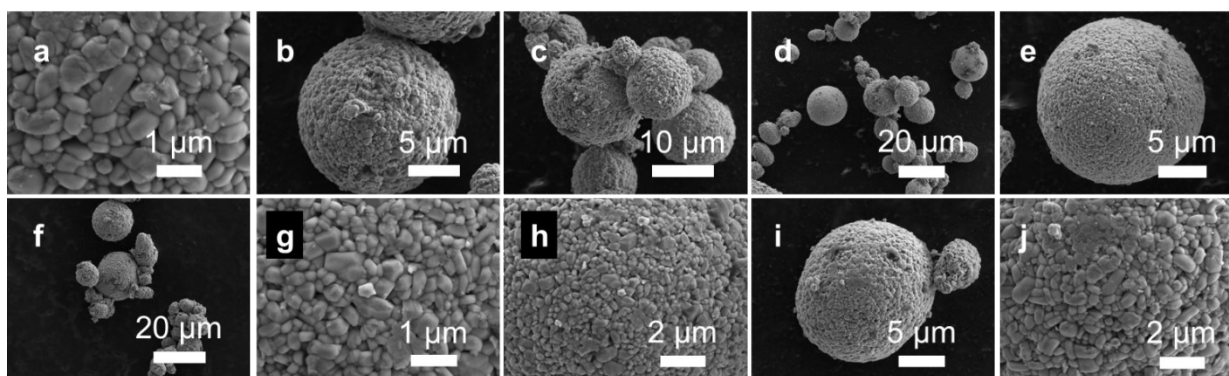


Figure S5.1.1.5. SEM images of NMC811 coated using precursor 3 and then annealed at 400 °C under air.

Coated with 3 / 400 °C - Air

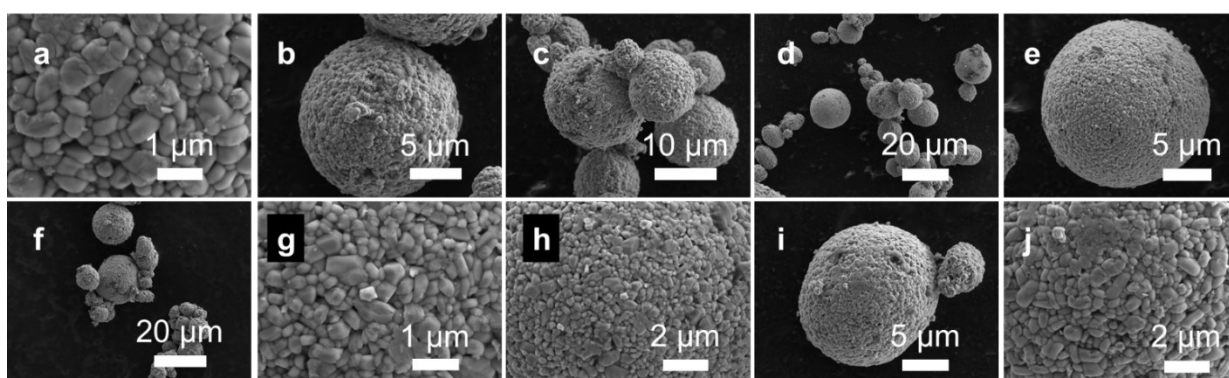


Figure S5.1.1.6. SEM images of NMC811 coated using precursor 3 and then annealed at 400 °C under oxygen

NMC811 soaked in water

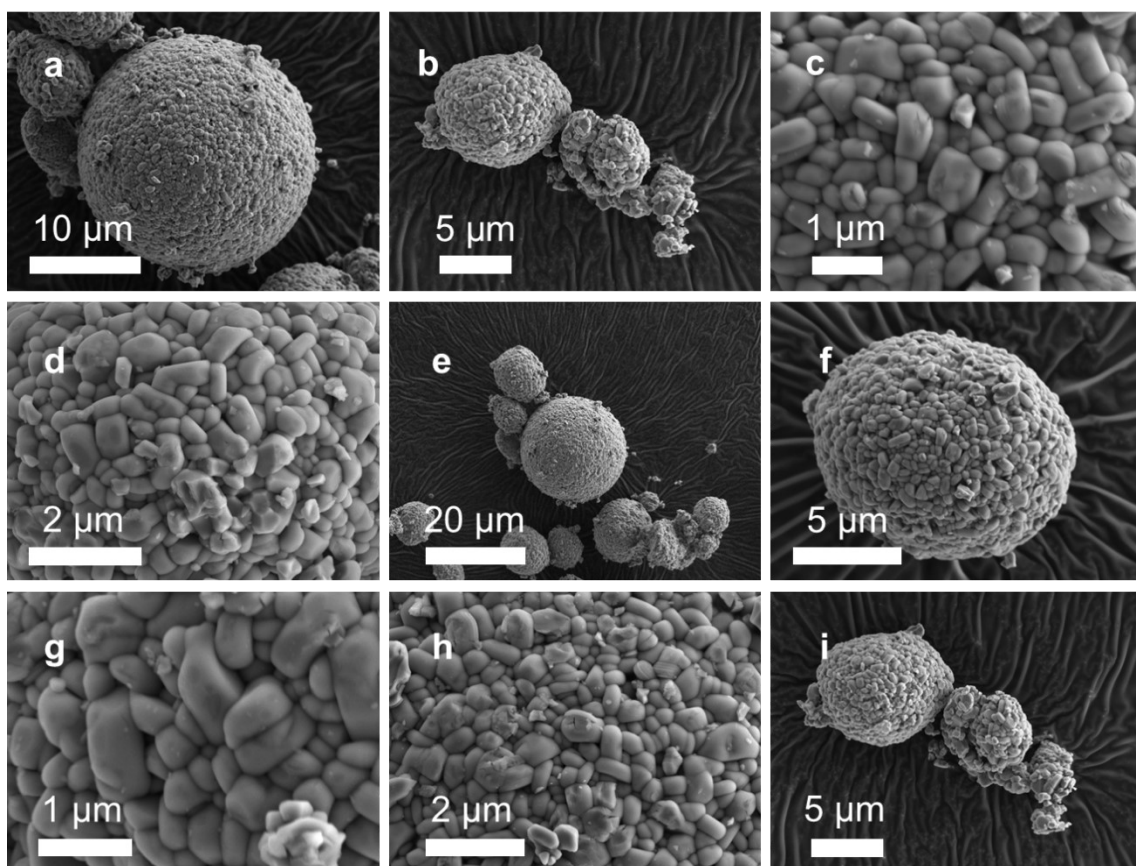


Figure S5.1.1.7. SEM images of NMC811 soaked in water.

NMC811 soaked in THF

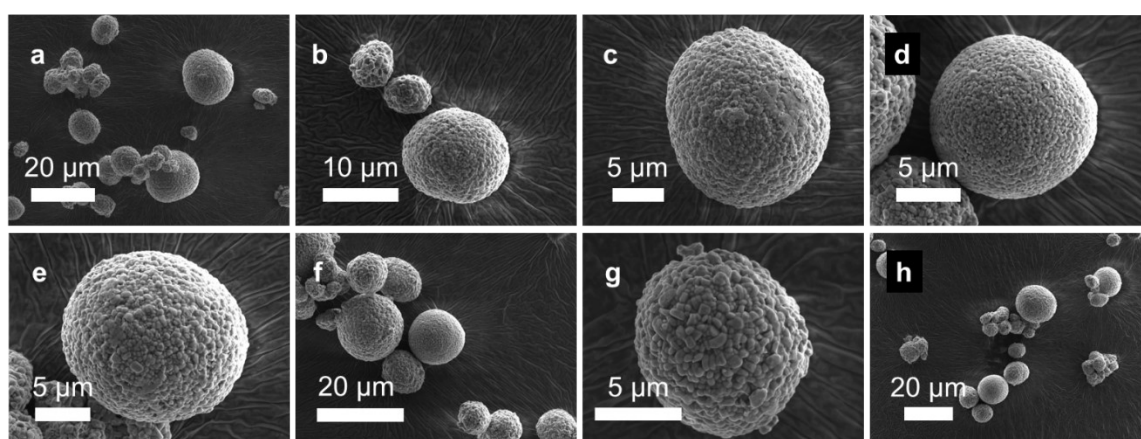
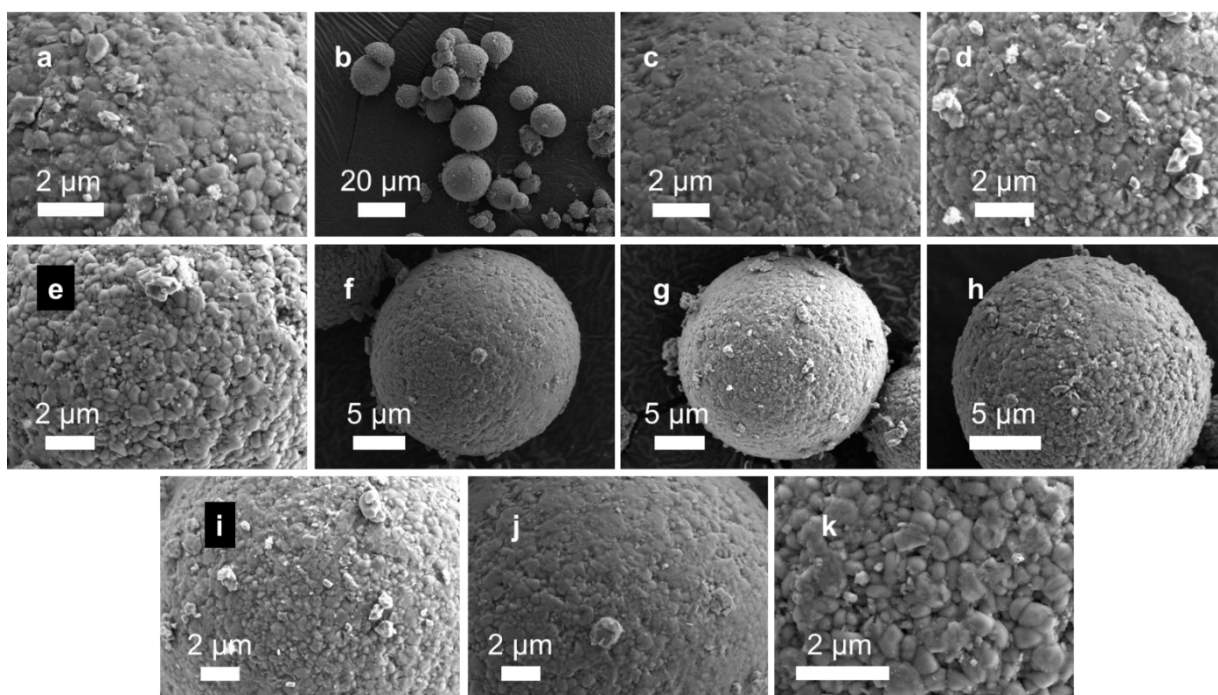


Figure S5.1.1.8. SEM images of NMC811 soaked in THF.

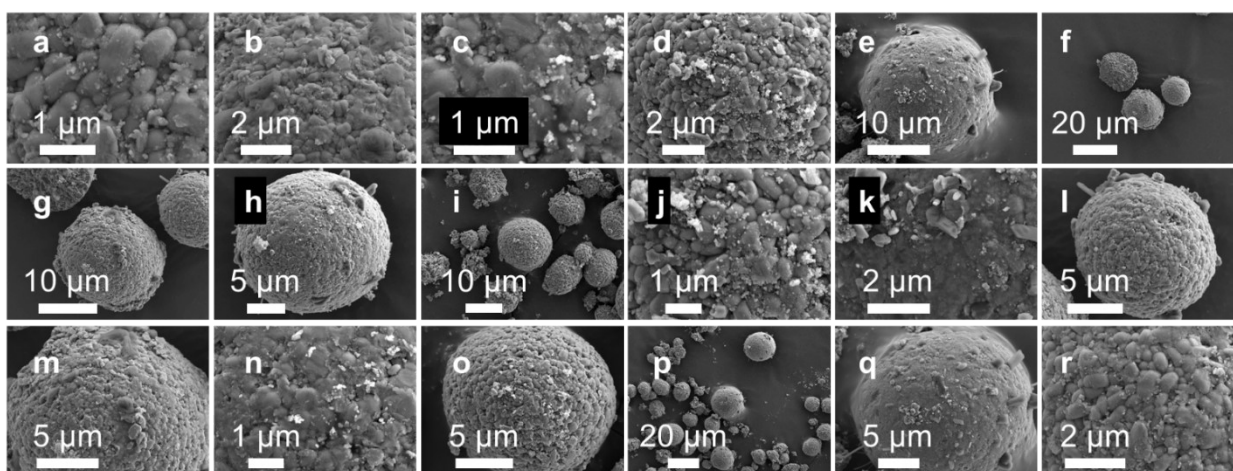


NMC811 soaked in water coated with **3** / Not annealed



**Figure S5.1.1.9.** SEM images of NMC811 soaked in water, coated using precursor **3**.

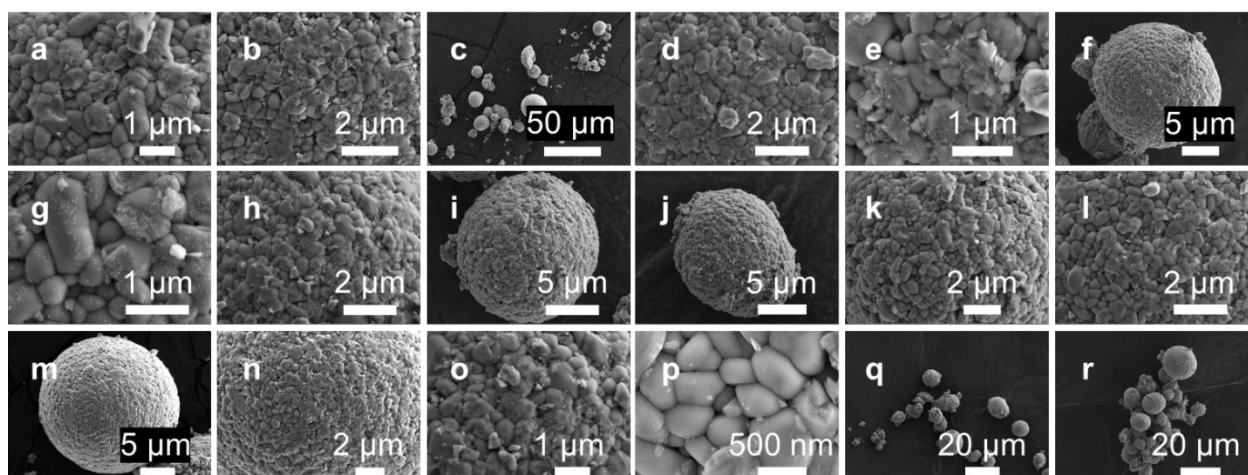
NMC811 soaked in water coated with **3** / 400 °C - air



**Figure S5.1.1.10.** SEM images of NMC811 soaked in water, coated using precursor **3** and annealed at 400 °C under air.

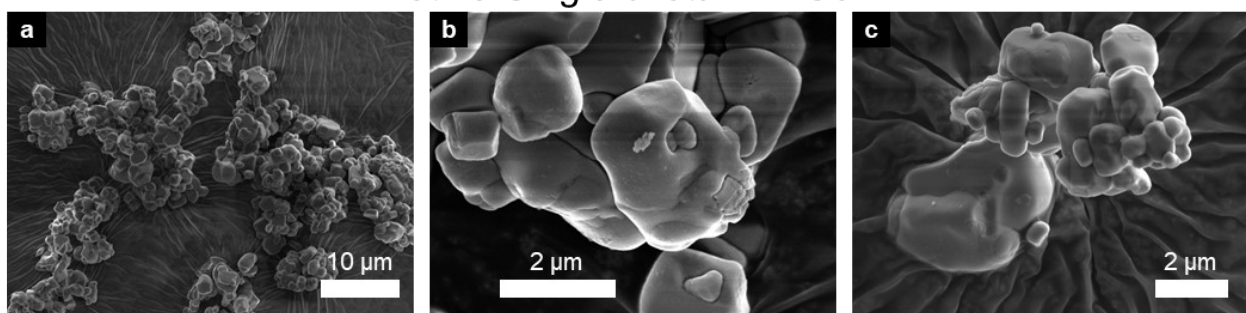


NMC811 soaked in water coated with **3** / 400 °C – O<sub>2</sub>



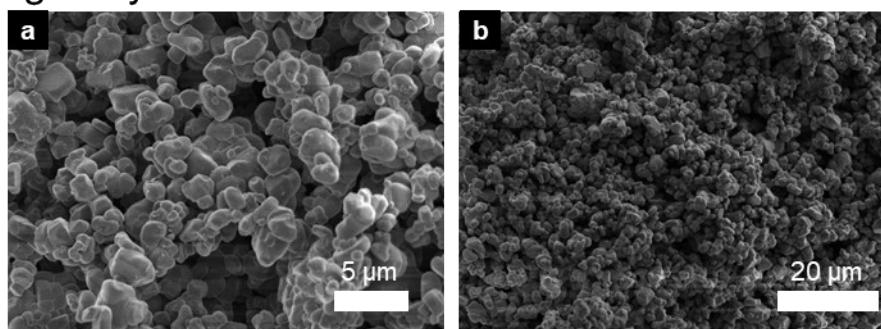
**Figure S5.1.1.11.** SEM images of NMC811 soaked in water, coated using precursor **3** and annealed at 400 °C under oxygen.

Pristine Single-crystal NMC811



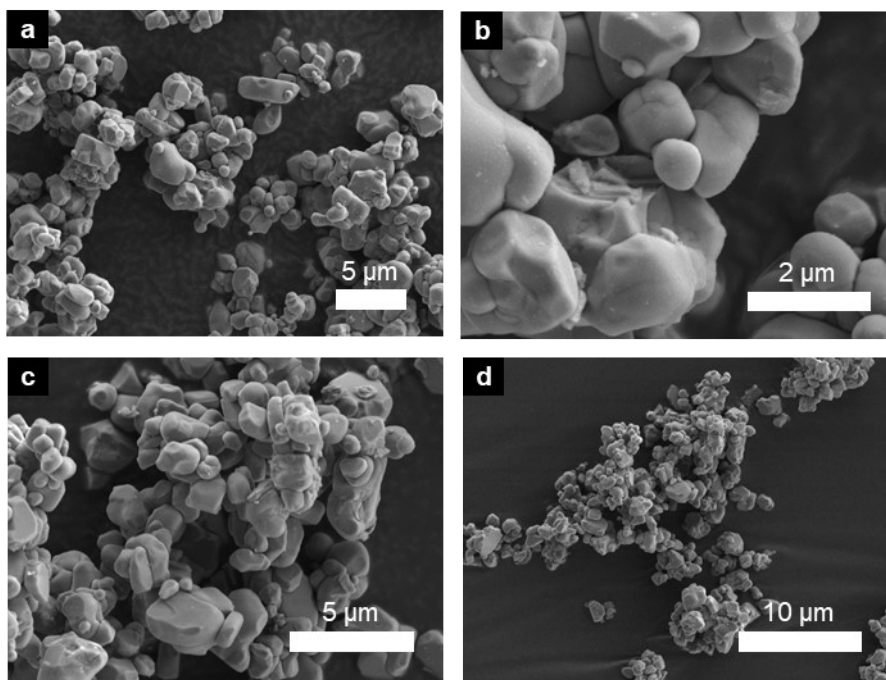
**Figure S5.1.1.12.** SEM images of pristine (Al<sub>2</sub>O<sub>3</sub> coated) single-crystal NMC811.

Single-crystal NMC811 coated with **3** / Air – 400 °C



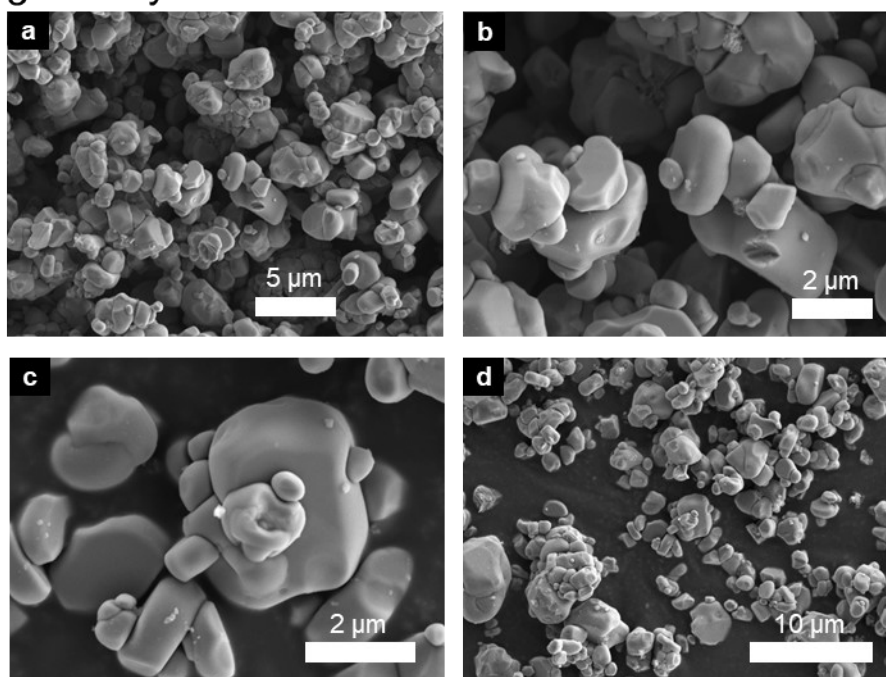
**Figure S5.1.1.13.** SEM images of (Al<sub>2</sub>O<sub>3</sub> coated) single-crystal NMC811 coated with **3** without a pre-soaking in water.

Single – crystal NMC811 soaked in water,  
coated with **3** / Air – 400 °C



**Figure S5.1.1.14.** SEM images of (Al<sub>2</sub>O<sub>3</sub> coated) single-crystal NMC811 soaked in water and coated with **3**.

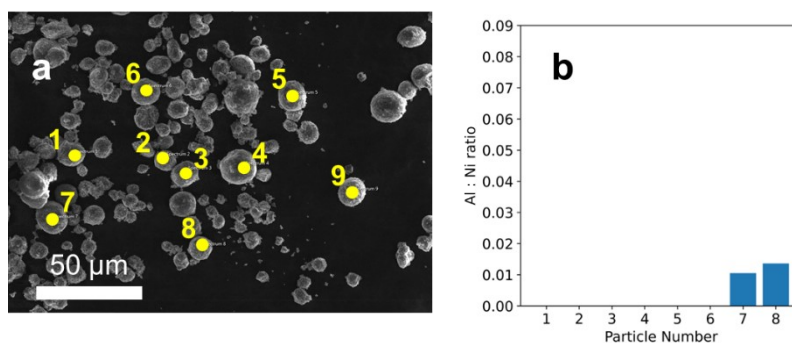
Single – crystal NMC811 soaked in THF/ Air – 400 °C



**Figure S5.1.1.15.** SEM images of the control (Al<sub>2</sub>O<sub>3</sub> coated) single-crystal NMC811 sample soaked in THF and annealed at 400 °C under air.

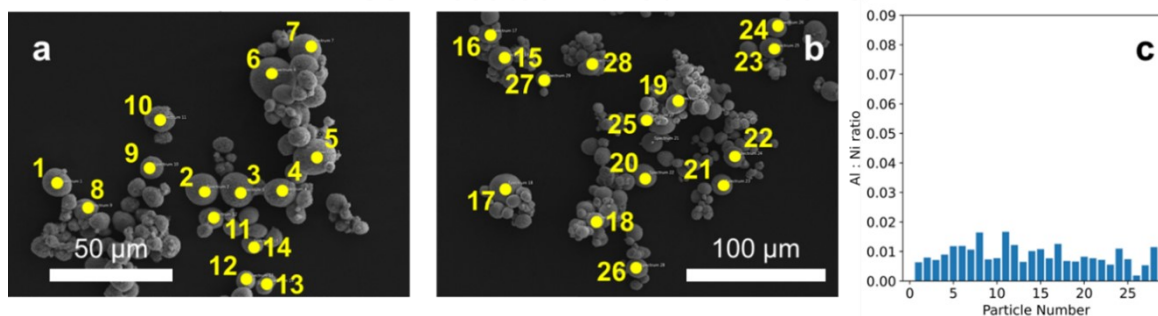
### S5.1.2. Energy-dispersive X-ray spectroscopy

#### NMC811 coated with 2 / Air - 400 °C



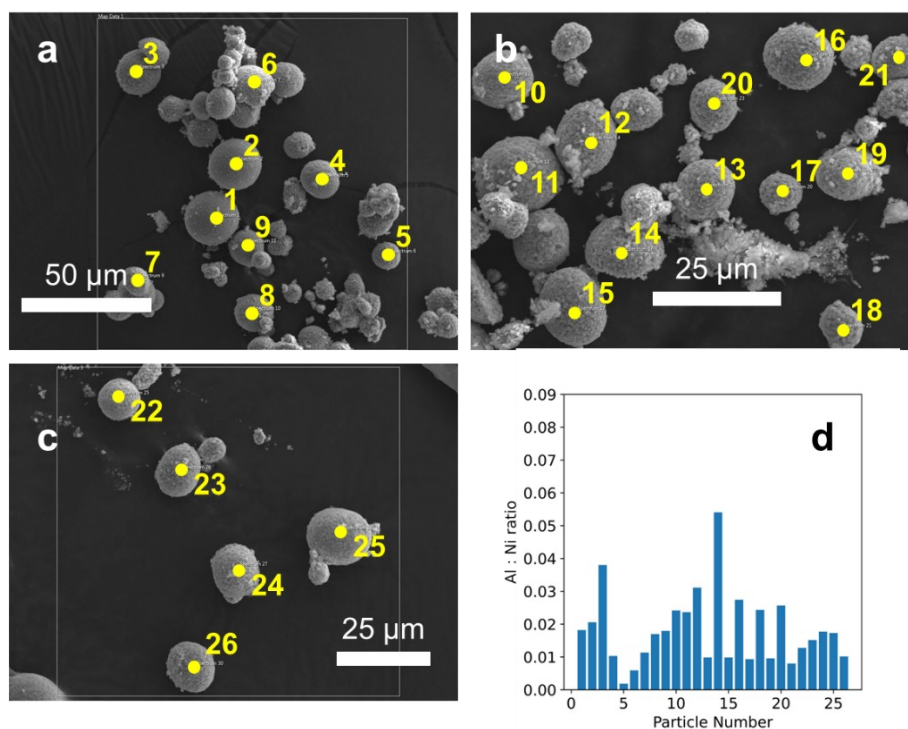
**Figure S5.1.2.1.** (a) SEM images and locations of the EDS point scans taken for NMC811 using precursor 2 and annealed at 400 °C under air. (c) Al : Ni ratios calculated for each of the particles.

#### NMC811 coated with 3 / Air - 400 °C



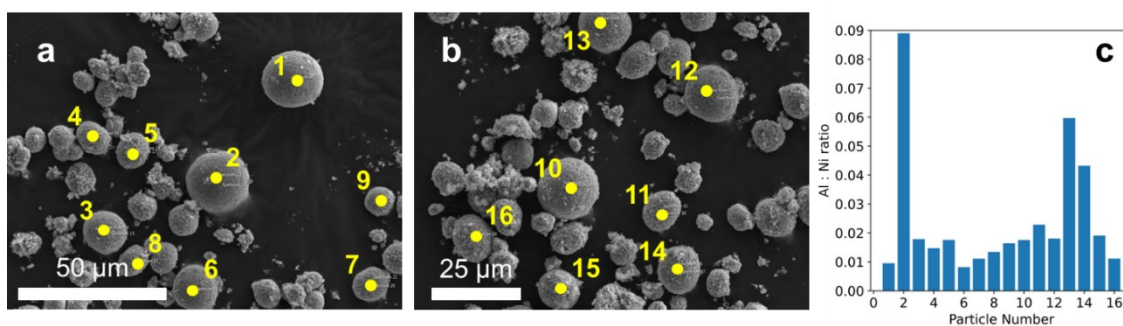
**Figure S5.1.2.2.** (a) SEM images and locations of the EDS point scans taken for NMC811 using precursor 3 and annealed at 400 °C under air. (c) Al : Ni ratios calculated for each of the particles.

NMC811 soaked in water coated with **3** / Not annealed



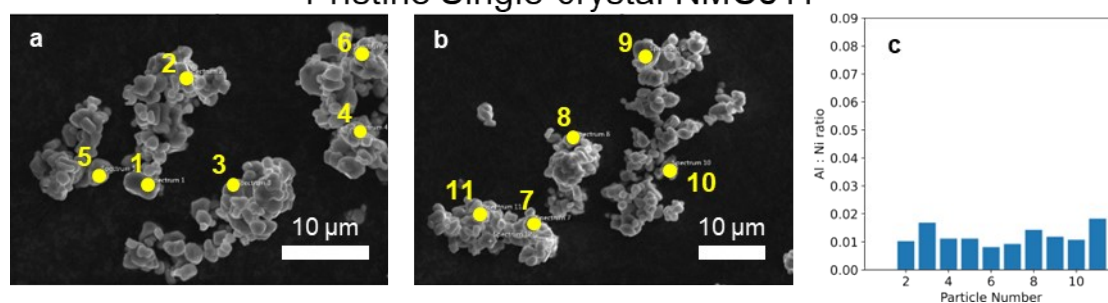
**Figure S5.1.2.3.** (a -c) SEM images and locations of the EDS point scans taken for NMC811 soaked in water and coated using precursor 3. (c) Al : Ni ratios calculated for each of the particles.

NMC811 soaked in water coated with **3** / Air - 400 °C



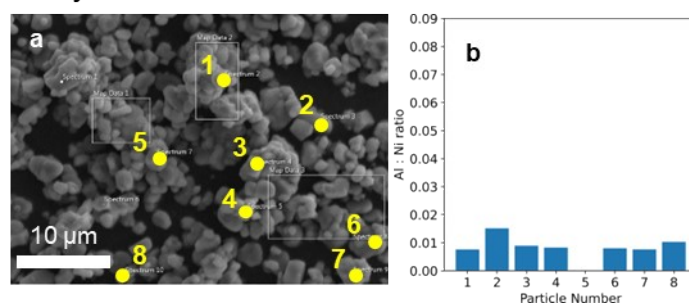
**Figure S5.1.2.4.** (a -b) SEM images and locations of the EDS point scans taken for NMC811 soaked in water, coated using precursor 3 and annealed at 400 °C under air. (c) Al : Ni ratios calculated for each of the particles.

### Pristine Single-crystal NMC811



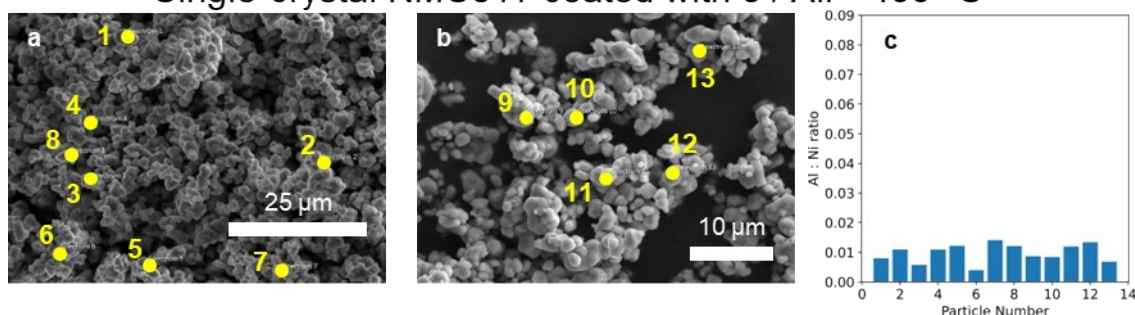
**Figure S5.1.2.5.** (a, b) SEM images and locations of the EDS point scans taken for the pristine ( $\text{Al}_2\text{O}_3$  coated) single crystal NMC811. (c) Al : Ni ratios calculated for each of the particles.

### Single-crystal NMC811 soaked in THF / Air - 400 °C



**Figure S5.1.2.6.** (a, b) SEM images and locations of the EDS point scans taken for the pristine ( $\text{Al}_2\text{O}_3$  coated) single crystal NMC811. (c) Al : Ni ratios calculated for each of the particles.

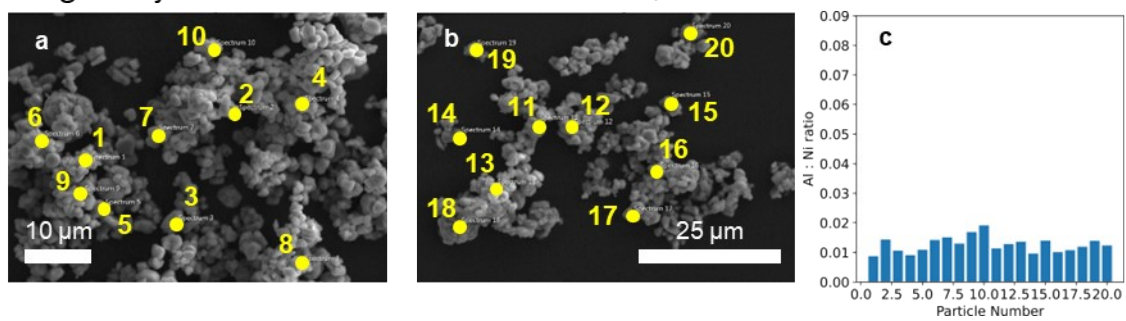
### Single-crystal NMC811 coated with 3 / Air - 400 °C



**Figure S5.1.2.7.** (a, b) SEM images and locations of the EDS point scans taken for ( $\text{Al}_2\text{O}_3$  coated) single crystal NMC811 coated with 3. (c) Al : Ni ratios calculated for each of the particles.



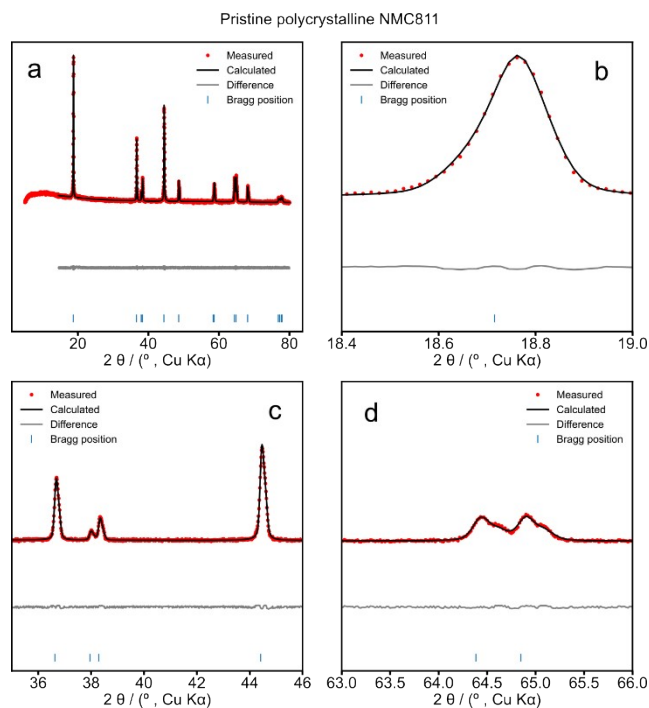
Single-crystal NMC811 soaked in water, coated with **3** / Air - 400 °C



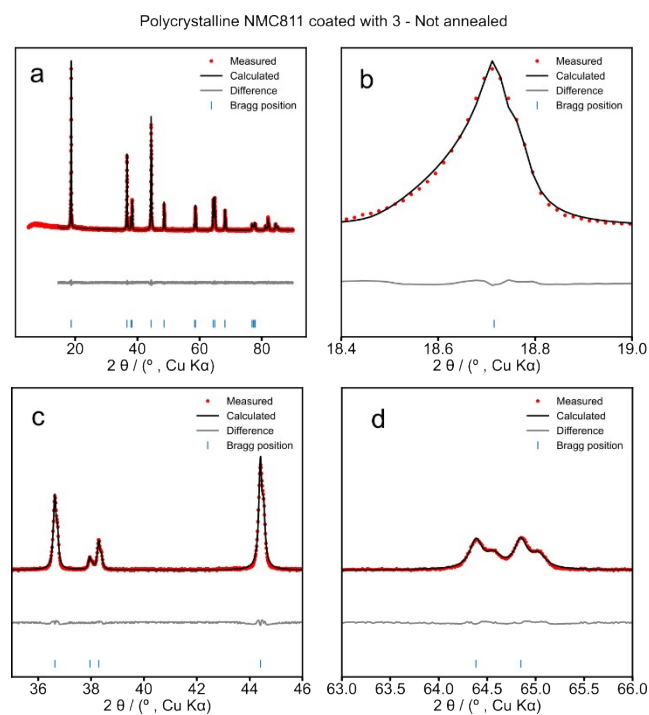
**Figure S5.1.2.8.** (a, b) SEM images and locations of the EDS point scans taken for (Al<sub>2</sub>O<sub>3</sub> coated) single crystal NMC811 soaked in water and coated with **3**. (c) Al : Ni ratios calculated for each of the particles.

## S5.2. Powder X-ray diffraction

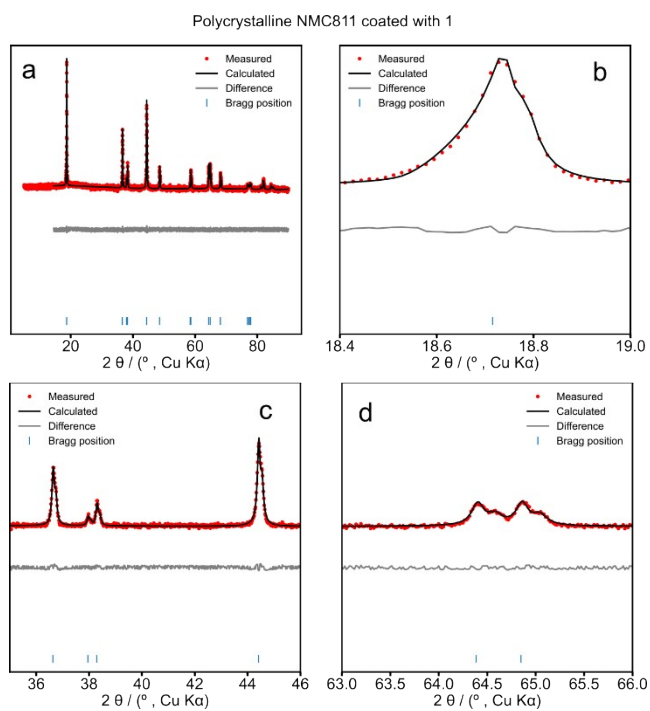
### S5.2.1. Laboratory Powder X-ray Diffraction Patterns



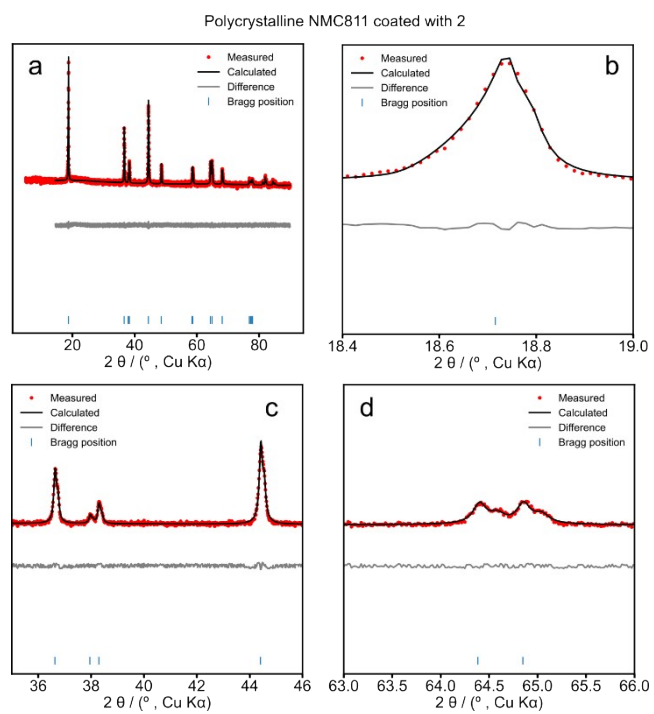
**Figure S5.2.1.1.** Polycrystalline NMC811 (pristine).



**Figure S5.2.1.2.** Polycrystalline NMC811 coated with **3** not annealed.

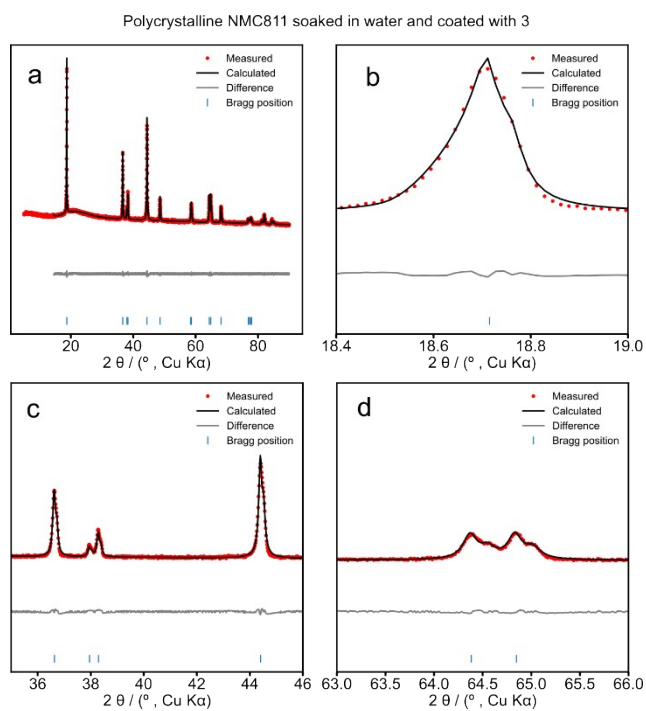


**Figure S5.2.1.3.** Polycrystalline NMC811 coated with 1.

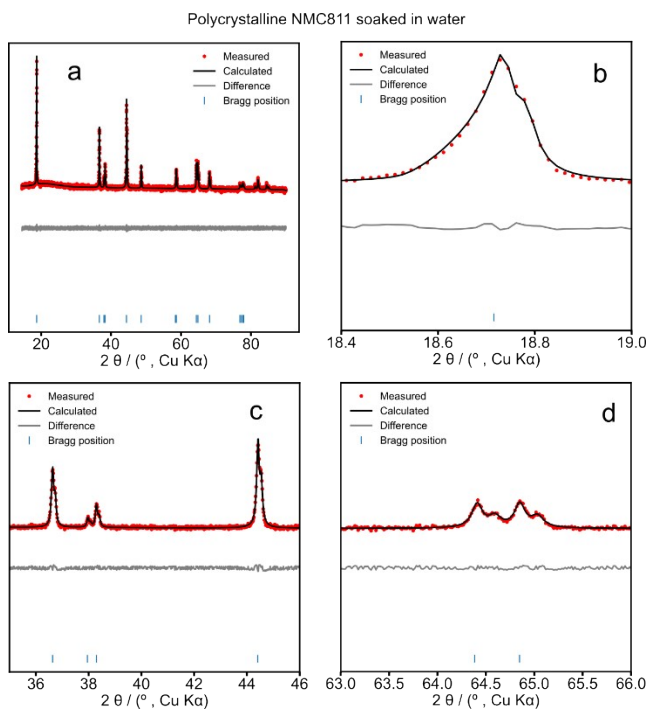


**Figure S5.2.1.4.** Polycrystalline NMC811 coated with 2.



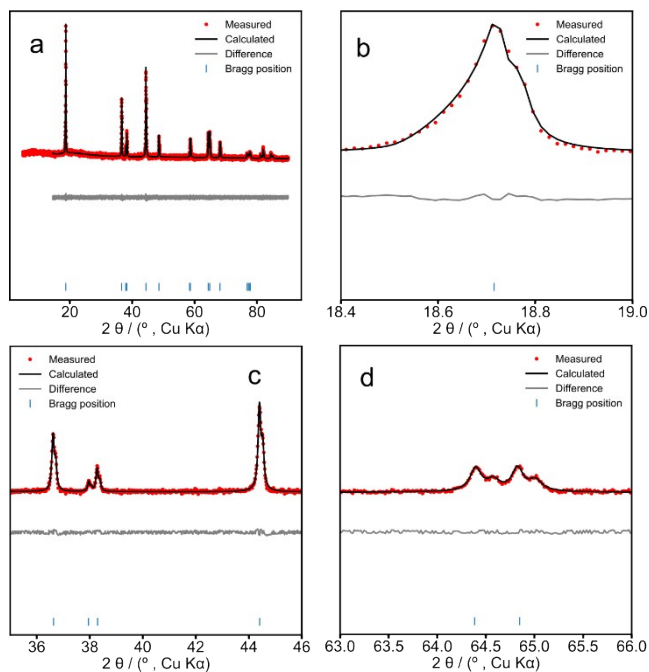


**Figure S5.2.1.5.** Polycrystalline NMC811 soaked in water and coated with 3.



**Figure S5.2.1.6.** Polycrystalline NMC811 soaked in water (not annealed).

Polycrystalline NMC811 soaked in water and annealed at 400 °C under air



**Figure S5.2.1.7.** Polycrystalline NMC811 soaked in water annealed at 400 °C.

### S5.2.2. Refinement Results for the Laboratory PXRD data

**Table S5.2.2.1.** Pristine polycrystalline NMC811 atomic coordinates and site occupancies.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	0.985
	Ni1	0	0	0	0.0152
3b	Ni2	0	0	0.5	0.785
	Co1	0	0	0.5	0.1
	Mn1	0	0	0.5	0.1
6c	O1	0	0	0.242	1

**Table S5.2.2.2.** Atomic coordinates and site occupancies polycrystalline NMC811 coated using precursor 1 and annealed under air at 400 °C.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	0.984
	Ni1	0	0	0	0.0164
3b	Ni2	0	0	0.5	0.784
	Co1	0	0	0.5	0.1
	Mn1	0	0	0.5	0.1
6c	O1	0	0	0.243	1

**Table S5.2.2.3.** Atomic coordinates and site occupancies polycrystalline NMC811 coated using precursor 2 and annealed under air at 400 °C.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	0.985
	Ni1	0	0	0	0.0147
3b	Ni2	0	0	0.5	0.785
	Co1	0	0	0.5	0.1
	Mn1	0	0	0.5	0.1
6c	O1	0	0	0.243	1

**Table S5.2.2.4.** Atomic coordinates and site occupancies polycrystalline NMC811 coated using precursor 3 and dried under vacuum at 100 °C.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	0.983
	Ni1	0	0	0	0.0174
3b	Ni2	0	0	0.5	0.783
	Co1	0	0	0.5	0.1
	Mn1	0	0	0.5	0.1
6c	O1	0	0	0.242	1

**Table S5.2.2.5.** Atomic coordinates and site occupancies NMC811 soaked in water and dried under vacuum at 100 °C.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	0.984
	Ni1	0	0	0	0.0158
3b	Ni2	0	0	0.5	0.784
	Co1	0	0	0.5	0.1
	Mn1	0	0	0.5	0.1
6c	O1	0	0	0.243	1

**Table S5.2.2.6.** Atomic coordinates and site occupancies NMC811 soaked in water and annealed under air at 400 °C.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	0.978
	Ni1	0	0	0	0.0216
3b	Ni2	0	0	0.5	0.778
	Co1	0	0	0.5	0.1
	Mn1	0	0	0.5	0.1
6c	O1	0	0	0.242	1

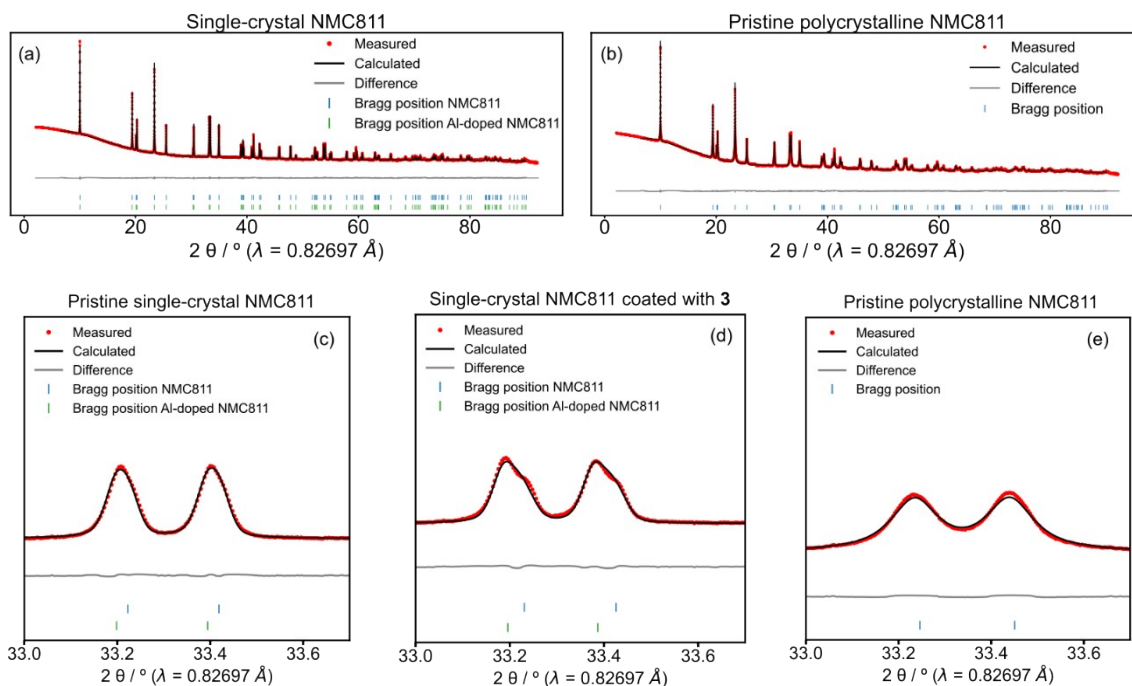
**Table S5.2.2.7.** Atomic coordinates and site occupancies NMC811 soaked in water, coated with **3** and annealed under air at 400 °C.

Site	Atom	x	y	z	Occupancy
3a	Li1	0	0	0	0.98
	Ni1	0	0	0	0.0202
3b	Ni2	0	0	0.5	0.78
	Co1	0	0	0.5	0.1
	Mn1	0	0	0.5	0.1
6c	O1	0	0	0.244	1

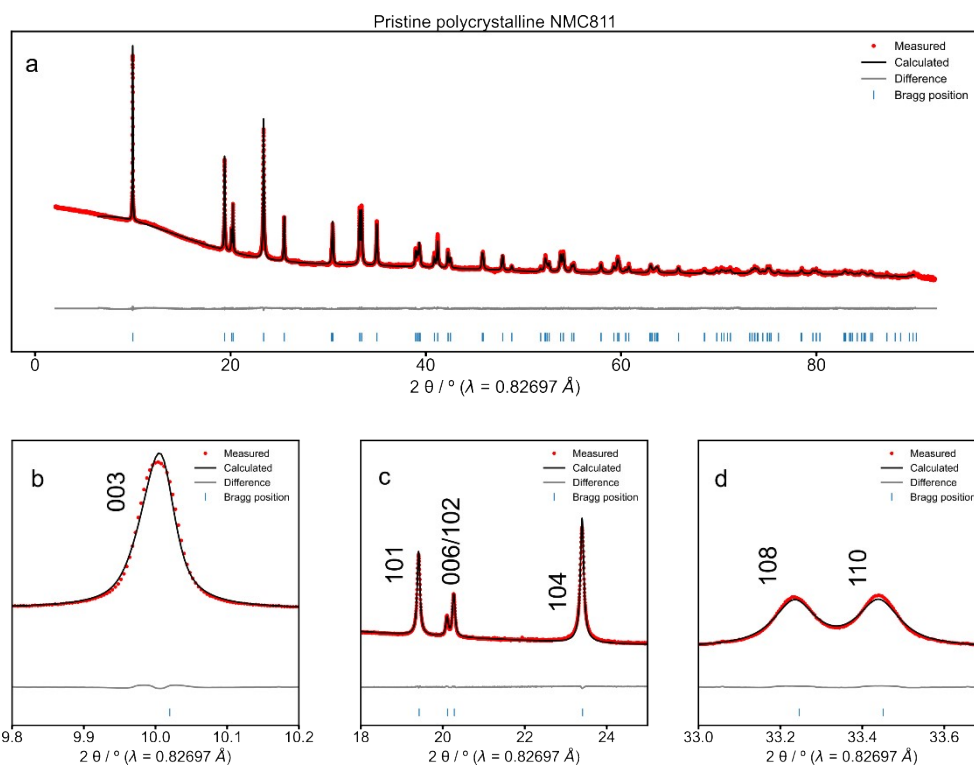
**Table S5.2.2.8.**  $R_{wp}$  and lattice parameters obtained from Rietveld refinement of laboratory data.

Sample	$R_{wp}$ / %	a / Å	b / Å	c / Å
NMC811	1.906	2.873	2.873	14.212
Coated with <b>1</b> / 400 °C (air)	5.076	2.874	2.874	14.213
Coated with <b>2</b> / 400 °C (air)	5.112	2.874	2.874	14.212
Coated with <b>3</b> / 100 °C (vacuum)	2.723	2.873	2.873	14.212
Soaked in water	4.873	2.874	2.874	14.207
Soaked in water / 400 °C (air)	4.802	2.874	2.874	14.206
Soaked in water - Coated with <b>3</b> / 400 °C (air)	3.032	2.874	2.874	14.214

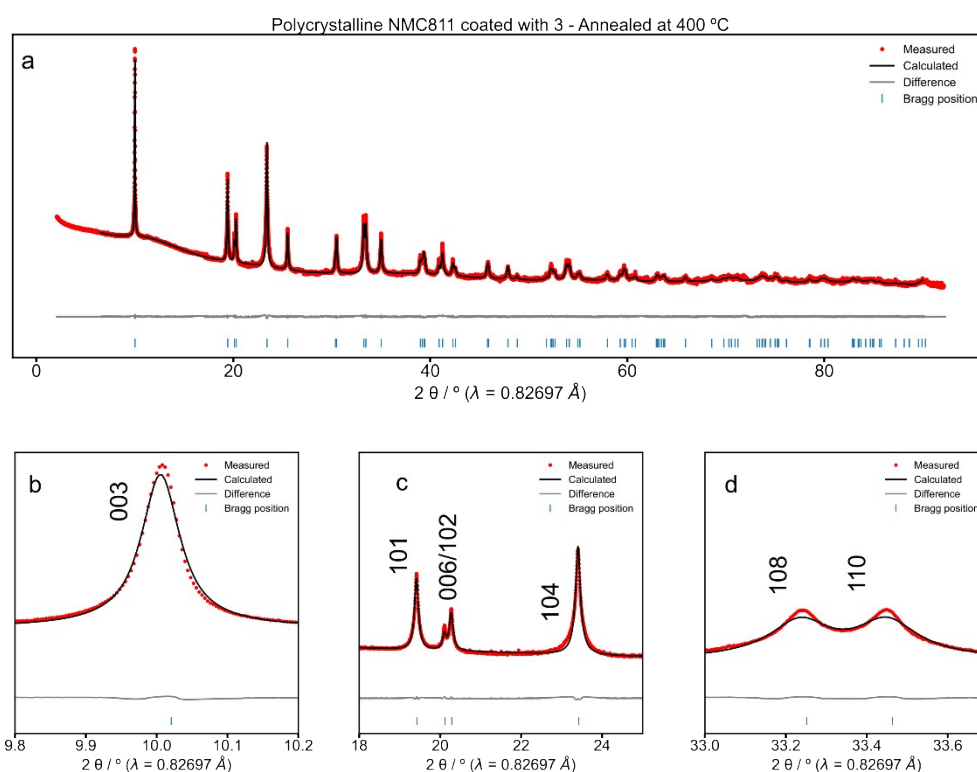
### S5.2.3. Synchrotron Powder X-ray Diffraction Patterns



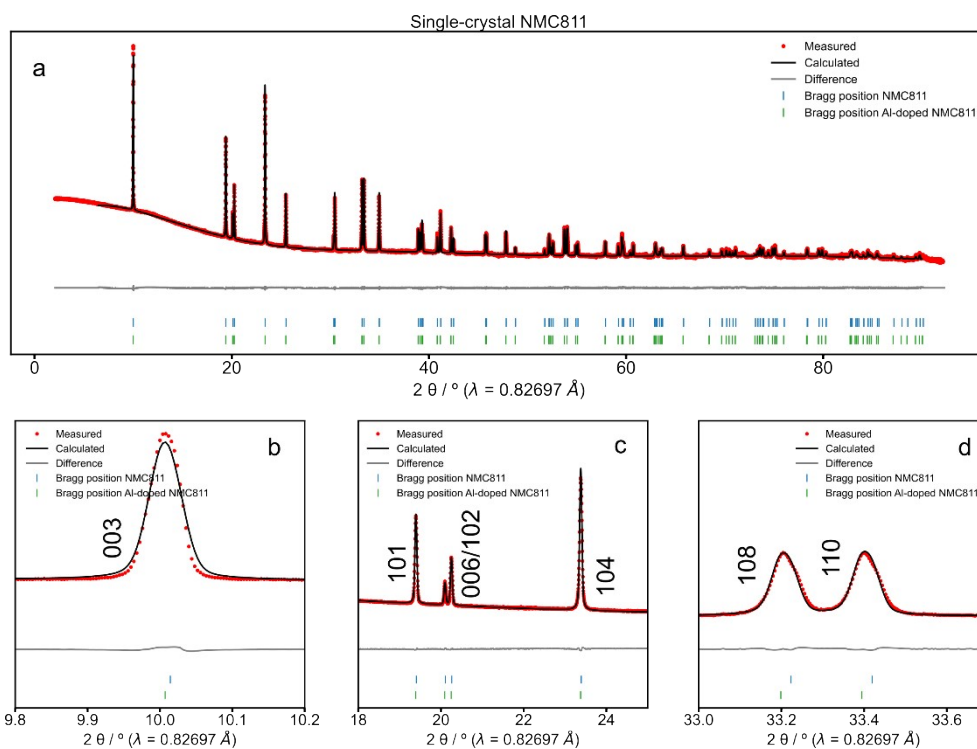
**Figure S5.2.3.1.** Synchrotron powder XRD patterns and Rietveld refinements of (a) pristine SC-NMC811, (b) pristine PC-NMC811, (c) zoom-in on 108 and 110 reflections of pristine SC-NMC811, (d) zoom-in on 108 and 110 reflections of SC-NMC811 coated with **3** and annealed at 400 °C under air for 4 h. (e) zoom-in on 108 and 110 reflections of PC-NMC811.



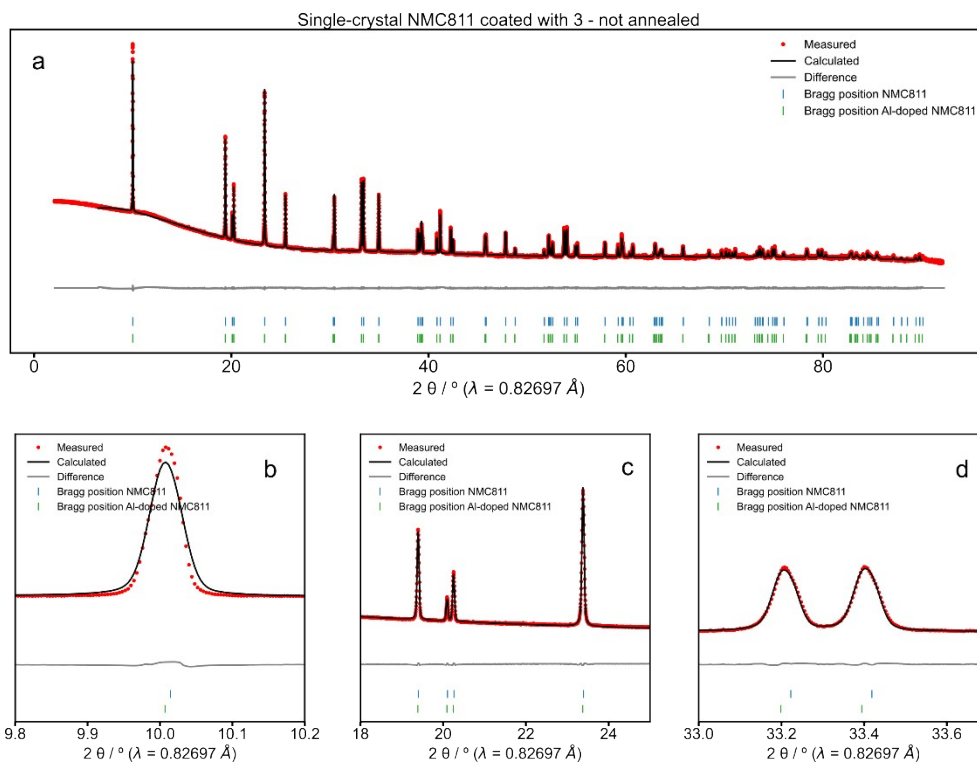
**Figure S5.2.3.2.** Synchrotron powder XRD patterns and Rietveld refinements of polycrystalline NMC811 (pristine).



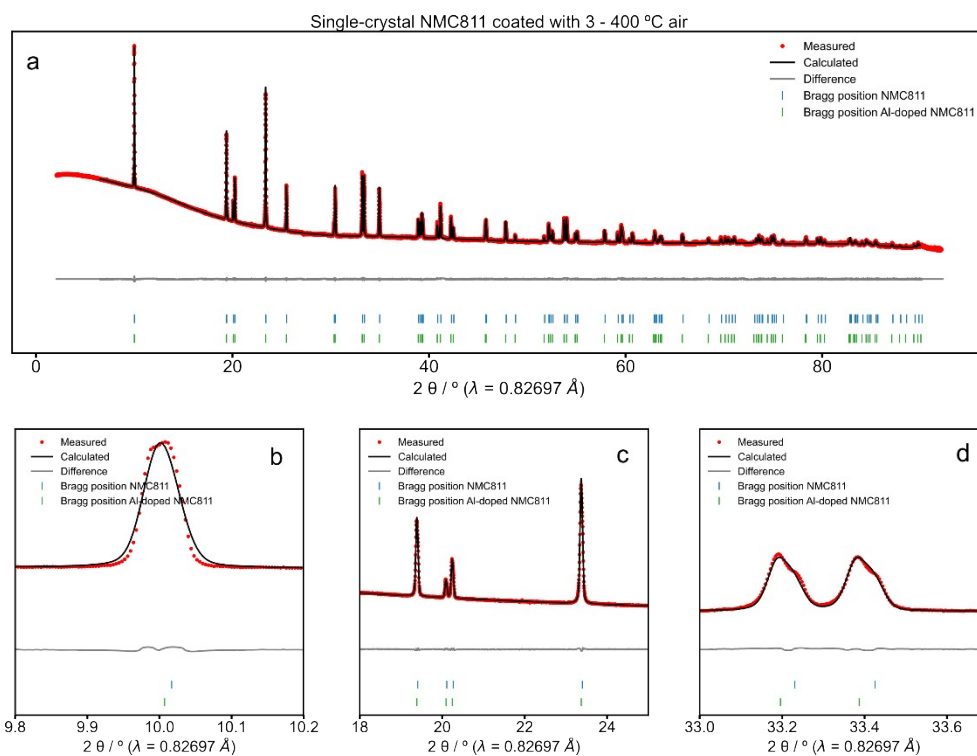
**Figure S5.2.3.3.** Synchrotron powder XRD patterns and Rietveld refinements of polycrystalline NMC811 coated with **3** and annealed at  $400^\circ\text{C}$  under air.



**Figure S5.2.3.4.** Synchrotron powder XRD patterns and Rietveld refinements of single-crystal NMC811 (pristine).



**Figure S5.2.3.5.** Synchrotron powder XRD patterns and Rietveld refinements of single-crystal NMC811 coated with 3 and dried at  $100^\circ\text{C}$  under vacuum.



**Figure S5.2.3.6.** Synchrotron powder XRD patterns and Rietveld refinements of single-crystal NMC811 coated with **3** and annealed at 400 °C under air.



#### S5.2.4. Refinement Results for the Synchrotron PXRD data

The refinement of the PC-NMC811 showed the pattern of a single-phase material with a well-defined layered structure that could be indexed in the  $R\bar{3}m$  space group (Figures S5.2.3.1-S5.2.3.2 and Table S5.2.4.1). Furthermore, a low level of anti-site mixing (2.21% of Ni is Li sites) was observed. After coating with **3** and annealing, the PC-NMC811 structure shows a modest increase in cation mixing (3.58%) and a slight broadening of the peaks suggesting that the heat treatment induces some disorder (Figure S5.2.3.3 and Table S5.2.4.1). The Al-doped NMC811 phase is formed by substituting TM sites with Al ions and has larger c parameters compared to the undoped phase (Table S5.2.4.1).

The cation mixing results for SC-NMC811 showed that the Al-doped NMC811 phase had generally a larger amount of Ni in Li sites compared to the undoped phase, which is consistent with the idea that the aluminum is displacing some of the Ni atoms. However, no clear differences in the cation mixing was seen before and after annealing for the SC-NMC811 material.

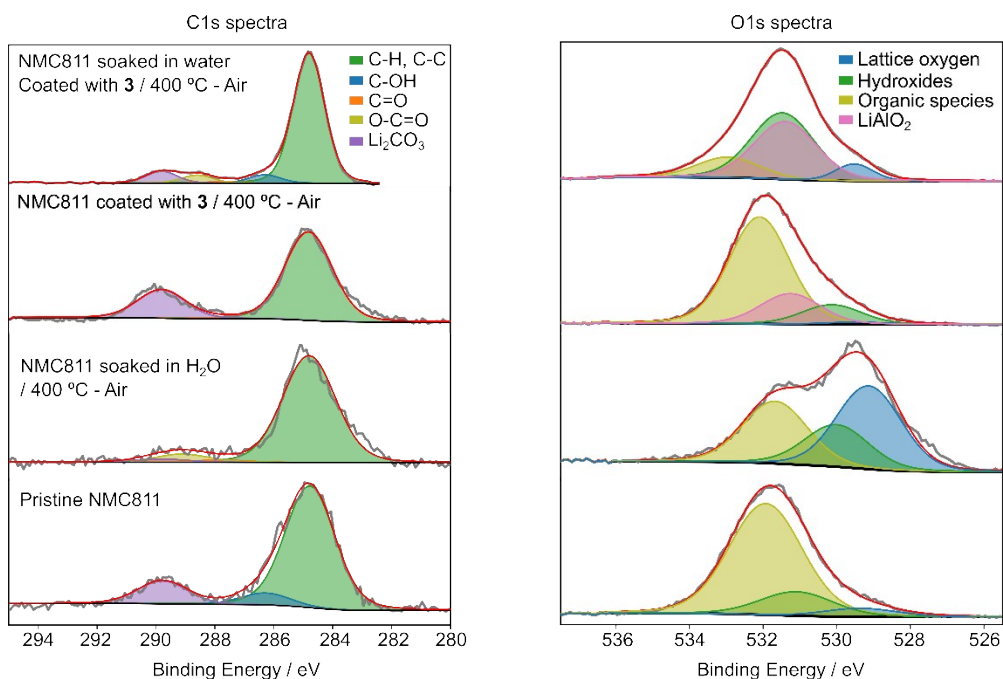
**Table S5.2.4.1.**  $R_{wp}$  and lattice parameters from Rietveld refinement of synchrotron data.

Sample	$R_{wp}$ / %	Refined phases	Wt %	a / Å	b / Å	c / Å
Polycrystalline (PC) NMC811	3.541	NMC811	100%	2.87	2.87	14.188
PC-NMC811 coated with <b>3</b> / 400 °C (air)	4.898	NMC811	100%	2.869	2.869	14.187
Single-crystal (SC)-NMC811	4.204	Al-NMC NMC	57% 43%	2.875 2.873	2.875 2.873	14.205 14.196
SC-NMC811 coated with <b>3</b> 100 °C (vacuum)	4.114	Al-NMC NMC	57% 43%	2.875 2.873	2.875 2.873	14.207 14.196
SC-NMC811 coated with <b>3</b> 400 °C (air)	3.118	Al-NMC NMC	62% 38%	2.876 2.872	2.876 2.872	14.206 14.192

## S5.3. X-ray Photoelectron Spectroscopy

### S5.3.1. O 1s and C 1s spectra

The O 1s and C 1s spectra of the pristine PC-NMC811 show the presence of organic species (mainly aliphatic) and  $\text{Li}_2\text{CO}_3$  on the surface (Figure S5.3.1.1). The lack of a lattice oxygen peak and the high intensity of the organic component of the O 1s spectrum suggest that the surface of the pristine PC-NMC811 is significantly covered by impurities. From the position of the main O 1s peak (531.9 eV) these can be assigned to  $\text{Li}_2\text{CO}_3$ ,<sup>3</sup> in agreement with the C 1s results which confirms its presence (289.78 eV).<sup>4</sup> Furthermore, the C 1s spectrum allows us to conclude that, although  $\text{Li}_2\text{CO}_3$  is present, the predominant surface impurities are aliphatic. Finally, the C 1s and O 1s spectra of the NMC811 coated using **3** are quite similar to those of the pristine NMC811 due to the low amounts of coating present suggesting that the surface chemistry of the material was not affected by the coating step.



**Figure S5.3.1.1.** C 1s and O 1s high-resolution XPS spectra and fittings measured for the pristine PC-NMC811, PC-NMC811 soaked in water and annealed at 400 °C, PC-NMC811 coated with **3**, and pre-soaked PC-NMC811 coated with **3**. All the coated samples undergo a post-annealing treatment at 400 °C for 4 h after solution deposition of the coating.

### S5.3.2. Peak parameters in the fittings of the C 1s, O 1s and Al 2p spectra

**Table S5.3.2.1.** Position and area of the peaks used in the fitting of the C 1s spectra.

Component	Pristine		Soaked annealed		Soaked coated		coated	
	B.E./eV	Area	B.E./eV	Area	B.E./eV	Area	B.E./eV	Area
C-C, C-H	284.78	1334.25	284.8	1175.5	284.81	21765.4	284.8	2566.22
C-OH , C-O-C	286.28	130.79	286.3	0	286.31	1400.37	286.3	0
C=O	287.78	0	287.8	116.17	287.81	264.93	287.8	19.39
O-C=O	289.08	0	289.1	12.51	288.61	1347.75	289.1	53.48
LiCO <sub>3</sub>	289.78	250.66	289.8	217.62	289.81	1983.69	289.8	602.99

**Table S5.3.2.2.** Position and area of the peaks used in the fitting of the O 1s spectra.

Component	Pristine		H <sub>2</sub> O soaked/ annealed		Coated with <b>3</b>		Soaked coated	
	B.E./eV	Area	B.E./eV	Area	B.E./eV	Area	B.E./eV	Area
Lattice oxygen	529.35	418.3	529.39	3091.16	529.3	287.67	529.52	7710.1
Hydroxides	531.09	1232.29	531.31	487.18	530.69	2774	531.45	50945.7
Organic species	531.93	5697.1	531.76	1970.38	532.12	6127.2	532.94	15757.6
LiAlO <sub>2</sub>	-	-	-	-	530.99	584.18	531.42	42971.2

**Table S5.3.2.3.** Position and area of the peaks used in the fitting of the Al 2p spectra.

Component	H <sub>2</sub> O soaked / coated with <b>3</b>		Coated with <b>3</b>		Pristine NMC811	
	B.E./eV	Area	B.E./eV	Area	B.E./eV	Area
Al 2p	74.28	2974.24	74.18	166.47	-	-
Mn 3p	50.63	435.7	50.63	56.6	50.63	85.13
Li 1s	55.4	1055.5	55.33	317.63	55.61	230.68
Co 3p	61.65	229.62	63.63	10.4	63.07	91.28
Co 3p sat	71.65	22.96	73.63	1.04	73.07	9.13
Ni 3p 3/2	67.63	2499.81	67.77	241.01	68.39	625.76
Ni 3p 1/2	69.83	1249.9	69.97	120.5	70.59	312.88
Ni 3p sat	73.63	774.94	73.77	74.71	74.39	193.98

### S5.3.3. Method for fitting the O 1s spectra

The O 1s spectra were fitted to three components for the cases of pristine PC-NMC811 and PC-NMC811 soaked in water and annealed at 400 °C: lattice oxygen, hydroxides, and organic species. By comparing the O 1s and C 1s spectra it can be deduced that the main contribution to the O 1s spectra is  $\text{Li}_2\text{CO}_3$ . An additional component was added for the coated samples to account for the presence of  $\text{LiAlO}_2$ . The organic species and  $\text{LiAlO}_2$  components were constrained to the same full-width half maxima (FWHM). Since the lattice oxygen component tends to be narrower,<sup>5</sup> this component was set at 0.6 times the FWHM of the other two components as this gave the best fit for NMC811 soaked in water. Finally, the hydroxide component was allowed to vary freely in FWHM between 1 and 2. The position of the components was also constrained to the regions in which these peaks are expected to appear (values for the position constraints can be found in Table S5.3.1).<sup>5</sup>

### S5.3.4. Peak constraints used to fit of the O 1s, C 1s and Al 2p spectra

**Table S5.3.4.1.** Constraints used to fit the O 1s spectra

Component	FWHM	Position / eV
Lattice Oxygen	= Hydroxides	529.6 – 528.5
Hydroxides	1 - 3	531.5 – 530
Organic Species	= Hydroxides	533.7 – 531.5
$\text{LiAlO}_2$	= Hydroxides	531.5 – 530

**Table S5.3.4.2.** Constraints used to fit the C 1s spectra

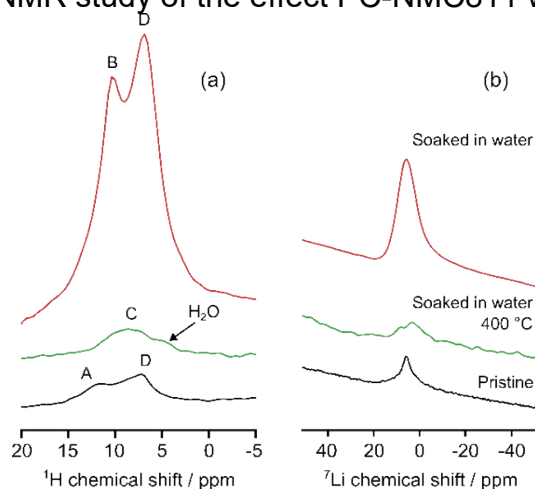
Component	FWHM	Position / eV
C-C, C-H	0.3 – 7	292 – 282
C-OH	= C-C, C-H	C-C, C-H + 1.5
C=O	= C-C, C-H	C-C, C-H + 3
O-C=O	= C-C, C-H	C-C, C-H + 3.8
$\text{Li}_2\text{CO}_3$	= C-C, C-H	C-C, C-H + 5

**Table S5.3.4.2.** Constraints used to fit the Al 2p spectra

<b>Component</b>	<b>FWHM</b>	<b>Position / eV</b>	<b>Area</b>
Al 2p	1 – 3	75.9 – 73.6	
Mn 3p	2 – 4	52.6 – 50.6	
Li 1s	2 – 4	56.6 – 54.6	
Co 3p	2 – 4	63.6-61.6	
Co 3p satellite	3 – 5	= Co 3p + 10	= Co 3p x 0.1
Ni 3p 3/2	2 – 4	68.6 – 67.6	
Ni 3p 1/2	= Ni 3p 3/2	Ni 3p 3/2 + 2.2	= Ni 3p 3/2 x 0.5
Ni 3p satellite	3 – 5	Ni 3p 3/2 + 6	= Ni 3p 3/2 x 0.31

## S5.4. Solid-state NMR spectra

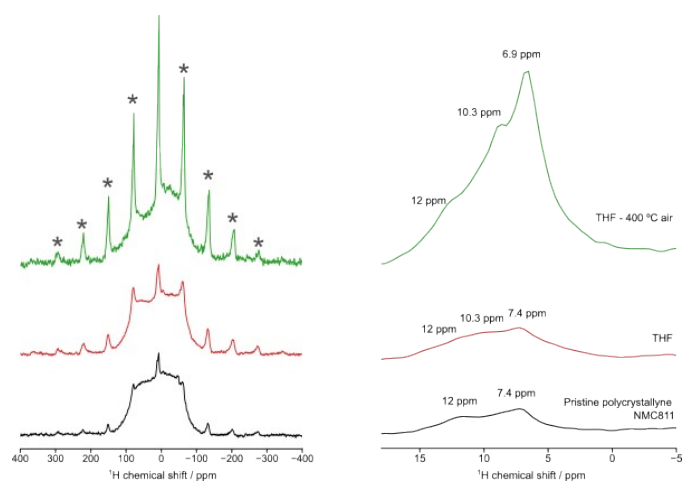
### S5.4.1. $^1\text{H}$ and $^7\text{Li}$ SSNMR study of the effect PC-NMC811 water treatment



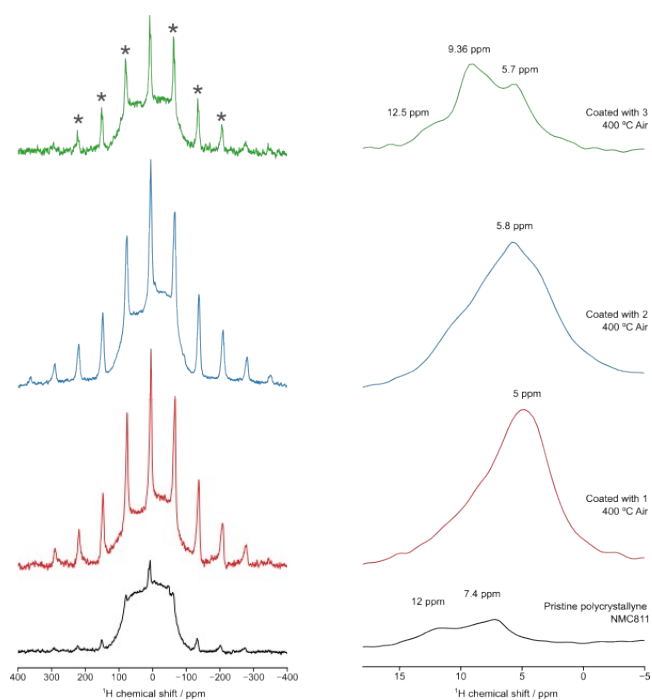
**Figure S5.4.1.1.** SSNMR spectra of pristine PC-NMC811 (black), PC-NMC811 soaked in water and dried at 50 °C under vacuum (red), and PC-NMC811 soaked in water, dried at 100 °C under vacuum and then annealed at 400 °C under air for 4 h (green). The  $^1\text{H}$  NMR (a) and  $^7\text{Li}$  (b) spectra of these samples are shown in the Figure. Four different surface proton environments are seen: (A) 11.9 ppm and (B) 10.5 ppm, assigned here to carbonic acids, and (C) 8.9 ppm and (D) 7.1 ppm which are assigned to diamagnetic TM(O)-OH species. Additionally, a water environment is present in the soaked samples at ca. 5 ppm. The  $^7\text{Li}$  spectra (b) show a peak at 6 ppm which is assigned to surface Li species. Water on oxide surfaces has a  $^1\text{H}$  chemical shift of ca. 5 ppm,<sup>6-8</sup> and it is present in both soaked samples (with and without an annealing step, Figure 5a). The peaks at higher shifts A and B are likely to be  $\text{LiHCO}_3$ , and the remaining peaks C and D are attributed to diamagnetic TM(O)-OH species where the TM are low-spin  $\text{Co}^{3+}$  and  $\text{Ni}^{4+}$ . The absence of  $\text{LiOH}$  was also confirmed by the lack of peaks at negative ppm (-1 to -1.5 ppm)<sup>9</sup>

#### S5.4.2. $^1\text{H}$ SSNMR Measurements

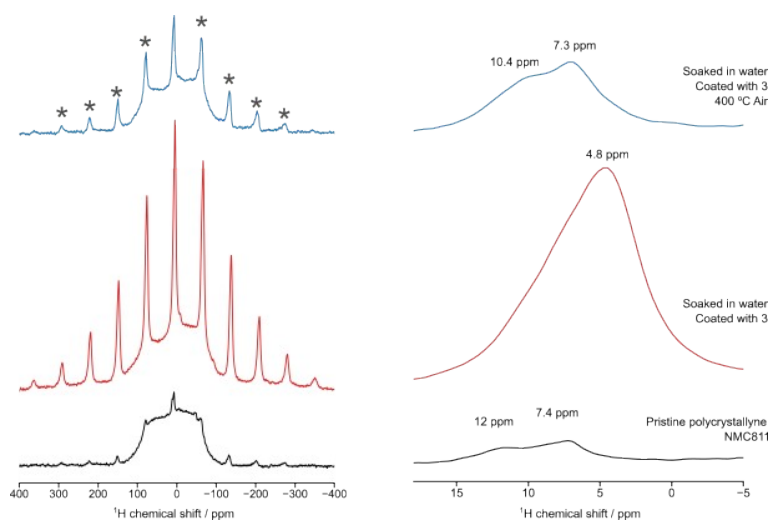
All the  $^1\text{H}$  NMR spectra showed a broad feature centered at 0 ppm and with a linewidth of approximately 200 ppm due to the probe background (Figures S5.4.1.2-S5.4.1.4). Narrower peaks are due to protic surface species.



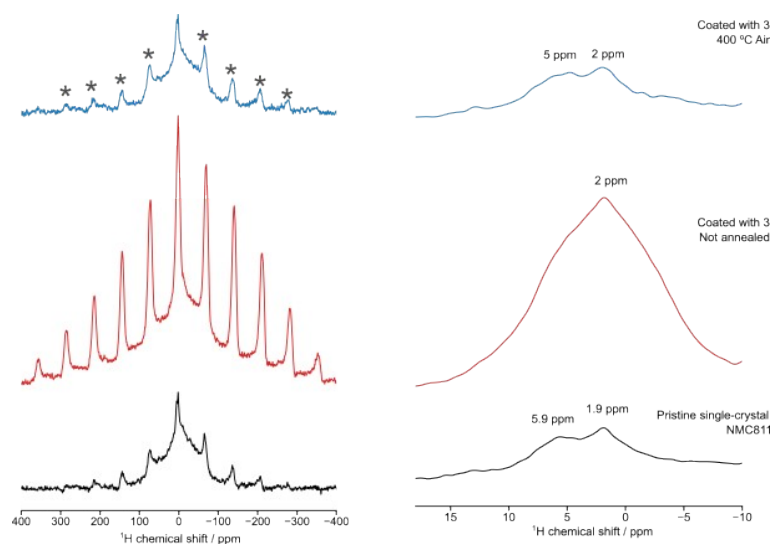
**Figure S5.4.1.2.**  $^1\text{H}$  SSNMR of the pristine PC-NMC811, PC-NMC811 soaked in THF and dried at  $100\text{ }^\circ\text{C}$  under vacuum for 2 h and THF soaked NMC811 annealed under air. Full spectra and zoom-in on the central transition are shown. The spinning sidebands are marked with stars.



**Figure S5.4.1.3.**  $^1\text{H}$  SSNMR of pristine PC-NMC811, PC-NMC811 coated with **1**, PC-NMC811 coated with **2** and PC-NMC811 coated with **3**. Full spectra and zoom-in on the central transition are shown. The spinning sidebands are marked with stars.



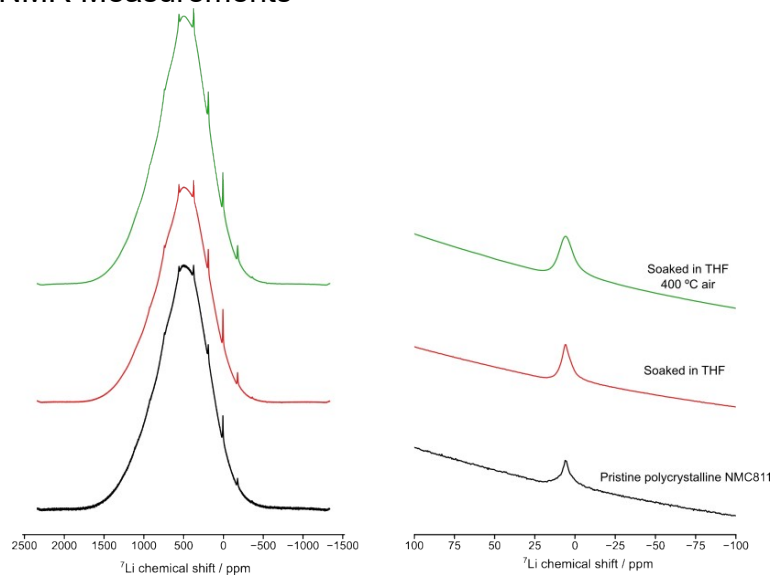
**Figure S5.4.1.4.**  $^1\text{H}$  SSNMR of pristine PC-NMC811, PC-NMC811 soaked in water and dried under vacuum at 50 °C for 2h, PC-NMC811 soaked in water, dried under vacuum at 50 °C for 2h and the annealed at 400 °C under air for 4h. Full spectra and zoom-in on the central transition are shown. The spinning sidebands are marked with stars.



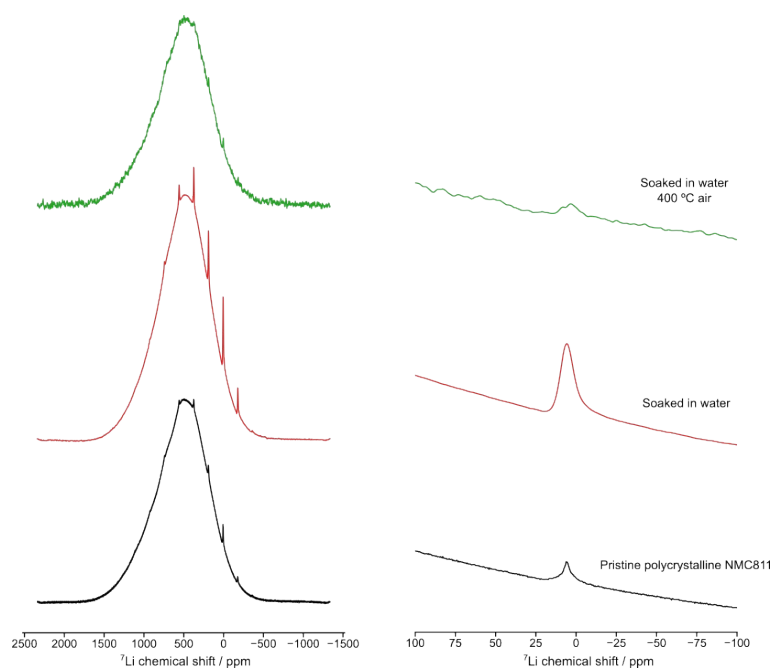
**Figure S5.4.1.5.**  $^1\text{H}$  SSNMR of pristine SC-NMC811, SC-NMC811 coated with **3** before annealing, SC-NMC811 coated with **3** and annealed at 400 °C under air for 4h. Full spectra and zoom-in on the central transition are shown. The spinning sidebands are marked with stars.



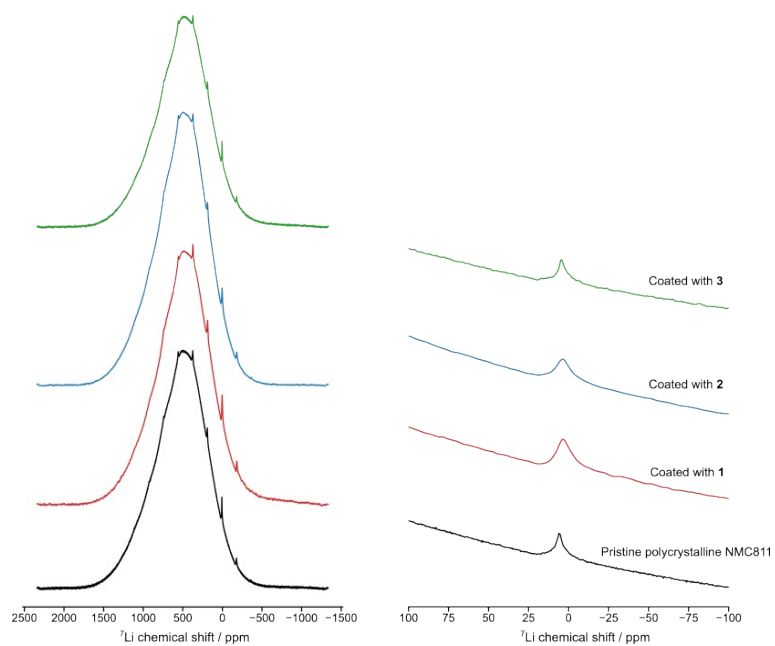
### S5.4.3. $^7\text{Li}$ SSNMR Measurements



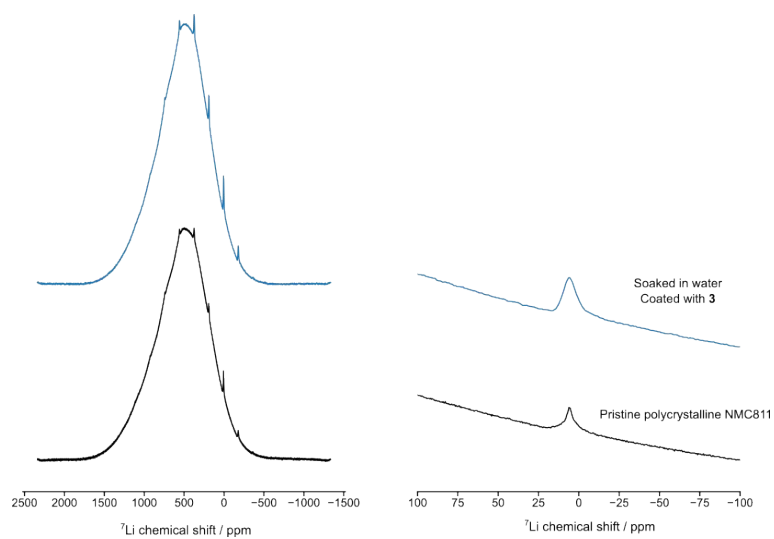
**Figure S5.4.2.1.**  $^7\text{Li}$  SSNMR of pristine PC-NMC811, PC-NMC811 soaked in THF, PC-NMC811 soaked in THF and then annealed at 400 °C for 4 h. Full spectra and zoom-in on the central transition are shown.



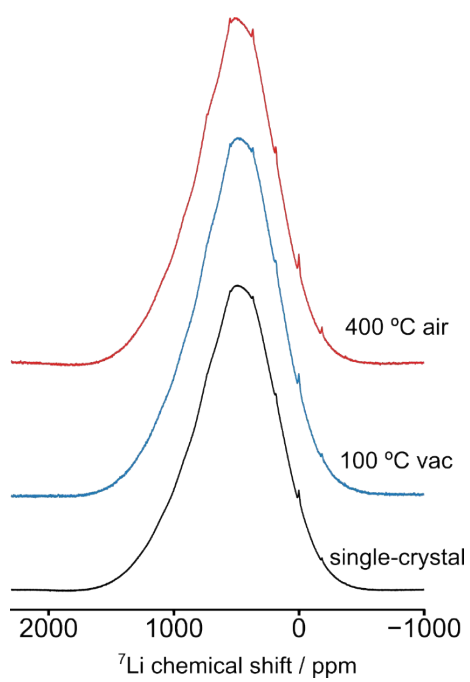
**Figure S5.4.2.2.**  $^7\text{Li}$  SSNMR of pristine PC-NMC811, PC-NMC811 soaked in water, PC-NMC811 soaked in water and then annealed at 400 °C for 4h. Full spectra and zoom-in on the central transition are shown.



**Figure S5.4.2.3.**  $^7\text{Li}$  SSNMR of pristine PC-NMC811, PC-NMC811 coated with **1**, PC-NMC811 coated with **2** and PC-NMC811 coated with **3**. Full spectra and zoom-in on the central transition are shown.

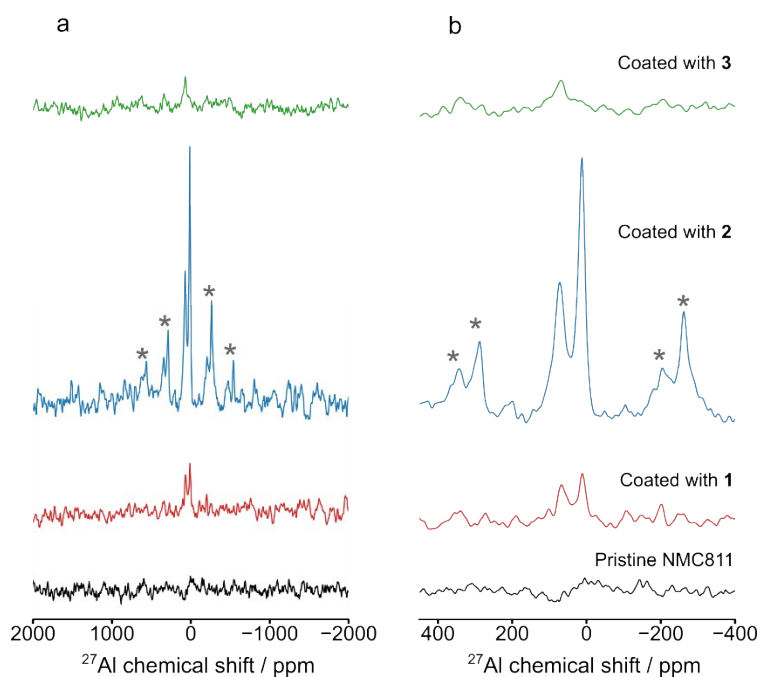


**Figure S5.4.2.4.**  $^7\text{Li}$  SSNMR of pristine PC-NMC811, PC-NMC811 soaked in water and coated with **3**.

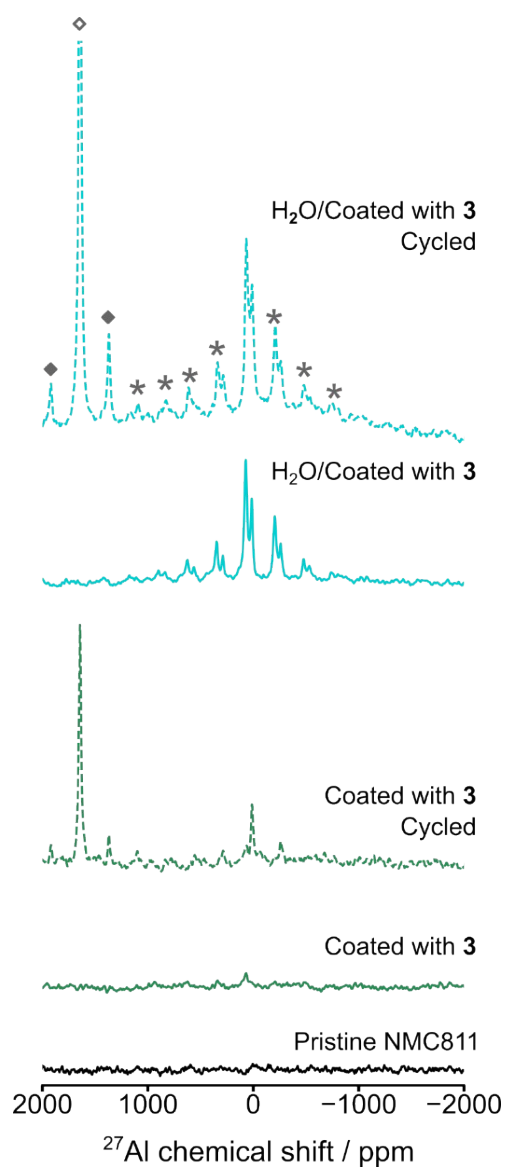


**Figure S5.4.2.5.**  ${}^7\text{Li}$  SSNMR of pristine SC-NMC811, SC-NMC811 coated with **3** and dried under vacuum at 100 °C and SC-NMC811 coated with **3** and annealed at 400 °C under air.

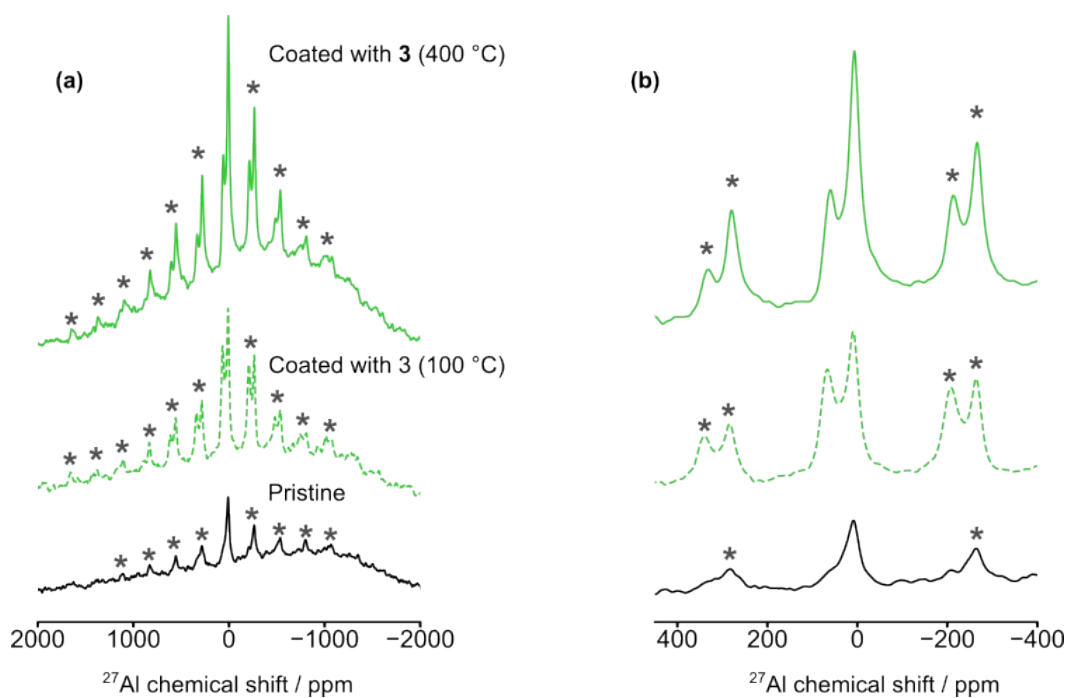
#### S5.4.4. $^{27}\text{Al}$ SSNMR Measurements



**Figure S5.4.3.1.**  $^{27}\text{Al}$  NMR of PC-NMC811 coated with precursors **1**, **2** and **3**. (a) Full view of the spectra (b) Zoom-in on the isotropic region. The spectra were recorded at a magnetic field strength of 16.4 T and a MAS frequency of 50 kHz and normalized to the sample mass and the number of scans. The spinning-sidebands are marked with asterisks.



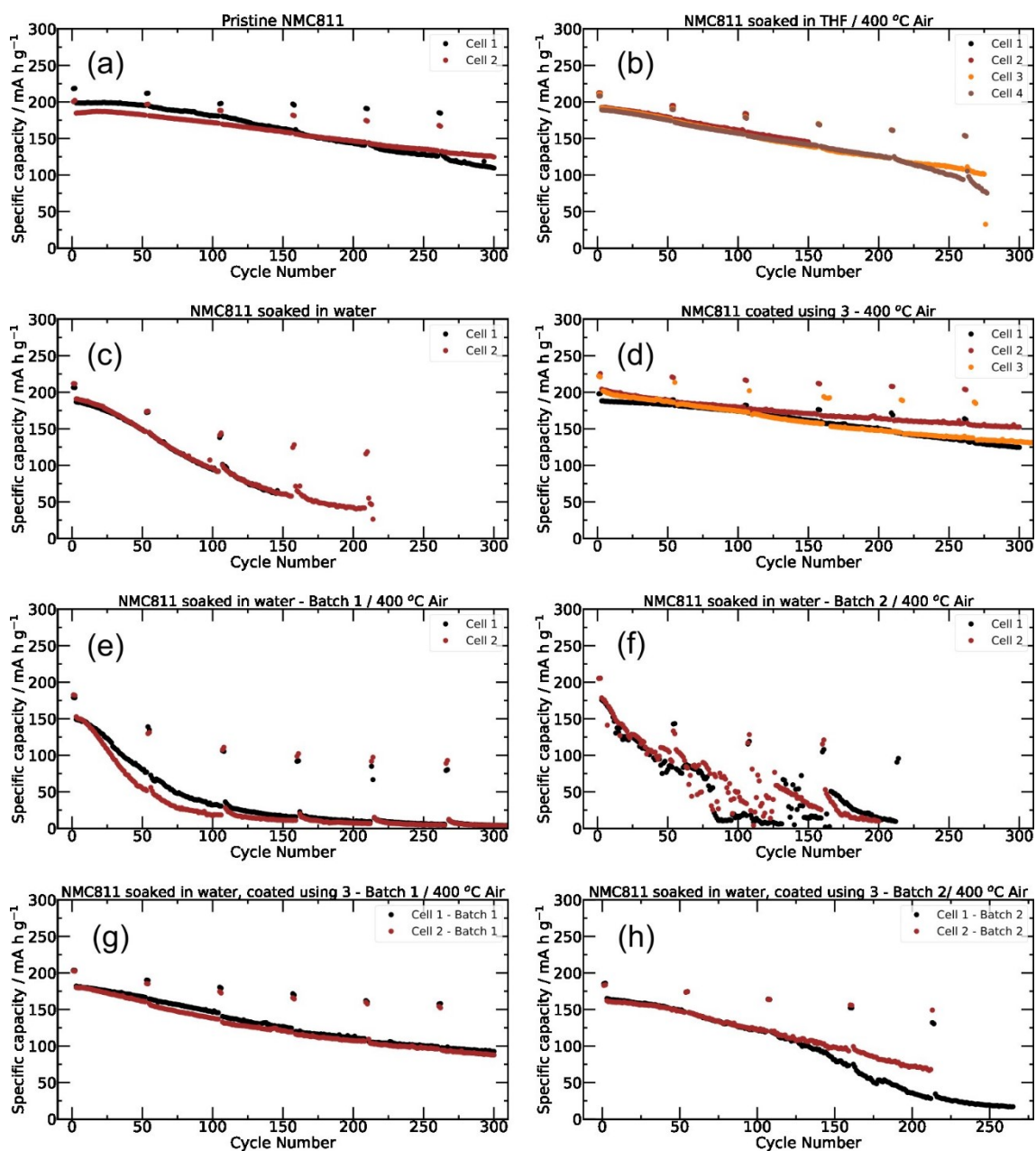
**Figure S5.4.3.2.**  $^{27}\text{Al}$  SSNMR of pristine PC-NMC811, PC-NMC811 coated with **3**, PC-NMC811 coated with **3** after 300 cycles, PC-NMC811 soaked in water and coated with **3**, PC-NMC811 soaked in water and coated with **3** after 300 cycles. Al peaks at higher chemical shift (1636 ppm) are due to aluminum metal that was scraped off from the current collector during sample preparation. The isotropic resonance and spinning sidebands of those are marked with empty and full diamonds respectively.



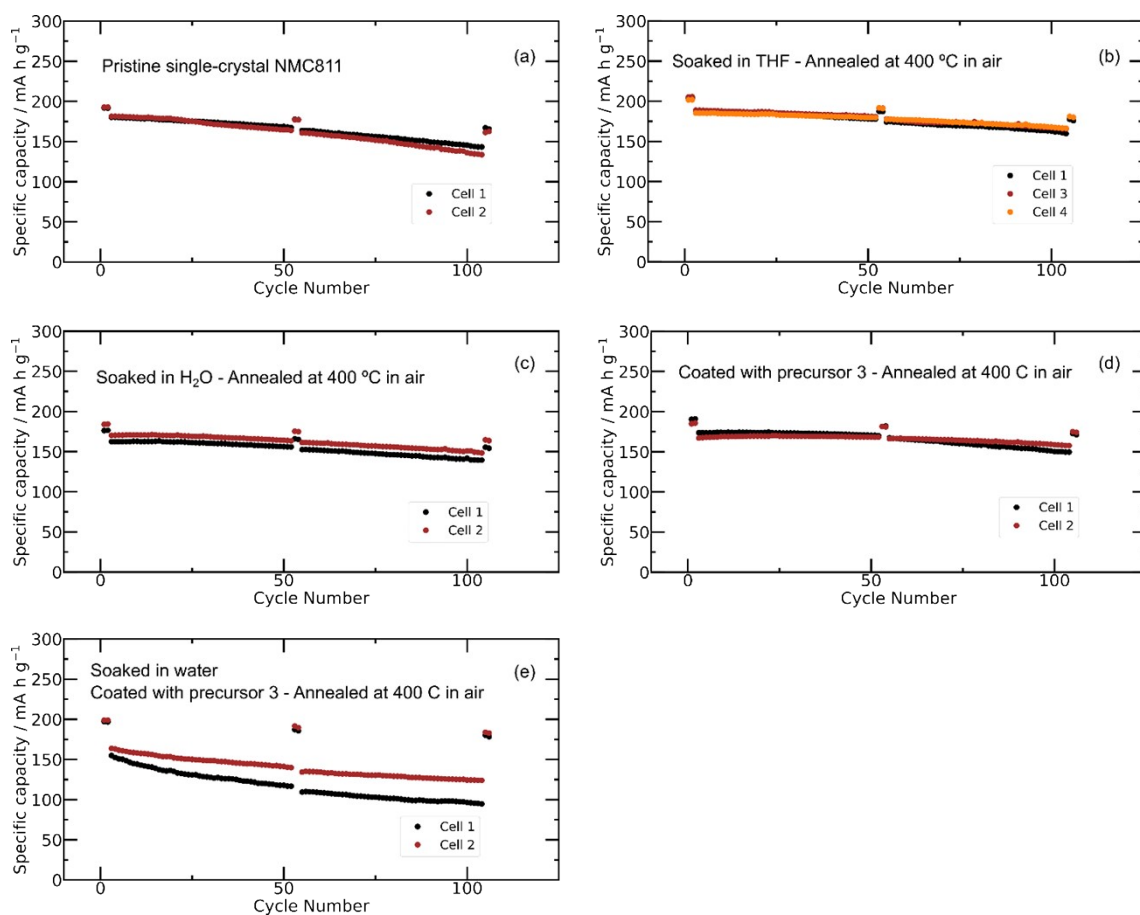
**Figure S5.4.3.3.**  $^{27}\text{Al}$  SSNMR of the SC-NMC811 materials. (a) Full spectra, (b) zoom-in on the isotropic region. The spectra were recorded at a magnetic field strength of 16.4 T and a MAS frequency of 50 kHz and normalized to the sample mass and the number of scans. The spinning sidebands were marked with stars.

## S5.5. Electrochemical cycling

### S5.5.1. Specific Capacity vs. Cycle Number Data



**Figure S5.5.1.** Full long-term cycling dataset for polycrystalline NMC811 materials. (a) Pristine NMC811. (b) NMC811 soaked in THF and annealed at 400 °C under air. (c) NMC811 soaked in water. (d) NMC811 coated using precursor 3 and annealed under air at 400 °C. (e) First batch of NMC811 soaked in water and annealed under air at 400 °C. (f) Second batch of NMC811 soaked in water and annealed under air at 400 °C. (g) First batch of NMC811 soaked in water and then coated with precursor 3 and annealed under air at 400 °C. (h) Second batch of NMC811 soaked in water and then coated with precursor 3 and annealed under air at 400 °C.

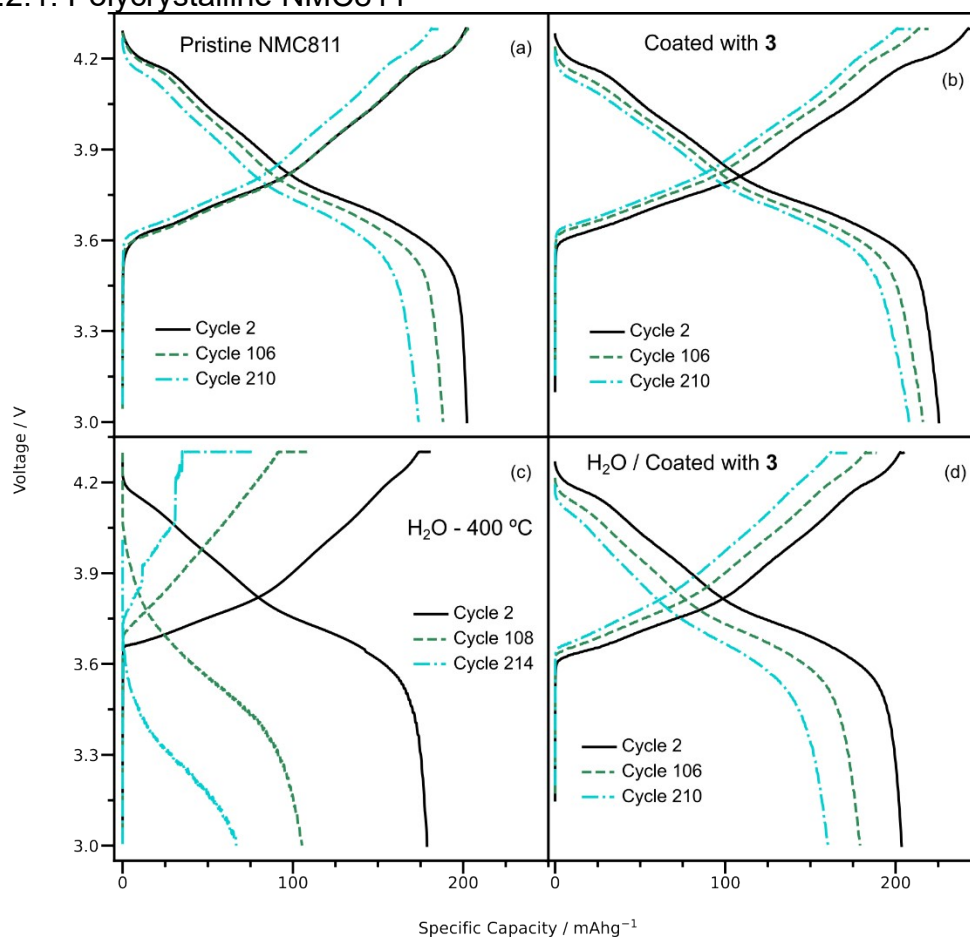


**Figure S5.5.2.** Full long-term cycling dataset for single-crystal NMC811 materials. (a) Pristine NMC811. (b) NMC811 soaked in THF and annealed at 400 °C under air. (c) NMC811 soaked in water. (d) NMC811 coated using precursor 3 and annealed under air at 400 °C. (e) NMC811 soaked in water and annealed under air at 400 °C.



## S5.5.2. Voltage Profiles

### S5.5.2.1. Polycrystalline NMC811

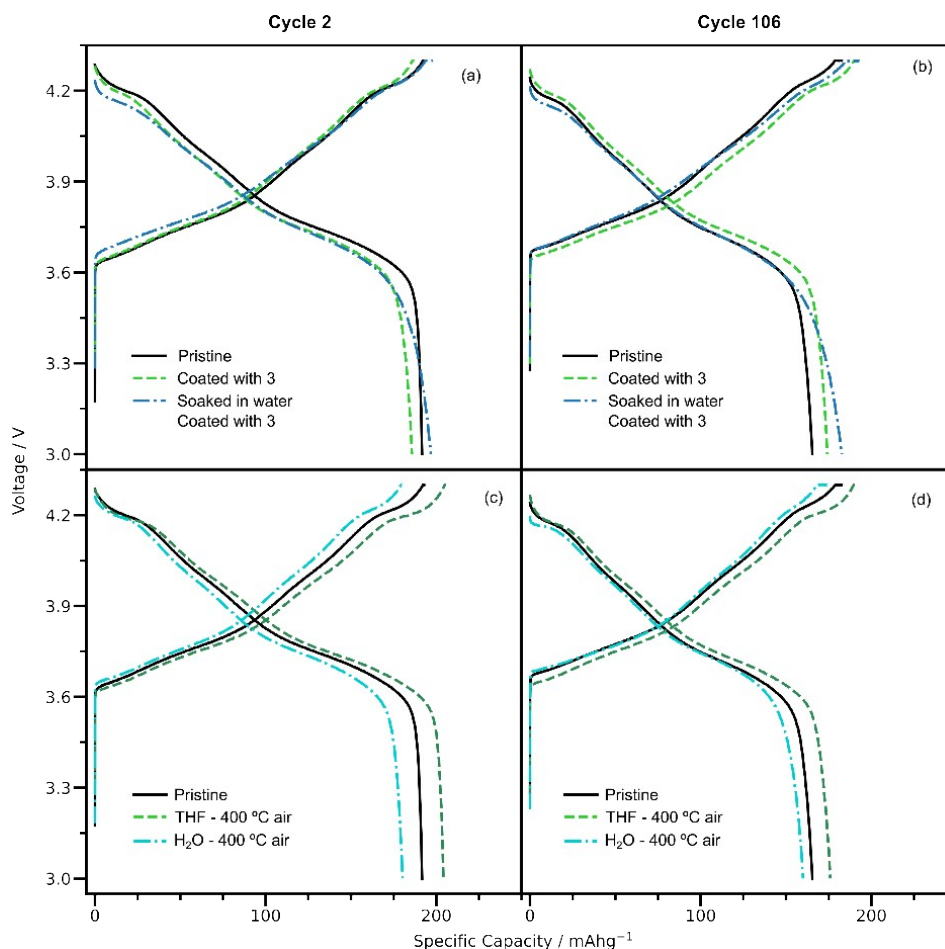


**Figure S5.5.3.** Voltage profiles measured for the slow cycles of PC-NMC811 samples. (a) Pristine PC-NMC811, (b) PC-NMC811 coated with **3**, (c) PC-NMC811 soaked in water and then annealed at 400 °C under air. (d) PC-NMC811 soaked in water, coated with **3** and annealed under air at 400 °C.

### S5.5.2.2. Single-crystal NMC811

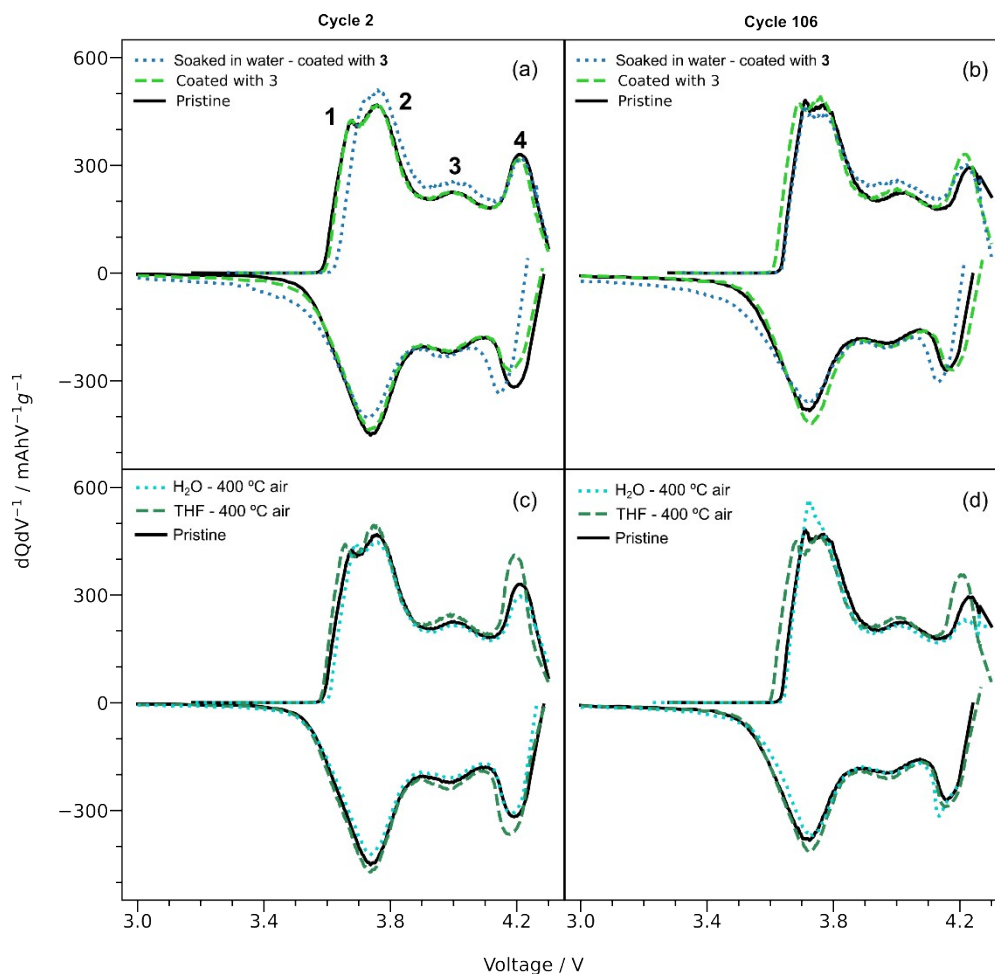
The voltage profiles obtained on the second and last (106<sup>th</sup>) C/20 cycles for the SC-NMC811 samples are analyzed to determine the processes responsible for the differences in capacity retention (Figure S5.5.4). Comparing the voltage profiles of cycle 2 (Figure S5.5.4a) reveals that both the coated and soaked/coated SC-NMC811 display larger overpotentials compared to pristine SC-NMC811. However, the pre-soaked and coated SC-NMC811 is able to deliver higher discharge capacities than both pristine and coated SC-NMC811 at C/20. The effect of soaking in water or THF and annealing under air on the C/20 voltage profiles of the SC-NMC811 can be seen in bottom panels of Figure S5.5.4 (c and d). The THF-soaked SC-NMC811 shows lower overpotential in cycle 2 whereas the water soaked shows a higher overpotential compared to the pristine SC-NMC811. As discussed previously, THF soaking removes some of the aluminum from the surface, which should hinder lithium transport. Meanwhile, the water treatment is expected to form some surface rock-salt. At cycle 106 the voltage profile of pristine SC-NMC811 and SC-NMC811 soaked in water are almost superimposable, due to the greater degradation rate of the pristine sample (Figure S5.5.4d). Meanwhile, THF treated SC-NMC811 degrades slightly but continues to show higher capacities and lower overpotentials.

The dQ/dV profiles of the pristine SC-NMC811 and the SC-NMC811 coated using **3** are very similar (Figure S5.5.5a, b) with the lower capacity of the coated material on the second cycle mainly coming from the high voltage process on discharge. In cycle 106 the symmetrical shape of the dQ/dV curve is mostly retained both for the pristine and the coated SC-NMC811 confirming the good cycling stability of both materials. The higher discharge capacity of the coated material at cycle 106 comes both from the high voltage charge process 4 and the low voltage discharge process 1. The SC-NMC811 soaked in water and coated with **3** shows greater polarization of its dQ/dV profile at cycle 2 and 106 compared to the other two, consistent with the findings of PC-NMC811 (Figure S5.5.5a and b). The effect of water and THF soaking on the dQ/dV of the SC-NMC811 is also considered (Figure S5.5.5c and d). In general, the dQ/dV of the soaked samples have similar shape and are symmetric, like that of pristine SC-NMC811, confirmed that neither soaking in water or THF has a negative effect on the cyclability of the SC-NMC811. The higher capacity of the sample soaked in THF comes from the high voltage process 4 which shows a more intense peak in the dQ/dV compared to the pristine SC-NMC811.



**Figure S5.5.4.** Voltage profiles obtained from the slow (C/20) cycles of the SC-NMC811 samples. (a) 2<sup>nd</sup> cycle of the pristine SC-NMC811, SC-NMC811 coated with **3** and SC-NMC811 soaked in water and coated with **3**. (b) 106<sup>th</sup> cycle of the pristine SC-NMC811, SC-NMC811 coated with **3** and SC-NMC811 soaked in water and coated with **3**. (c) 2<sup>nd</sup> cycle of pristine SC-NMC811, SC-NMC811 soaked in water and annealed at 400 °C under air, SC-NMC811 soaked in THF and annealed at 400 °C under air. (d) 106<sup>th</sup> cycle of pristine SC-NMC811, SC-NMC811 soaked in water and annealed at 400 °C under air, SC-NMC811 soaked in THF and annealed at 400 °C under air.

### S.5.5.3. dQ/dV Profiles for Single-crystal NMC811 Electrochemical Cycling data



**Figure S5.5.5.** dQ/dV profiles obtained from the slow ( $C/20$ ) cycles of the SC-NMC811 samples. (a) 2<sup>nd</sup> cycle of the pristine SC-NMC811, SC-NMC811 coated with **3** and SC-NMC811 soaked in water and coated with **3**. (b) 106<sup>th</sup> cycle of the pristine SC-NMC811, SC-NMC811 coated with **3** and SC-NMC811 soaked in water and coated with **3**. (c) 2<sup>nd</sup> cycle of pristine SC-NMC811, SC-NMC811 soaked in water and annealed at 400 °C under air, SC-NMC811 soaked in THF and annealed at 400 °C under air. (d) 106<sup>th</sup> cycle of pristine SC-NMC811, SC-NMC811 soaked in water and annealed at 400 °C under air, SC-NMC811 soaked in THF and annealed at 400 °C under air.

## S5.5.4. Comparison with Literature

**Table S5.5.4.1.** Summary of previous LiAlO<sub>2</sub> coatings work Ni-rich cathodes including coating method and coating phases obtained, cathode material used, electrochemical testing conditions and performance metrics. In all cases, testing was carried out using a half cell configuration (vs. lithium) and in coin cells except for reference 15 which used pouch cells. The active material loadings varied considerably between studies ( 2 – 8 mg cm<sup>-2</sup>) and different electrolyte blends were used (1 – 1.2 M LiPF<sub>6</sub> dissolved in EC/EMC or EC/EMC/DMC with or without VC). See Table S5.5.4.2 for details regarding active material loadings and electrolyte used.

Method	Phase	Cathode	$\delta V$	Rate	Discharge Cap. 1 <sup>st</sup> Cycle (mA h g <sup>-1</sup> )			Cap. Retention (%)			Ref.
					Coated	Pristine	#Cycle	Pristine	Coated	Difference	
Dry powder coating	$\alpha$ -LiAlO <sub>2</sub>	PC-NMC701515	3-4.3	C/2	180 C/10	170 C/10	100	72.5	92.5	20	10
Co-precipitation	$\alpha / \gamma$ LiAlO <sub>2</sub> or $\alpha$ -LiAlO <sub>2</sub>	PC-NMC622	3-4.4	C/5	200	190	100	86.4	96.4	10	11
Co-precipitation	-	PC-LiNi <sub>0.9</sub> Co <sub>0.1</sub> O <sub>2</sub>	2.7-4.3	1C	199	185	100	85.9	97.4	11.5	12
Polymer-assisted coating	-	PC-NMC811	2.8 - 4.5	C/2	200	180	100	76	90	14	13
Sol-gel	Al <sub>2</sub> O <sub>3</sub> / $\gamma$ -LiAlO <sub>2</sub>	PC-NMC811	3 - 4.5	C/3	190	194	50	64	61	-3	14
ALD	$\beta$ -LiAlO <sub>2</sub>	PC-NCA801505	2.7-4.3	C/2.5	184	188	100	78	94	16	15
Sol-gel	-	PC-NMC622	2.7-4.5	C/5	206.8	187.2	100	79.7	87	7.3	16
Hydrothermal	$\alpha$ -LiAlO <sub>2</sub> + $\beta$ -LiAlO <sub>2</sub>	PC-NMC523	2.7-4.6	1C	200	140	100	73.8	91	17.2	17
Sol-gel	$\gamma$ -LiAlO <sub>2</sub>	PC-NMC811	2.8-4.3	C/2	186.4	175	100	82	93	11	18
Single-source deposition	$a$ -Li <sub>1-x</sub> Al <sub>1+x/3</sub> O <sub>2</sub>	PC-NMC811	3 - 4.3	C/2	200	180	100	96	87	-9	This work
Single-source deposition	$a$ -Li <sub>1-x</sub> Al <sub>1+x/3</sub> O <sub>2</sub>	SC-NMC811	3 - 4.3	C/2	165	180	100	78	95	17	

**Table S5.5.4.2.** Cathode active mass loadings and electrolyte blends used in relevant LiAlO<sub>2</sub> coating work compared in Table S5.5.4.1.

<b>Loading (mg cm<sup>-2</sup>)</b>	<b>Electrolyte</b>	<b>Reference</b>
7 - 8	1 M LiPF <sub>6</sub> 50:50 (w/w) EC : EMC	<b>10</b>
4.3	1M LiPF <sub>6</sub> 30:70 EC : EMC 5 v/v VC	<b>11</b>
-	1.2 M LiPF <sub>6</sub> 30:70 (v/v) EC:EMC 2 wt. % VC	<b>12</b>
-	1M LiPF <sub>6</sub> , 1:1:1 (v/v/v) EC:DMC:EMC	<b>13</b>
-	1.2 M LiPF <sub>6</sub> , EC:EMC 3:7 v/v	<b>14</b>
5	1 M LiPF <sub>6</sub> , EC:EMC 3:7 w/w	<b>15</b>
-	1 M LiPF <sub>6</sub> , 1:1:1 (v/v/v) DMC:EMC:EC	<b>16</b>
1.95 - 2	1 M LiPF <sub>6</sub> 1:1:1 (v/v/v) DMC:EMC:EC	<b>17</b>
-	1M LiPF <sub>6</sub> 1:1:1 (v/v/v) EC:DMC:EMC	<b>18</b>
6 - 7	1 M LiPF <sub>6</sub> 3:7 (w/w) EC:EMC	This work
6 - 7	1 M LiPF <sub>6</sub> 3:7 (w/w) EC:EMC	

## References

- (1) Dahéron, L.; Dedryvère, R.; Martinez, H.; Ménétrier, M.; Denage, C.; Delmas, C.; Gonbeau, D. Electron Transfer Mechanisms upon Lithium Deintercalation from  $\text{LiCoO}_2$  to  $\text{CoO}_2$  Investigated by XPS. *Chem. Mater.* **2008**, *20* (2), 583–590. <https://doi.org/10.1021/cm702546s>.
- (2) Andreu, N.; Flahaut, D.; Dedryvère, R.; Minvielle, M.; Martinez, H.; Gonbeau, D. XPS Investigation of Surface Reactivity of Electrode Materials: Effect of the Transition Metal. *ACS Appl. Mater. Interfaces* **2015**, *7* (12), 6629–6636. <https://doi.org/10.1021/am5089764>.
- (3) Dedryvère, R.; Gireaud, L.; Grugeon, S.; Laruelle, S.; Tarascon, J.-M.; Gonbeau, D. Characterization of Lithium Alkyl Carbonates by X-Ray Photoelectron Spectroscopy: Experimental and Theoretical Study. *J. Phys. Chem. B*, **2005**, *109* (33), 15868–15875. <https://doi.org/10.1021/jp051626k>.
- (4) Appapillai, A. T.; Mansour, A. N.; Cho, J.; Shao-Horn, Y. Microstructure of  $\text{LiCoO}_2$  with and without “ $\text{AlPO}_4$ ” Nanoparticle Coating: Combined STEM and XPS Studies. *Chem. Mater.* **2007**, *19* (23), 5748–5757. <https://doi.org/10.1021/cm0715390>.
- (5) Biesinger, M. C.; Payne, B. P.; Grosvenor, A. P.; Lau, L. W. M.; Gerson, A. R.; Smart, R. St. C. Resolving Surface Chemical States in XPS Analysis of First Row Transition Metals, Oxides and Hydroxides: Cr, Mn, Fe, Co and Ni. *Appl. Surf. Sci.* **2011**, *257* (7), 2717–2730. <https://doi.org/10.1016/j.apsusc.2010.10.051>.
- (6) Shen, L.; Wang, Y.; Du, J.; Chen, K.; Lin, Z.; Wen, Y.; Hung, I.; Gan, Z.; Peng, L. Probing Interactions of  $\gamma$ -Alumina with Water via Multinuclear Solid-State NMR Spectroscopy. *ChemCatChem* **2020**, *12* (6), 1569–1574. <https://doi.org/10.1002/cctc.201901838>.
- (7) Liu, F.; Feng, N.; Wang, Q.; Xu, J.; Qi, G.; Wang, C.; Deng, F. Transfer Channel of Photoinduced Holes on a  $\text{TiO}_2$  Surface As Revealed by Solid-State Nuclear Magnetic Resonance and Electron Spin Resonance Spectroscopy. *J. Am. Chem. Soc.* **2017**, *139* (29), 10020–10028. <https://doi.org/10.1021/jacs.7b04877>.
- (8) Mogilevsky, G.; Karwacki, C. J.; Peterson, G. W.; Wagner, G. W. Surface Hydroxyl Concentration on  $\text{Zr}(\text{OH})_4$  Quantified by  $^1\text{H}$  MAS NMR. *Chem. Phys. Lett.* **2011**, *511* (4–6), 384–388. <https://doi.org/10.1016/j.cplett.2011.06.072>.
- (9) Leskes, M.; Moore, A. J.; Goward, G. R.; Grey, C. P. Monitoring the Electrochemical Processes in the Lithium–Air Battery by Solid State NMR Spectroscopy. *J. Phys. Chem. C* **2013**, *117* (51), 26929–26939. <https://doi.org/10.1021/jp410429k>.
- (10) Herzog, M. J.; Gauquelin, N.; Esken, D.; Verbeeck, J.; Janek, J. Increased Performance Improvement of Lithium-Ion Batteries by Dry Powder Coating of High-Nickel NMC with Nanostructured Fumed Ternary Lithium Metal Oxides. *ACS Appl. Energy Mater.* **2021**, *9*, 8832–8848. <https://doi.org/10.1021/acsaem.1c00939>.
- (11) Touag, O.; Coquil, G.; Charbonneau, M.; Foran, G.; Ghosh, A.; Mankovsky, D.; Dollé, M. One-Pot Synthesis of  $\text{LiAlO}_2$ -Coated  $\text{LiNi}_{0.6}\text{Mn}_{0.2}\text{Co}_{0.2}\text{O}_2$  Cathode Material. *Energy Adv.* **2023**, *2* (5), 701–711. <https://doi.org/10.1039/D3YA00061C>.
- (12) Yu, H.; Cao, Y.; Chen, L.; Hu, Y.; Duan, X.; Dai, S.; Li, C.; Jiang, H. Surface Enrichment and Diffusion Enabling Gradient-Doping and Coating of Ni-Rich Cathode toward Li-Ion Batteries. *Nat. Commun.* **2021**, *12* (1), 4564. <https://doi.org/10.1038/s41467-021-24893-0>.
- (13) Ding, G.; Yan, F.; Zhu, Z.; Chen, J.; Hu, Z.; Li, G.; Liu, J.; Gao, L.; Jiang, W.; Sun, F. Mussel-Inspired Polydopamine-Assisted Uniform Coating of  $\text{Li}^+$  Conductive  $\text{LiAlO}_2$  on Nickel-Rich  $\text{LiNi}_{0.8}\text{Co}_{0.1}\text{Mn}_{0.1}\text{O}_2$  for High-Performance Li-Ion Batteries. *Ceram. Int.* **2022**, *48* (4), 5714–5723. <https://doi.org/10.1016/j.ceramint.2021.11.118>.
- (14) Han, B.; Key, B.; Lapidus, S. H.; Garcia, J. C.; Iddir, H.; Vaughey, J. T.; Dogan, F. From Coating to Dopant: How the Transition Metal Composition Affects Alumina

- Coatings on Ni-Rich Cathodes. *ACS Appl. Mater. Interfaces* **2017**, 9 (47), 41291–41302. <https://doi.org/10.1021/acsami.7b13597>.
- (15) Srur-Lavi, O.; Miikkulainen, V.; Markovsky, B.; Grinblat, J.; Talianker, M.; Fleger, Y.; Cohen-Taguri, G.; Mor, A.; Tal-Yosef, Y.; Aurbach, D. Studies of the Electrochemical Behavior of  $\text{LiNi}_{0.80}\text{Co}_{0.15}\text{Al}_{0.05}\text{O}_2$  Electrodes Coated with  $\text{LiAlO}_2$ . *J. Electrochem. Soc.* **2017**, 164 (13), A3266–A3275. <https://doi.org/10.1149/2.1631713jes>.
- (16) Liu, W.; Li, X.; Xiong, D.; Hao, Y.; Li, J.; Kou, H.; Yan, B.; Li, D.; Lu, S.; Koo, A.; Adair, K.; Sun, X. Significantly Improving Cycling Performance of Cathodes in Lithium Ion Batteries: The Effect of  $\text{Al}_2\text{O}_3$  and  $\text{LiAlO}_2$  Coatings on  $\text{LiNi}_{0.6}\text{Co}_{0.2}\text{Mn}_{0.2}\text{O}_2$ . *Nano Energy* **2018**, 44, 111–120. <https://doi.org/10.1016/j.nanoen.2017.11.010>.
- (17) Li, L.; Chen, Z.; Zhang, Q.; Xu, M.; Zhou, X.; Zhu, H.; Zhang, K. A Hydrolysis-Hydrothermal Route for the Synthesis of Ultrathin  $\text{LiAlO}_2$ -Inlaid  $\text{LiNi}_{0.5}\text{Co}_{0.2}\text{Mn}_{0.3}\text{O}_2$  as a High-Performance Cathode Material for Lithium Ion Batteries. *J. Mater. Chem. A* **2015**, 3 (2), 894–904. <https://doi.org/10.1039/C4TA05902F>.
- (18) Tang, W.; Chen, Z.; Xiong, F.; Chen, F.; Huang, C.; Gao, Q.; Wang, T.; Yang, Z.; Zhang, W. An Effective Etching-Induced Coating Strategy to Shield  $\text{LiNi}_{0.8}\text{Co}_{0.1}\text{Mn}_{0.1}\text{O}_2$  Electrode Materials by  $\text{LiAlO}_2$ . *J. Power Sources* **2019**, 412, 246–254. <https://doi.org/10.1016/j.jpowsour.2018.11.062>.