## Supplementary information

# One-step synthesis of hollow spherical Co/Ni hydroxides as multifunctional polysulfide mediators to steer sulfur redox kinetics for high performance lithium-

#### sulfur batteries

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#### 1. Material characterizations

The crystallinity of the materials was characterized a BRUKER D2 PHASER X-Ray Diffractometer with Cu K $\alpha$  radiation ( $\lambda = 1.5418$  Å). The micro-structure of the materials was monitored by using scanning electron microscope quipped with an energy dispersive X-ray spectrometer (SEM-EDX, Hitachi S4800) and transmission electron microscope (TEM, Hitachi H-800 TEM at 200 kV). Electron paramagnetic resonance (EPR, Bruker A300) was adopted to study the vacancies in the materials. The porous feature of the materials was proofed by the N<sub>2</sub> adsorption and desorption technique (Micromeritics ASAP 2020). X-ray photoelectron spectroscopy (XPS) was collected on a Thermo ESCALAB 250XI machine equipped with an Al K $\alpha$  X-ray source. Thermal gravity analysis (TGA, Thermo Scientific Nicolet iS50) was used to investigate the thermal behaviors of the materials. The Ni and Co K-edge X-ray absorption fine structure (XAFS) spectra were collected at BL11B beamline of Shanghai Synchrotron Radiation Facility, China. The acquired EXAFS data were analyzed by adopting a standard procedure through the ATHENA module in the IFEFFIT software packages <sup>1, 2</sup>. The Wavelet transform (WT) of k<sup>2</sup>-weighted EXAFS spectra (WT-EXAFS) was performed through the Hama software.

#### 2. Electrochemical measurements

Li-S Cell Assembly and Measurements. The active materials, Super P, and binder (polyvinylidene fluoride, PVDF) in a weight ratio of 7:2:1 was dispersed into N-methyl pyrrolidone (NMP) to form a stable slurry. The working electrode was prepared by coating the slurry of onto a round-shaped Al foil (1.1 cm<sup>2</sup>). The electrochemical performance of the as-fabricated working electrodes was examined using CR2032 coin-type cells. 1 M lithium bis(trifluoro-methane-sulfonyl) imide (LiTFSI) and 0.1 M of LiNO<sub>3</sub> in a mixture of dioxolane (DOL) and 1,2-dimethoxy ethane (DME) with a volume ratio of 1: 1 was used as the electrolyte. The loading density of the active material is about 1.0 mg cm<sup>-2</sup> and the electrolyte/sulfur ratio (E/S) is 15  $\mu$ L mg<sup>-1</sup> for each cell, which is a popular parameter for coin-type Li-S cells in the lab.<sup>3-5</sup> Lithium foil was used as both the counter electrode and the reference electrode. The loading density of Li foil is about 21 mg cm<sup>-2</sup>. The ratio between the theoretical capacities of the negative electrode and the positive electrode (N/P) is calculated to be 50.7, which is comparable to the previous reported value for the coin-type Li-S cells in the lab (N/P = 61.2).<sup>6</sup> Celgard 2400 membranes were used as separators. The electrochemical performance of the working electrodes was tested by using a battery test system (NEWARE CT-4008Tn). Li-S cells with a higher sulfur loading density (5.2 mg cm<sup>-2</sup>, corresponding to N/P = 9.7) and lean electrolyte content (E/S = 4  $\mu$ L mg<sup>-1</sup>) were also assembled. Cells were galvanostatically cycled within the potential window of 1.7-2.8 V vs. Li/Li<sup>+</sup> and activation treatment at a low current of 0.05 C was conducted before the cycling test. Electrochemical impedance spectra (EIS) were collected for the cells at an open potential over the frequency range from 100 kHz to

0.01 Hz on an electrochemical workstation (Shanghai Chen Hua, CHI660E). Cyclic voltammetry (CV) was performed on the cathodes at various scan rate ranging from 0.1 to 0.4 mV s<sup>-1</sup> within the potential window of 1.7-2.8 V.

**Preparation of Li<sub>2</sub>S<sub>6</sub> and Li<sub>2</sub>S<sub>8</sub> Solutions**. Sulfur and Li<sub>2</sub>S in different mass ratios (1:5 for Li<sub>2</sub>S<sub>6</sub> and 1:7 for Li<sub>2</sub>S<sub>8</sub>) were added into a mixed solvent of DME and DOL (1:1, v/v) and the mixture was vigorously stirred for 12 h until a dark yellow solution was formed.

 $Li_2S_6$  adsorption test of the materials. 10 mg of the materials were dispersed into 3.0 mL of  $Li_2S_6$  solution (0.05 M) and stirred for 4 h. The UV-vis spectra (Shimadzu, UV-2700) of the supernatant solution was checked to evaluate the adsorption performance of the materials.

Symmetric cell fabrication and test: Bare H-Co and H-Ni, as well as the bimetallic H-Co<sub>x</sub>Ni<sub>1-x</sub> were used as the active materials. The working electrodes for symmetric cells were fabricated in the same way as electrodes for LSBs. Two pieces of the same electrodes were used as identical working and counter electrodes.  $Li_2S_6$  solution (0.5 M) containing 1 M LiTFSI in DOL/DME (v/v, 1/1) was used as the electrolyte. The symmetric cells were assembled in the same way with that of Li-S cointype cells. CV measurements were performed at a scan rate of 5 mV/s.

Measurement of nucleation of Li<sub>2</sub>S. The nucleation of Li<sub>2</sub>S were tested in CR2032 coin-type cells. The working electrodes were prepared by dispersing the host materials in absolute ethyl alcohol to form a slurry which was subsequently coated on a carbon paper disk (1.1 cm<sup>2</sup>) and dried at 60 °C overnight. Li foil served as the counter electrode, 20  $\mu$ L of 0.25 M Li<sub>2</sub>S<sub>8</sub> solution containing 1.0 M LiTFSI was used as catholyte, and 20  $\mu$ L of 1.0 m LiTFSI in DOL/DME (v/v=1:1) solution as anolyte. The cells were firstly discharged to 2.19 V at a constant current of 0.112 mA to reduce the higher-order LiPSs to Li<sub>2</sub>S<sub>4</sub> and then further held at 2.05 V until the current decreased to 0.01 mA to monitor the Li<sub>2</sub>S nucleation. Similar experimental procedure has also been adopted in the previous reports.<sup>3, 5, 7</sup>

#### 3. DFT calculations

The monolayered model was constructed based on the previous reports <sup>8-10</sup>. Dmol3 module in the Material Studio software package was used for DFT calculations. The generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) functional and the double numerical plus d-functions (DNP) basis set were used in geometry optimization. A vacuum layer of 15 Å was established along the planer direction of the materials. The convergence criteria of the SCF calculation were set to  $10^{-5}$  eV. For the geometry optimization of the system, the energy, force and displacement convergence criterions were set as  $2 \times 10^{-5}$  Ha,  $4 \times 10^{-3}$  Ha and  $5 \times 10^{-3}$  Å, respectively. The adsorption energy of LiPSs on the surface of the host materials was defined as following:

 $E_b = E(Li_2S_n/host) - E(Li_2S_n) - E(host)$ 

where  $E_b$  was the electronic adsorption energy,  $E(Li_2S_n/host)$  is the electronic energy of the slab and  $Li_2S_x$  complex, E(host) and  $E(Li_2S_n)$  represent the electronic energy of the standing alone host material and  $Li_2S_x$  molecules, respectively.

### 4. Supplementary data



Fig. S1 SEM image of Cu<sub>2</sub>O spheres



Fig. S2 SEM image of H-Co $_{0.4}$ Ni $_{0.6}$  (a) and the corresponding particle size distribution curve (b).



Fig. S3 SEM images of the spherical  $H-Co_xNi_{1-x}$  materials at different magnifications.



Fig. S4 XPS Survey spectra of the materials



Fig. S5 X-ray absorption near edge structure (XANES) data of the materials



Fig. S6  $N_2$  adsorption–desorption isotherms (a) and their corresponding pore size distribution curves based on the desorption curves of H-Co, H-Ni and H-Co<sub>0.4</sub>Ni<sub>0.6</sub>.

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Samples	BET surface area	Pore size (nm)	Pore volume (cm <sup>3</sup> /g)
	$(m^{2}/g)$		
Н-Со	158.08	3.45	0.96
H-Co <sub>0.4</sub> Ni <sub>0.6</sub>	185.34	3.60	0.83
H-Ni	210.12	1.90	0.53



Fig. S7 TGA curves of H-Co<sub>x</sub>Ni<sub>1-x</sub>@S composites



**Fig. S8** (a and c) CV curves of H-Ni@S and H-Co@S electrodes at various scanning rates; (b and d) plot of CV peak current versus the square root of scan rates for H-Ni@S and H-Co@S electrodes.



Fig. S9 Charge and discharge curves of H-Ni@S (a) and H-Co@S (b) cathodes at various current rates.



Fig. S10 EIS spectra of the H-Co@S and H-Ni@S electrodes at different cycling states.



Fig. S11 Cycling stability test of the  $H-Co_{0.4}Ni_{0.6}@S$ -based cell with a higher sulfur loading density and lean electrolyte content.



Fig. S12 Comparison of Li-O bond distance between  $Li_2S/H-Ni$  (a) and  $Li_2S/H-Co_{0.4}Ni_{0.6}$  (b) systems.

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