

Supplementary Information

Ultrathin Oxygen Deficient SnO_x Films as Electron Extraction Layers for Perovskite Solar Modules

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S1. EXPERIMENTAL SECTION

1.1 Materials

Materials used in the experiments include, tin oxide (Alfa), home-made formamidinium lead iodide (FAPbI₃) black powder (KRICT), home-made methylammonium bromide (MAPbBr₃), methylammonium chloride (MACl, Sigma-Aldrich), poly(triarylamine) (PTAA) (MS Solution), *N,N*-dimethylformamide (DMF; 99.8%, Sigma-Aldrich), dimethyl sulfoxide (DMSO; 99.9%, Sigma-Aldrich), chlorobenzene (99.8%, Sigma-Aldrich), toluene (99.8%, Sigma-Aldrich), 2-methoxyethanol (2-ME; 99.8%, Sigma-Aldrich), ethanol (99.9%, DUKSAN PURE CHEMICALS), and ethylacetate (99.0%, DUKSAN PURE CHEMICALS).

1.2 Fabrication of perovskite unit cells

As transparent electrodes, glass/ITO (indium tin oxide on glass) substrates were prepared by cleaning with a special detergent followed by ultra-sonication in deionized (DI) water, acetone, and isopropyl alcohol. After drying, an ALD SnO_x or nanoparticle SnO_x (*np*-SnO_x) was first deposited onto the precleaned ITO substrate. ALD SnO_x was grown at 190°C in a commercial ALD system (CN-1, Atomic Premium). The precursors used were tetrakis(dimethylamino)tin (TDMASn) and deionized H₂O for the tin and oxygen sources, respectively. The conventional thermal ALD cycle consisted of a dose of TDMASn for 0.5 s, followed by Ar purge of 10 s, followed by a dose of H₂O for 0.1 s, and then Ar purge of 10 s. The plasma-modified ALD cycle as shown in **Figure 1a** was grown at 190°C. The technique consisted of a dose of TDMASn for 0.5 s, followed by Ar purge of 10 s, followed by a dose of Ar plasma for 3 s which is intended to remove ligands and activate the TDMASn surface, followed by Ar purge of 10 s, followed by a dose of TDMASn for 0.5 s, followed by Ar purge of 10 s, followed by a dose of H₂O for 0.1 s, and then Ar purge of 10 s. ALD SnO_x films were

grown for 79 cycles which is approximately 10 nm, and were post-annealed at 150°C for 1 h in air.

Nanoparticle SnO₂ (*np*-SnO₂) solution (diluted with H₂O to 2.5 wt%) were obtained from Alfa Aesar was coated by spin coating at 3000 rpm for 30 s, and the substrates were annealed on a hotplate at 150°C for 1 h. The perovskite solution was prepared by dissolving 800 mg of FAPbI₃, 30 mg of MACl, and 30 mg of MAPbBr₃ in an DMF/DMSO (8:1 v/v) mixed solvent. The (FAPbI₃)_{0.95}(MAPbBr₃)_{0.05} perovskite solutions were spin-coated onto the ITO/SnO_x substrates at 500 rpm for 5 s, 1000 rpm for 8 s, and 5000 rpm for 12 s, and the ethylacetate in the final spin-stage was dripped onto the substrate during spin coating. After that, the substrates were dried on a hotplate at 100°C for 1 h, 150°C for 4 min. PTAA solutions were prepared in toluene (10 mg/1 mL) with octylammonium bis(trifluoromethylsulfonyl)imide of 2.5 mg. PTAA solutions were spin-coated onto the ITO/SnO_x/(FAPbI₃)_{0.95}(MAPbBr₃)_{0.05} substrates at 3000 rpm for 30 s. Finally, a metal electrode consisting of Au (80 nm) with an area of 9.94 mm² was deposited by thermal evaporation in a vacuum for all devices.

1.3 Fabrication of perovskite modules (9-stripe cells connected in series)

The perovskite solar module composed of nine stripes in series on 7 × 7 cm² glass/ITO substrates patterned by a laser patterning system (EO TECHNICS). P1 lines were patterned by scribing to separate the ITO substrate with a power of 1.4 W. As transparent electrodes, glass/ITO (indium tin oxide on glass) substrates were prepared by cleaning with a special detergent followed by ultra-sonication in deionized (DI) water, acetone, and isopropyl alcohol. After drying, an ALD SnO_x or nanoparticle SnO₂ (*np*-SnO₂) was first deposited onto the precleaned ITO substrate. ALD SnO_x was grown at 190°C in a commercial ALD system (CN-1, Atomic Premium). The precursors used were tetrakis(dimethylamino)tin (TDMASn) and deionized H₂O for the tin and oxygen sources, respectively. The conventional thermal ALD

cycle consisted of a dose of TDMASn for 0.5 s, followed by Ar purge of 10 s, followed by a dose of H₂O for 0.1 s, and then Ar purge of 10 s. The plasma-modified ALD cycle consisted of a dose of TDMASn for 0.5 s, followed by Ar purge of 10 s, followed by a dose of Ar plasma for 3 s, followed by Ar purge of 10 s, followed by a dose of TDMASn for 0.5 s, followed by Ar purge of 10 s, followed by a dose of H₂O for 0.1 s, and then Ar purge of 10 s. ALD SnO_x films were grown for 79 cycles which is approximately 10 nm, and were post-annealed at 150°C for 1 h in air.

For the modules, nanoparticle SnO_x was coated by shearing (PCM-200, MITSUBISHI ELECTRIC) at 0.5 mm/s followed by thermal annealing at 100°C for 30 min. The perovskite solution was prepared by dissolving 800 mg of FAPbI₃, 30 mg of MAcl, and 30 mg of MAPbBr₃ in an DMF/DMSO (8:1 v/v) mixed solvent. And then the perovskite solution were diluted using same solvent, with volume ratio with 1:0.2 (perovskite solution : solvent). The diluted (FAPbI₃)_{0.95}(MAPbBr₃)_{0.05} perovskite solutions were spin-coated onto the ITO/SnO_x substrates at 500 rpm for 5 s, 1000 rpm for 8 s, and 3000 rpm for 10 s, and the ethylacetate in the final spin-stage was dripped onto the substrate during spin coating. After that, the substrates were dried on a hotplate at 100°C for 1 h, 150°C for 4 min. PTAA solutions were prepared in toluene (10 mg/1 mL) with octylammonium bis(trifluoromethylsulfonyl)imide of 2.5 mg. PTAA solutions were spin-coated onto the ITO/SnO_x/(FAPbI₃)_{0.95}(MAPbBr₃)_{0.05} substrates at 3000 rpm for 30 s. ALD SnO_x/np-SnO_x/perovskite/PTAA layers were coated and P2 lines were scribed to expose the bottom ITO substrate to connect the series linkages between cells with a power of 1 W. Finally, Au electrodes formed by thermal evaporation and each sub-modules were separated by laser scribing to form P3 lines with a power of 0.12 W.

1.4 Photocurrent density vs. voltage (*J-V*) measurements. Illuminated *J-V* characteristics were measured using a Keithley 2420 sourcemeter. The standard 100 mW/cm² (1 SUN) illumination was generated by a Newport Oriel Class A 91195A solar simulator using a 450 W

Xe-lamp (Oriel) with an AM 1.5 G filter, while the light intensity was calibrated by a Si-reference cell certified by NREL. The J - V curves were measured from 1.5 V to -0.2 V along the reverse scan direction, with a step voltage and scan speed fixed at 10 mV and 150 mV/s, respectively. All devices were measured with a metal mask with an active area of 0.094 cm².

1.5 Damp heat test (85°C and 85% relative humidity). A climatic test was conducted in a chamber (C 4-340 E series, Votschtechnik), and was carried out in a chamber set to a constant temperature (85°C) and constant humidity (85%). The efficiency of the PSCs was measured under illumination at AM 1.5 G after removing the devices from the chamber and cooling them down to room temperature.

1.6 Energy-filtered photoemission electron microscopy (EF-PEEM) analysis. Energy-filtered photoemission electron microscopy (EF-PEEM) was performed under UHV conditions (base pressure of 2×10^{-11} mbar) in the Bristol Ultraquiet NanoESCA Laboratory. Prior to the analysis, the samples were sputtered with 0.5 kV Ar⁺ ions (5×10^{-5} mbar) at 45° for 2 minutes (total sputter flux of approximately 9 μA minutes) in a separate preparation chamber to remove surface contaminants. Following preparation, the samples were transferred into the EF-PEEM chamber equipped with a monochromated He I (21.2 eV) excitation light source. The measurements were carried out with an extraction field of 12 kV, 37.6 μm field of view, and at 50 eV pass energy (corresponding to a nominal instrument resolution of 100 meV). A 150 μm contrast aperture was inserted into the back focal plane to improve lateral resolution.

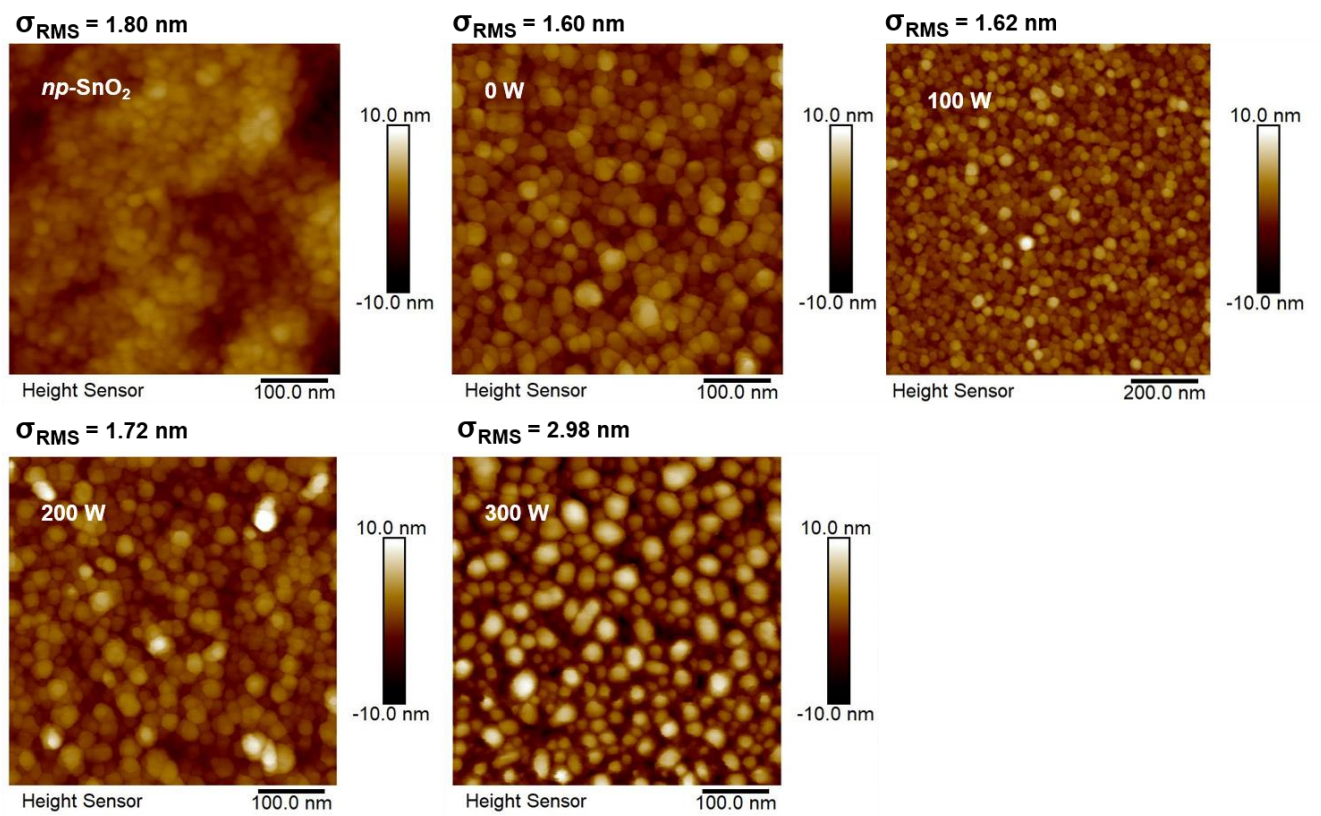


Figure S1. AFM images of the surfaces of the $np\text{-SnO}_2$, 0 W, 100 W, 200 W, and 300 W.

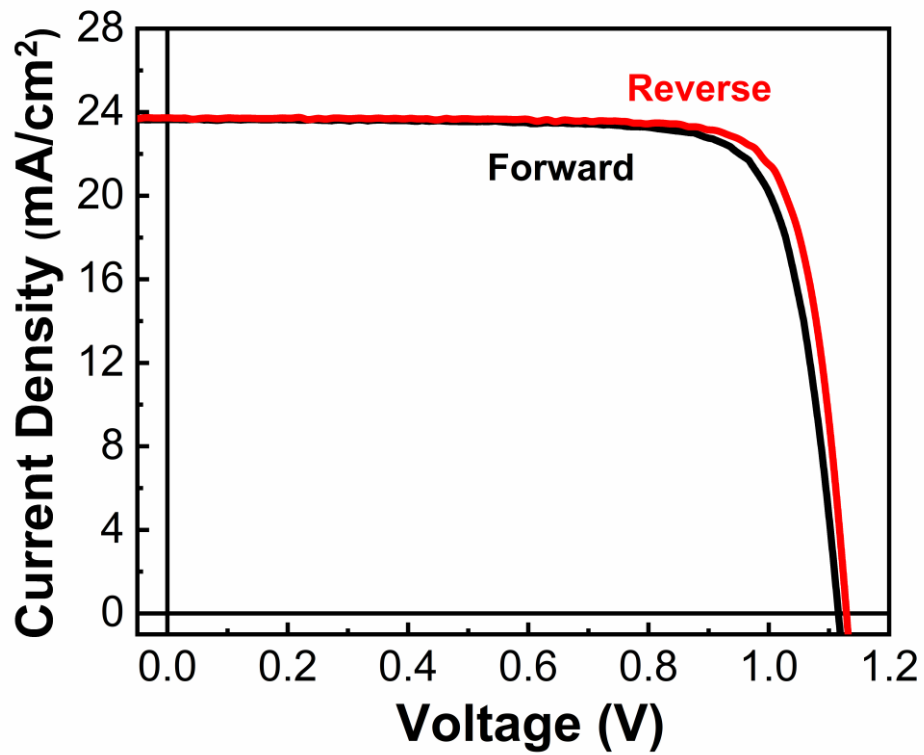


Figure S2. *J-V* characteristics of the photovoltaic device with PMALD SnO_x layer (200 W) under 1 SUN for reverse and forward scans.

Table S1. The photovoltaic device parameters of PMALD SnO_x (200 W) under forward and reverse scans.

	V_{oc} (V)	J_{sc} (mA/cm ²)	FF (%)	η (%)
Forward	1.11	23.6	79.4	21.0
Reverse	1.13	23.7	81.5	21.8

Table S2. Photovoltaic parameters. Average perovskite solar-cell performance values for with and without (reference) PMALD SnO_x electron extraction layers deposited with various plasma power.

	J_{SC} (mA/cm ²)	V_{OC} (V)	FF (%)	η (%)	R_{SH} ($\Omega \cdot \text{cm}^2$)	R_S ($\Omega \cdot \text{cm}^2$)
Reference	23.4 ± 0.3	1.07 ± 0.03	77.2 ± 0.9	19.4 ± 0.6	1.3 ± 1.8 × 10 ⁴	3.6 ± 0.4
0 W	23.6 ± 0.3	1.11 ± 0.00	79.1 ± 0.4	20.7 ± 0.3	0.5 ± 0.3 × 10 ⁴	3.0 ± 0.1
100 W	23.3 ± 0.1	1.11 ± 0.01	79.1 ± 1.0	20.4 ± 0.4	1.7 ± 1.5 × 10 ⁴	2.9 ± 0.2
200 W	23.4 ± 0.3	1.13 ± 0.00	80.2 ± 0.9	21.1 ± 0.4	1.3 ± 1.5 × 10 ⁴	2.8 ± 0.1
300 W	23.2 ± 0.1	0.98 ± 0.08	68.5 ± 3.7	15.6 ± 1.4	0.5 ± 0.1 × 10 ⁴	6.2 ± 0.7

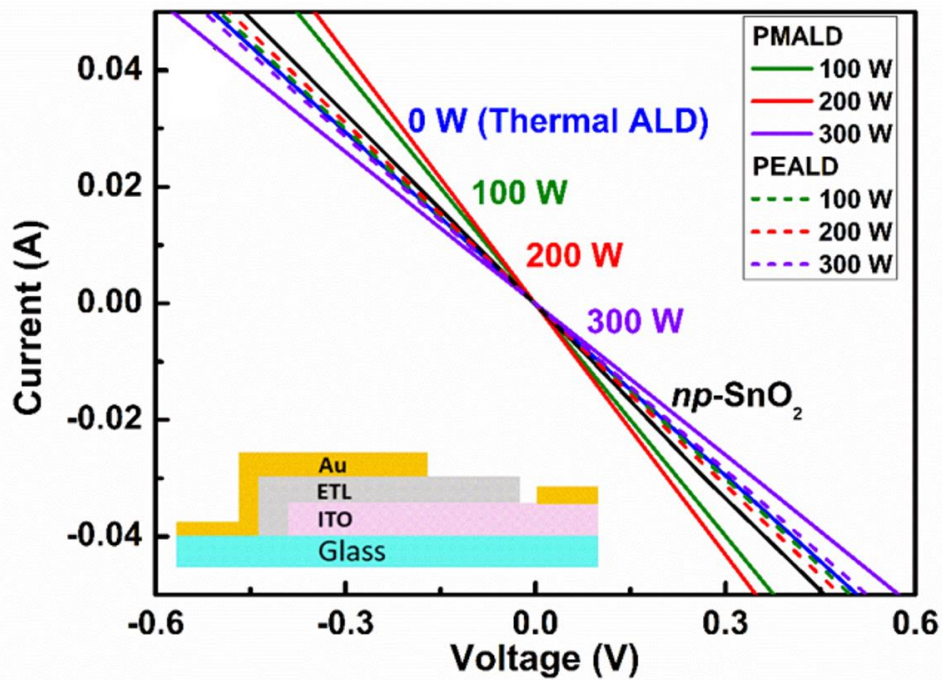


Figure S3. Current vs. voltage (I - V) scans of the glass/ITO/SnO_x/Au stacks featuring nanoparticle SnO₂ films (np -SnO₂) and SnO_x layers by thermal ALD, PEALD, and PMALD with varying plasma power. These measurements clearly show that the conductance of np -SnO₂ spin coated films, thermal ALD and PEALD deposited SnO_x obtained with different oxygen plasma powers are very similar (98.2 ± 5.0 mS). On the other hand, PMALD films show an increase in conductance with increasing plasma power from 137.9 mS (100 W) to 143.6 mS (200 W). Increasing the plasma power to 300 W leads to a decrease in the SnO_x conductance to 86.7 mS.

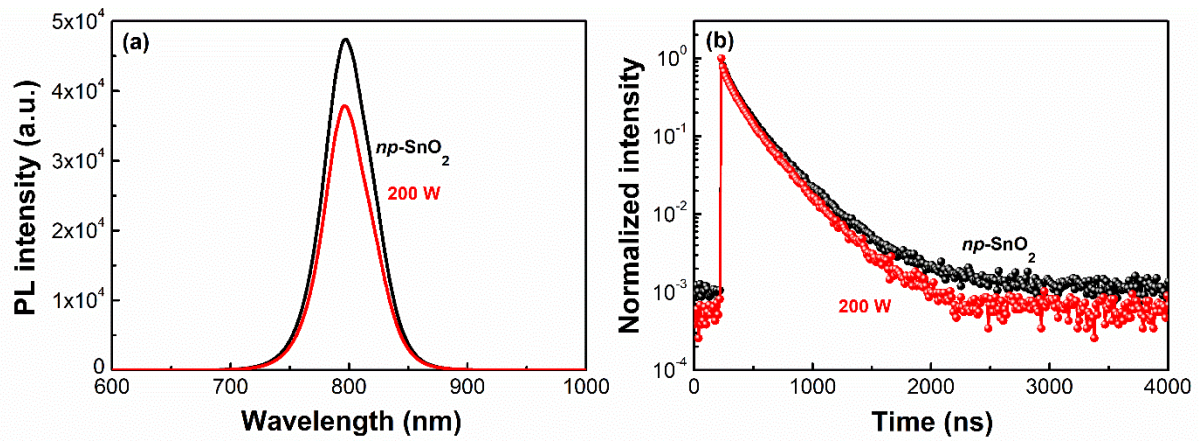


Figure S4. Steady-state photoluminescence (PL) (a) and time-resolved PL spectroscopy (b) of glass/ITO/ETL/perovskite samples with np -SnO₂ (Reference) and 200W PMALD SnO_x. The PL intensity of the perovskite decreased for the PMALD SnO_x with 200 W, in comparison to the reference case, strongly suggesting a more efficient electron extraction. The TRPL data was fitted to a double exponential decay with the fast decay (τ_1), associated with interfacial recombination of free carriers, and the slow decay (τ_2), linked to radiative decay. The TRPL decay times τ_1 and τ_2 are 301.3 and 109.2 ns, respectively, for the np -SnO₂ reference, whereas the decay times drop to 252.5 and 85.7 ns for the PMALD SnO_x case with 200 W. Such reduction in the TRPL decay times indicate fast electron transfer from the perovskite film into the PMALD SnO_x film, hence greatly suppressed carrier recombination, leading to enhanced V_{OC} .¹ The enhanced electron transfer from the perovskite film to the PMALD SnO_x is possibly caused by reduced trap density in the perovskite²⁻⁷ and the enhanced conductivity in the PMALD SnO_x compared to the np -SnO₂ reference.

Table S3. Photovoltaic parameters. The photovoltaic device parameters of PMALD SnO_x (200 W) with thickness optimization.

PMALD SnO _x 200 W	J_{SC} (mA/cm ²)	V_{OC} (V)	FF (%)	η (%)
0 nm (<i>np</i> -SnO ₂)	23.4	1.10	77.2	19.9
5 nm	23.2	1.10	79.7	20.2
10 nm	23.6	1.10	82.2	21.3
15 nm	23.3	1.10	81.1	20.7
20 nm	23.3	1.10	79.1	20.2

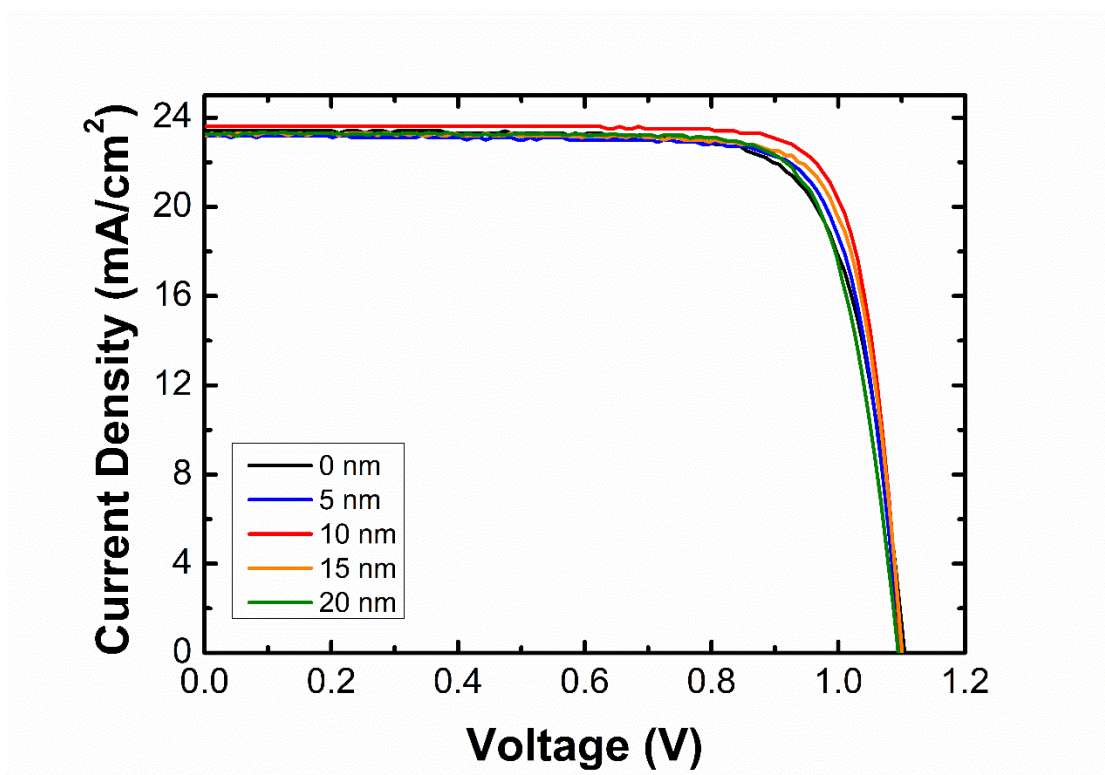


Figure S5. Photovoltaic performance depending on thickness variation. Current density vs. voltage (J - V) scans of PMALD SnO_x with varying film thickness.

Table S4. Photovoltaic performance parameters of 5 cm × 5 cm modules with a *np*-SnO₂ layer (Reference) and a plasma-modified ALD SnO_x layer (200 W) under 1 SUN.

	J_{SC} (mA/cm ²)	V_{OC} (V)	FF (%)	η (%)	R_{SH} ($\Omega \cdot \text{cm}^2$)	R_S ($\Omega \cdot \text{cm}^2$)
<i>np</i> -SnO ₂	2.3	11.1	71.0	17.9	1.3×10^5	5.2×10^2
200W PMALD	2.4	11.4	74.8	20.1	7.1×10^5	4.5×10^2

Table S5. Reports on the efficiency of perovskite solar modules.

ETL	HTL	Active area (cm ²)	V_{oc} (V)	J_{sc} (mA/cm ²)	FF (%)	η (%)	Ref.
SnO ₂	Spiro	15.03	8.37	3.17	77.97	20.71	Angew. Chem.Int. Ed. 2024 , 63, e2023161. ⁸
SnO ₂	Spiro	48.00	11.00	-	69.00	18.80	Chemical Engineering Journal 2023 , 456, 140894. ⁹
TiO ₂	PTAA	45.60	16.07	1.52	75.35	18.45	Nat. Comm. 2022 , 89. ¹⁰
TiO ₂	PTAA	30.24	-	-	-	20.99	Joule 2022 , 6, 1689. ¹¹
TiO ₂	Spiro	20.00	12.15	2.29	77.90	21.67	Science 2022 , 375, 302. ¹²
TiO ₂	Spiro	24.63	10.16	2.75	82.00	22.87	Nat. Nanotech. 2022 , 17, 598. ¹³
-	PTAA	27.14	8.715	2.83	75.41	18.60	Nat. Energy 2021 , 6, 633. ¹⁴
-	PTAA	18 (aperture)	5.809	4.25	78.00	19.3	Science 2021 , 373, 902. ¹⁵
TiO ₂	PTAA	112	7.64	2.51	72.09	13.82	Joule 2021 , 5, 481-494. ¹⁶
TiO ₂	Spiro	23.27	-	-	78.5	20.75	Energy Environ. Sci. 2021 , 14, 4903. ¹⁷
TiO ₂	PTAA	42.8	16.05	1.49	70.9	17.05	Nano Energy, 2021 , 82, p.105685. ¹⁸
TiO ₂	Spiro	52	12.03	-	54.9	11.6	Solar Energy Materials and Solar Cells, 2021 , 230, 111189. ¹⁹
TiO ₂ /SnO ₂	Spiro	21	6.71	3.68	73.44	18.13	Joule 2020 , 4, 1035. ²⁰
SnO ₂	Spiro	13.8	-	-	-	13.1	Solar Energy Materials and Solar Cells, 2018 , 185, 136. ²¹
SnO ₂	PTAA	16.07	6.54	3.30	69.00	14.89	Nature Communications, 2018 , 9, 4609. ²²
TiO ₂	Spiro	36.1	10.2	1.97	75.7	15.7	Nature, 2017 , 550(7674), pp.92-95. ²³
TiO ₂	PTAA	40	10.5	2.10	70.16	15.5	J. Mater. Chem. A, 2016 , 4, 17636-17642. ²⁴
SnO₂	PTAA	23.2	11.3	2.4	74.8	20.1	This work

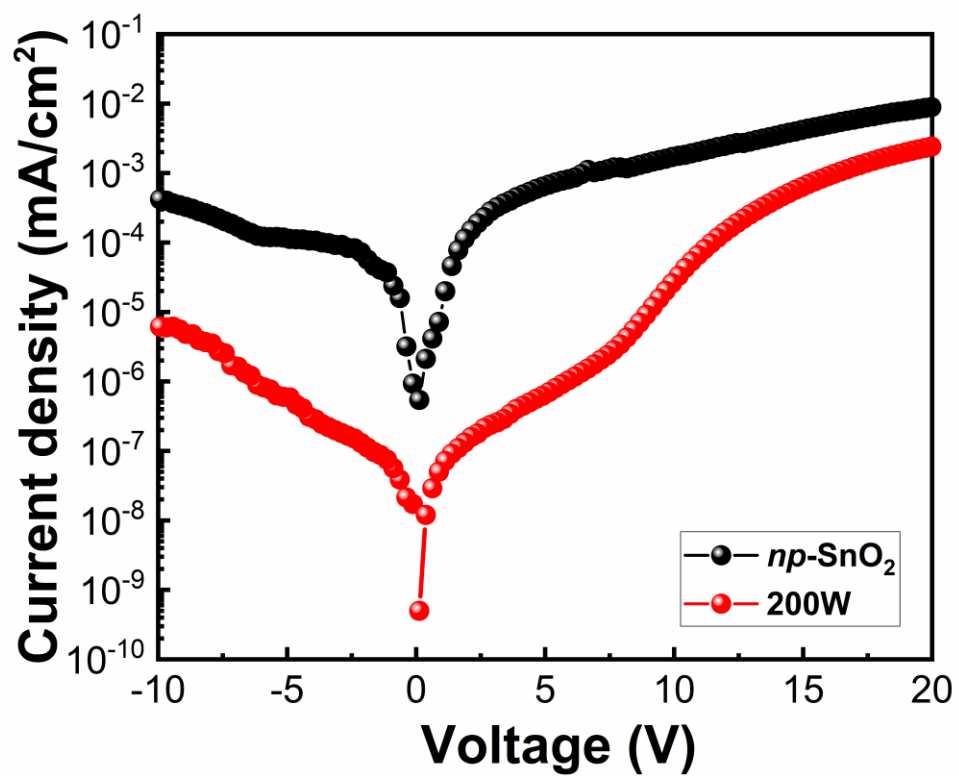


Figure S6. Module dark J - V curves. 5 cm \times 5 cm modules with a np -SnO₂ layer (Reference) and upon including a 200 W PMALD SnO_x electron extraction layer.

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