## SUPPORTING INFORMATION

# It is Supporting Information data for the publication 'An insight into separating H<sub>2</sub> from natural gas/H<sub>2</sub> mixtures using Mg-based systems'

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The SI was created to not only present but also discuss some of the supplementary research data. The data and descriptions were divided into ten subchapters.

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## S1. Transition of energy economy in Germany



Fig. S1 Development of the hydrogen core network in Germany. Picture adapted from [S1]



\* The scope of retrofitting refers to the retrofitting of all components necessary for operation, but the cost reference is the gas turbine as core component

A pre-planned modular design of the ancillary systems can significantly reduce retrofitting costs for new plants (H<sub>2</sub> capability)
In many cases, retrofitting existing systems can make much more economic sense, even if measures to extend the service life, etc. still have to be carried out on the GTP

\*\*\*\* Additional costs for H<sub>2</sub> capability

Fig. S2. Development levels of H<sub>2</sub> readiness for gas turbine plants [S2].

For many years, applications such as gas turbines or burners in the glass industry have operated on a foundation of relatively stable gas quality. However, because of changing heating values and other specifications (Wobbe index or flame velocity) of primary H<sub>2</sub> and CH<sub>4</sub> in the gas mixture, the above-mentioned applications must be adjusted to the new conditions. For instance, gas turbine plants are planned to achieve Level 1 of readiness for H<sub>2</sub>-contributed gas mixture in 2025 (Fig. S2) [S2].

Level 1 means that the thermal firing capacity (TFC) equals 6%, corresponding to a volume fraction of H<sub>2</sub> in the gas mixture of approximately 16%. Level 2 with 25% of TFC means approx. 52% of H<sub>2</sub> in the gas mixture in 2028. Level 3 of TFC, equal to 100%, will be achieved in 2031 and stands for pure H<sub>2</sub>. Therefore, regarding the mentioned example of gas turbines, which will work with different fuel blends with H<sub>2</sub>, until the hydrogen economy is completely implemented, there will be a time when intermediate technologies like Hydrogen separation techniques must be applied to fulfil the economy transition goals.

### S2. Materials and gases



Fig. S3 SE-SEM images obtained for Mg, Ni and Fe powders. Scale bars: 100  $\mu m.$ 



**Fig. S4** XRD patterns obtained for pure elements (Mg, Ni, and Fe) and Mg-Ni, Mg-Fe systems after one hour of shaking.

**Table S1** Composition of natural gas (Air Liquide, Natural gas L) provided by the supplier and asanalyzed by GC (using 30m, RT-Q-Bond 0.53/20df G/785 column and 0.5 bar He)

	CH₄ / %	C <sub>2</sub> H <sub>6</sub> / %	N <sub>2</sub> / %	CO <sub>2</sub> / %	C <sub>3</sub> H <sub>8</sub> / %	C <sub>4</sub> H <sub>10</sub> / %
Air Liquide specification	83 – 99	<12	0.5 - 3	<1.5	0.1 - 1	0.1 - 1
GC	92.50	-	7.47	0.03	-	-



**Fig. S5** Results of mass spectroscopy performed on natural gas (Air Liquide, Natural gas L). Initially, the apparatus was fed with Ar flowing through the system (to determine the background signal level). The vertical line points to the moment when Ar was exchanged for natural gas.

The composition of the natural gas was evaluated by gas chromatography (GC) and mass spectroscopy (MS; in this case, only qualitatively) and compared with the supplier specification (Fig. S5, Table S1). The GC results indicate that the CH<sub>4</sub> and N<sub>2</sub> concentrations exceed 92 % and 7 %, respectively. The gas composition included traces of C<sub>2</sub>H<sub>6</sub>, CO<sub>2</sub>, and H<sub>2</sub>S. The experimental analysis did not reveal the presence of C<sub>3</sub>H<sub>8</sub> or C<sub>4</sub>H<sub>10</sub> (n) – which were stated by the supplier. The GC and MS data showed that the actual composition of natural gas differs slightly from the supplier specification.

### S3. Calibration of gas chromatography



**Fig. S6** Results of GC made on  $CH_4/H_2$  gas mixtures with different concentrations of  $CH_4$  and  $H_2$ : (a) relatively low  $H_2$  concentrations; (b,c) relatively high  $H_2$  concentrations. The dark blue area in (b) points out the region of patterns enlarged in (c). The percentage concentrations of each gas component are shown in the brackets. The dotted lines indicate the position of  $H_2$  and  $CH_4$  peaks.



**Fig. S7** Calibration curves for GC analyses of the  $CH_4$  content in the  $CH_4/H_2$  gas mixtures: (a) low  $CH_4$  concentrations; (b) high  $CH_4$  concentrations. Int in the y-axis description stands for Integral.

The calibration curves (Fig. S7) are based on the GC results (Fig. S6) obtained for the standard gas mixture concentrations (prepared on-site using the calibrated EL-FLOW Bronkhorst mass flow meters). The curve of the relatively low CH<sub>4</sub> concentration in the CH<sub>4</sub>/H<sub>2</sub> gas mixture (Fig. S7a) enables the quantitative determination of the H<sub>2</sub> concentration/purity in the gas mixture desorbed from the metal hydride. The curve of the relatively high CH<sub>4</sub> concentration in the CH<sub>4</sub>/H<sub>2</sub> gas mixtures (Fig. S7b) was used to evaluate the prepared on-site gas mixtures (5L bottles of gas mixtures).

#### S4. The gas-separation setup



#### Fig. S8 Scheme of the gas-separation setup

The setup scheme for conducting experiments on the separation of hydrogen from gas mixtures is shown in Fig. S8. The main part of the setup is the autoclave, which is located in the central position of the sketch. In the autoclave, the gas separation reaction has taken place, thus both hydrogenation of Mg and subsequent dehydrogenation of the MgH<sub>2</sub> (and other minor phases). The autoclave was manufactured on-site as a batch reactor with an inner volume of 100 mL. The material used to fabricate the autoclave was austenitic steel according to the DIN EN 10881-1 standard, numbered X6NiCrTiMoVB25-15-2 (other denominations: W.-Nr. 1.4980, Alloy A-286). The autoclave could be opened/closed from one side via a flange/lid system with six screws and was sealed with a gold O-ring gasket. The autoclave filled with Mgbased system was half-inserted into an electrically heated furnace during the experiments. The upper part of the autoclave was only thermally insulated. In this way, the heater actively influenced the lower part of the autoclave containing the active material, while the upper part of the autoclave was at a lower temperature (especially in the area of the flange/lid system), which prevented leaks during the operation of the autoclave. The pressure inside the autoclave was measured using a JUMO dTrans p30 pressure transmitter. The temperature of the active material was controlled using a K-type thermocouple.

The calculation of the gravimetric hydrogen storage capacity (of formed hydride) during the H<sub>2</sub> separation was accomplished by measuring both the pressure increase and the temperature of H<sub>2</sub> in the reservoir (which was connected to the autoclave) during the dehydrogenation process. The reservoir's internal volume was 500 mL. It had a heating system, which maintained the gas temperature in the reservoir at a constant 30°C. The pressure transducer for measuring the gas pressure in the reservoir was type PAA- 33X provided by KELLER Gesellschaft für Druckmesstechnik GmbH, and a type K thermocouple controlled the temperature. Pressure and temperature values in the autoclave and reservoir were recorded by the JUMO 2000. Appropriate temperature controller and temperature limiter systems supervised both heaters.

As shown in Fig. S8, the main parts of the setup (autoclave and reservoir) were connected via valves and pressure reducers to three gas cylinders containing H<sub>2</sub>, Ar, and a gas mixture.

H<sub>2</sub> was used to activate the Mg-based systems placed inside the autoclave, as well as for hydrogenation tests with pure H<sub>2</sub>. The activation protocol has already been published elsewhere [S3]. Ar was used to purge the sample from the residual gas after separation of H<sub>2</sub> - from the remaining gas mixture present in the autoclave after subsequent cooling. In addition to the Ar purge, a short evacuation (under dynamic vacuum) of the hydrogenated Mg-based systems was also applied to get rid of any gases that were still in the reactor. The abovementioned purging/evacuation allowed us to investigate the quality of released H<sub>2</sub> in in the subsequent dehydrogenation process (using GC). After the dehydrogenation process was completed, the desorbed gas was expanded through a valve connected to the GC gas sampling bag (shown on the right side in Fig S8). The vent system was used to release the overpressure after the hydrogenation and dehydrogenation process as well as during the flushing of the system with Ar.

S5. Hydrogen absorption and desorption in Mg-Ni system



**Fig. S9** Change of the partial  $H_2$  pressure in time (representing  $H_2$  absorption) in autoclave loaded with Mg-Ni system during hydrogenation with 100 bar of CH<sub>4</sub>/H<sub>2</sub> gas mixture (20 bar of partial H<sub>2</sub> pressure) at 350 °C.



**Fig. S10** Amount of H<sub>2</sub> desorbed from hydrogenated Mg-Ni system in the following steps of dehydrogenation at ~195 °C and ~ 215 °C. Each dehydrogenation step lasted five minutes. Prior to first step and between each subsequent one, the autoclave filled with hydrogenated Mg-Ni system was evacuated under vacuum for 10 seconds.

The hydrogenated Mg-Ni system was used to estimate the amount of  $H_2$  desorbed from the material at ~195 °C and ~ 215 °C. The material after hydrogenation was evacuated under vacuum for 10 seconds and then left for five minutes to determine amount of desorbed  $H_2$ . The amount of desorbed hydrogen is presented in Fig. S10 (step 1 at each temperature). The process of evacuation (10 seconds) and dehydrogenation (five minutes) was repeated stepwise using the same material.

Fig. S10 shows that Mg-Ni system desorbs more than 1 wt.%  $H_2$  within the first dehydrogenation step at ~ 215 °C. The following dehydrogenation steps performed at ~ 215 °C

show reduction of desorbed  $H_2$  in the following dehydrogenation steps, to reach no dehydrogenation in the fifth step. The amount of dehydrogenated  $H_2$  is significantly lower when the process is performed at ~195 °C. Not more than 0.4 wt.%  $H_2$  is desorbed in the first step at this temperature. Moreover, the material does not desorb any hydrogen already in the third step at ~195 °C.

The data presented in Fig. S10 are particularly useful when one considers that both systems (Mg-Ni and Mg-Fe), were vented, and the used setup was five times purged using an alternately dynamic vacuum and 5 bar Ar, between each hydrogenation and dehydrogenation (in  $H_2$  separation experiments presented in Fig. 1 a,b). This venting and purging process corresponds to the part of the  $H_2$  desorbed in the first step of dehydrogenation presented in Fig. S10.



**Fig. S11**  $H_2$  desorption kinetic curve recorded for the hydrogenated Mg-Ni system during its heating from ~215 °C to ~350 °C. The dashed line indicates the start of dynamic  $H_2$  desorption.



## S6. Mg-Ni system after separation of $H_2$ from $CH_4/H_2$ and $NG/H_2$ gas mixtures

**Fig. S12** Pictures of the flakes and powders of the Mg-Ni system material after cycles of hydrogenation/dehydrogenation experiments under 20 bar H<sub>2</sub>, 100 bar of CH<sub>4</sub>/H<sub>2</sub> gas mixture (20 bar of partial H<sub>2</sub> pressure), and 100 bar of natural gas/H<sub>2</sub> gas mixture (20 bar of partial H<sub>2</sub> pressure). The sample/layer numbers correspond to the number presented in Fig. 2a.



**Fig. S13** (a,b,c) Photos of the Mg-Ni system material collected from the autoclave after a series of hydrogenation/dehydrogenation experiments run under 20 bar H<sub>2</sub>, 100 bar of  $CH_4/H_2$  gas mixture (20 bar of partial H<sub>2</sub> pressure), and 100 bar of natural gas/H<sub>2</sub> gas mixture (20 bar of partial H<sub>2</sub> pressure). When taking photos, the samples were placed in a glove box filled with Ar.

Fig. S13 presents a distinction between different parts of the Mg-Ni system after the gas separation experiments. The bright granules and darker powder, presented in Fig. S13a, correspond to the  $1^{st}$  and  $2^{nd}$  layers' samples (presented graphically in Fig. 2a), respectively. Fig S13b,c presents the difference in colour between MgH<sub>2</sub> light grey granules (a major part of the granules) and dark red/brown spots of Mg<sub>2</sub>NiH<sub>4</sub> (present mostly on the surface of the granules).



**Fig. S14** SE-SEM and BS-SEM images obtained at different magnifications for the samples collected from the Mg-Ni system after cyclic separation experiments (15 kV):  $1^{st}$  (a,b,c),  $2^{nd}$  (d,e,f,),  $3^{rd}$  (g,h,i),  $4^{th}$  (j,k,l),  $5^{th}$  (m,n,o). The layers' sample numbers correspond to the number presented in Fig. 2a. Scale bars: 500 µm (a), 100 µm (d,e,f,g,h,l,j,m), 30 µm (b,k,l,n,o), 20 µm (c).

Fig. S14 shows the morphology of samples collected from different layers of the Mg-Ni system after separation experiments. The darker (less dense) part of the material is mostly composed of MgH<sub>2</sub> while the brighter (denser) part of the Ni-containing phases.



**Fig. S15** SE-SEM images obtained at different magnifications for the 1<sup>st</sup> layer' sample collected from the Mg-Ni system after cyclic separation experiments (30 kV). The sample number corresponds to the number presented in Fig. 2a. Scale bars: 5  $\mu$ m (a), 500 nm (b), 100 nm (c).



**Fig. S16** SE-SEM (a) and corresponding EDX images (b,c,d) of the 1<sup>st</sup> layer' sample collected from the Mg-Ni system after cyclic separation experiments (30 kV). The sample number correspond to the number presented in Fig. 2a.



**Fig. S17** SE-SEM micrographs and corresponding EDX images of the samples collected from different layers of the Mg-Ni system after cyclic separation experiments (15 kV):  $1^{st}$  (a),  $2^{nd}$  (b),  $3^{rd}$  (c),  $4^{th}$  (d),  $5^{th}$  (e). The sample numbers correspond to the number presented in Fig. 2a. Scale bars in all cases: 250  $\mu$ m



**Fig. S18** DSC (a) and TG (b) curves obtained for the hydrogenated Mg-Ni system powder samples collected from different layers (the sample numbers correspond to the number presented in Fig. 2a) in the autoclave after a series of hydrogenation/dehydrogenation experiments run under 20 bar H<sub>2</sub>, 100 bar of  $CH_4/H_2$  gas mixture (20 bar of partial H<sub>2</sub> pressure), and 100 bar of natural gas/H<sub>2</sub> gas mixture (20 bar of partial H<sub>2</sub> pressure). The heating was performed at 10 °C/min. The heat flow has been normalized to the mass of the sample.



**Fig. S19** Results of MS performed on the gases desorbed during TG/DSC experiment (10 °C/min) of hydrogenated Mg-Ni system (2<sup>nd</sup> layer' sample presented in Fig. 2a). The MS was connected directly to the outlet line of TG/DSC device.



**Fig. S20** XRD patterns and Rietveld refinement plots of the samples collected from different layers of the Mg-Ni system after cyclic separation experiments: 1<sup>st</sup> (a), 2<sup>nd</sup> (b), 3<sup>rd</sup> (c), 4<sup>th</sup> (d), 5<sup>th</sup> (e). The patterns were fitted using the following phases: Mg ICSD-258531 (ICSD release 2024.1), Mg<sub>2</sub>NiH<sub>0.29</sub> ICSD-49534 (ICSD release 2024.1), Mg<sub>2</sub>NiH<sub>4</sub> - monoclinic ICSD-201606 (ICSD release 2024.1), Mg<sub>2</sub>NiH<sub>4</sub> cubic ICSD-600065 (ICSD release 2024.1), MgH<sub>2</sub> ICSD-155807 (ICSD release 2024.1), MgNi<sub>2</sub> ICSD-193568 (ICSD release 2024.1), MgO ICSD-158103 (ICSD release 2024.1), Ni ICSD-8688 (ICSD release 2024.1). The sample numbers correspond to the number presented in Fig. 2a.

S7. Mg-Fe system after separation of  $H_2$  from  $CH_4/H_2$  and  $NG/H_2$  gas mixtures



**Fig. S21** Pictures of the powders of the Mg-Fe system material after cycles of hydrogenation/dehydrogenation experiments under 20 bar H<sub>2</sub>, 100 bar of  $CH_4/H_2$  gas mixture (20 bar of partial H<sub>2</sub> pressure), and 100 bar of natural gas/H<sub>2</sub> gas mixture (20 bar of partial H<sub>2</sub> pressure). The sample/layer numbers correspond to the number presented in Fig. 4a.



**Fig. S22** SE-SEM micrographs and corresponding EDX images of the samples collected from different layers of the Mg-Fe system after cyclic separation experiments (15 kV):  $1^{st}$  (a),  $2^{nd}$  (b),  $3^{rd}$  (c),  $4^{th}$  (d). The sample numbers correspond to the number presented in Fig. 4a. Scale bars in all cases: 250 µm.



**Fig. S23** SE-SEM and BS-SEM images obtained at different magnifications for the samples collected from the Mg-Fe system after cyclic separation experiments (15 kV):  $1^{st}$  (a,b,c),  $2^{nd}$  (d,e,f,),  $3^{rd}$  (g,h,i),  $4^{th}$  (j,k,l). The layers' sample numbers correspond to the number presented in Fig. 4a. Scale bars: 100 µm (a,b,d,e,g,h,j,i), 30 µm (c,f,i,l).



**Fig. S24** DSC (a) and TG (b) curves obtained for the hydrogenated Mg-Fe system powder samples collected from different layers (the sample numbers correspond to the number presented in Fig. 4a) in the autoclave after a series of hydrogenation/dehydrogenation experiments run under 20 bar H<sub>2</sub>, 100 bar of  $CH_4/H_2$  gas mixture (20 bar of partial H<sub>2</sub> pressure), and 100 bar of natural gas/H<sub>2</sub> gas mixture (20 bar of partial H<sub>2</sub> pressure). The heating was performed at 10 °C/min. The heat flow has been normalized to the mass of the sample.



**Fig. S25** XRD patterns and Rietveld refinement plots of the samples collected from different layers of the Mg-Fe system after cyclic separation experiments:  $1^{st}$  (a),  $2^{nd}$  (b),  $3^{rd}$  (c),  $4^{th}$  (d). The patterns were fitted using the following phases: Fe ICSD-48382 (ICSD release 2024.1), Mg ICSD-258531 (ICSD release 2024.1), Mg<sub>2</sub>FeH<sub>6</sub> ICSD-107500 (ICSD release 2024.1), MgH<sub>2</sub> ICSD-155807 (ICSD release 2024.1), MgO ICSD-158103 (ICSD release 2024.1), Ni ICSD-8688 (ICSD release 2024.1). The sample numbers correspond to the number presented in Fig. 4a.

## S8. Mg-Ni system after separation of H<sub>2</sub> from H<sub>2</sub>/CO<sub>2</sub> gas mixtures



**Fig. S26** Pictures of the flakes and powder of the Mg-Ni system material after four cycles of hydrogenation/dehydrogenation experiments under ~ 50 bar of  $H_2/CO_2$  gas mixture (0.5 bar of partial  $CO_2$  pressure). The sample names correspond to those presented in Fig. 5e.



**Fig. S27** (a,b) Photos of the Mg-Ni system material after four cycles of hydrogenation under ~ 50 bar of  $H_2/CO_2$  gas mixture (0.5 bar of partial  $CO_2$  pressure) in powder and flake form, respectively. The sample names correspond to those presented in Fig. 5e. The powdered sample partially covers the flakes; c) XRD pattern (and Rietveld refinement plot) of the hydrogenated Mg-Ni system flake sample. The concentrations of the main phases are presented in wt.%.

### S9. Mg-Fe system after separation of H<sub>2</sub> from H<sub>2</sub>/CO<sub>2</sub> gas mixtures



**Fig. S28** Picture of the powder of the Mg-Fe system material after four cycles of hydrogenation/dehydrogenation experiments under  $\sim$  50 bar of H<sub>2</sub>/CO<sub>2</sub> gas mixture (0.5 bar of partial CO<sub>2</sub> pressure).



**Fig. S29** XRD pattern (and Rietveld refinement plot) of the hydrogenated Mg-Fe system powder sample (after four cycles of hydrogenation/dehydrogenation experiments under  $\sim 50$  bar of H<sub>2</sub>/CO<sub>2</sub> gas mixture). The concentrations of the main phases are presented in wt.%

### S10. Literature comparison of different H<sub>2</sub> separation techniques

Separation method	Gas mixture	H₂ conc. / vol. %	H₂ rcy / %	H₂ purity / %	Ref.
PSA	H <sub>2</sub> :N <sub>2</sub> :CH <sub>4</sub> :Ar (58:25:15:2)	58	55.5-75.3	99.25-99.97	[S4]
Cryo-Cooling	H <sub>2</sub> :CH <sub>4</sub> :C <sub>2</sub> H <sub>6</sub> :C <sub>3</sub> H <sub>8</sub> : N <sub>2</sub> :C <sub>6</sub> H <sub>6</sub> (55:39:2:1:2:1)	55	25.1	88.1	[\$5]
Membrane	$H_2$ + $NH_3$	60	86	<99	[S6]
Membrane+PSA	$H_2$ +C $H_4$	4	<60	99.97	[S7]
Electrochemical pump	$H_2$ + $CH_4$	5	87	$<300 \text{ ppm CH}_4$	[S8]
	$H_2$ + $CH_4$	5	91.72	<99.1	[S9]
Metal hydride:					
LaNi <sub>4.8</sub> Mn <sub>0.3</sub> Fe <sub>0.1</sub>	$H_2$ + $CH_4$	10	74	>99	[S10]
MgH <sub>2</sub>	$H_2$ + $CH_4$	10	~99.9		[S3]
	$H_2$ +C $H_4$	20		>99.9	This work

Table S2 Comparison of different H<sub>2</sub> separation techniques.

 $H_2$  conc. – hydrogen concentration

H<sub>2</sub> rcy – hydrogen recovery

PSA – Pressure Swing Absortpion technique

Table S2 shows a general comparison of H<sub>2</sub> separation techniques. Some of the methods, like adsorption techniques (e.g. PSA- Pressure Swing Adsorption) are chosen, when high H<sub>2</sub> contents in the gas mixture are awaited. The adsorbents used in this case are usually active in absorbing all components of the gas mixture except  $H_2$ .  $H_2$  is rather flowing through the adsorption columns. In the metal hydride reactors (like in the case of Mg-based systems),  $H_2$  is the gas, to be captured by the used adsorbents (metals). The  $H_2$ -rich gas mixtures, as being used with PSA, would not be economically to be separated by metal hydride reactors because of increased volume and adsorbents demand. Nevertheless, the metal hydride-based separation systems performance could be compared with electrochemical hydrogen pumps. Both technologies show satisfactory H<sub>2</sub> recovery and purity. However, it should be emphasized that Table S2 present mainly data obtained under different laboratory conditions (e.g. batch reaction in case of MgH<sub>2</sub>-based separation and experiments using adapted fuel cell test stand in the case of electrochemical pumps). Therefore, the proper benchmarking of different technologies requires detailed analysis of separation techniques on the system level. In this case, it includes not only additional hydrogen separation parameters (like amount of hydrogen separated in time) but also the economic aspects including for example material costs and necessary energy operation input.

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