## Supplementary Information to In-Situ Generation of Cu- and Ag-Sn Alloys for CO<sub>2</sub> Reduction from Metal Sulfides



**Figure S1**:  $CO_2RR$  on GDE coated with  $3mg \text{ cm}^2 Cu_2S$  and 20 wt. % Sustainlon binder. Measurements were conducted in triplicates and standard errors are depicted as ranges at the corresponding FE. The average of the corresponding electrode potentials in V referenced to the RHE are displayed in red, with the standard deviations indicated as shaded areas.



**Figure S2:** PXRD of  $Ag_{\&}SnS_{6}$  as prepared through high-temperature synthesis at 900°C for 72h and the powder ball milled with 5% stearic acid, which was removed by a washing step with ethanol.



**Figure S3:** EDX-Mapping of an as prepared  $Cu_3SnS_4$  gas diffusion electrode 3.16 : 1.06 : 4.0 for Cu, Sn and S when normalized to the sulfur content..



Figure S4: EDX-Mapping of a Cu<sub>3</sub>SnS<sub>4</sub> gas diffusion electrode post electrolysis.



*Figure S5:* EDX-Mapping of an as prepared  $Ag_3SnS_4$  gas diffusion electrode with stoichiometry of  $Ag_{2.56}Sn_{0.97}S_4$  (normalized to sulfur).

	Ag								O C Ag S F Cu Sn	Vlap Si	um Spe 55.4 34.5 9.7 0.2 0.2 0.1 0.0 Powere	ctrum ed by Ti	u-Q®
4		šn Sn	'   ' I 6	. î O O	• • •	 <b>  '  </b> 12	'	• •		• •	'   19		'     kaV

Figure S6: EDX mapping Ag<sub>3</sub>SnS<sub>4</sub> gas diffusion electrode surface after electrolysis.



*Figure S7:* EDX mapping of Ag<sub>8</sub>SnS<sub>6</sub> gas diffusion electrodes; normalized to sulfur a stoichiometry of 8.9:0.9:6.0 is found.



*Figure S8: EDX mapping of Ag*<sub>8</sub>*SnS*<sub>6</sub> gas diffusion electrode surface after electrolysis.



**Figure S9:** Cu 2p XPS spectra of an as prepared Cu<sub>3</sub>SnS<sub>4</sub> GDE surface (Panel A) and Ar sputtering of the surface layer for 100s (B). Panel C and D show a Cu<sub>3</sub>SnS<sub>4</sub> GDE after electrolysis, before and after sputtering of the surface. The sum of fits is displayed in red and the individual peaks were fitted as follows: Cu<sub>2</sub>O 932.2 eV (green), Cu<sub>2</sub>SnS<sub>3</sub> 932.7 eV (orange), CuO 933.7 eV (blue), 934.6 Cu(OH)<sub>2</sub> (purple) and 935.8 eV CuSO<sub>4</sub> (dark green).



**Figure S10:** S 2p XPS spectra of an as prepared  $Cu_3SnS_4$  GDE surface (Panel A) and Ar sputtering of the surface layer for 100s (B). Panel C and D show a  $Cu_3SnS_4$  GDE after electrolysis, before and after sputtering of the surface. The sum of fits is displayed in red and 3/2 and 1/2 metal sulfide signal were observed at 161.7 eV and 162.9 eV respectively.



**Figure S11:** Sn 3d XPS spectra of an as prepared  $Cu_3SnS_4$  GDE surface (Panel A) and Ar sputtering of the surface layer for 100s (B). Panel C and D show a  $Cu_3SnS_4$  GDE after electrolysis, before and after sputtering of the surface. For both the metal sulfide precursor and the resulting SnO species, signals for the Sn 3d 5/2 transition were recorded at 486.4 eV.



**Figure S12:** Sn 3d XPS spectra of an as prepared  $Ag_3SnS_4$  GDE surface (Panel A) and Ar sputtering of the surface layer for 100s (B) after electrolysis for 1h at 100 mA cm<sup>-2</sup> and 300 mA cm<sup>-2</sup> each. Panel C corresponds to the  $Ag_3SnS_4$  GDE after electrolysis including a sonication step in 20 ml HPLC water for 30s, together with Ar sputtering for 100s before analysis. Panel D corresponds to an  $Ag_3SnS_4$  GDE before and after Ar sputtering, as well as a sonication step for the removal of the top layer before sputtering the electrode surface after electrolysis. The red trace corresponds to the sum of fits, blue to the metal sulfide precursor and the resulting SnO species at 486.4 eV, green to  $SnO_2$  at 487.2 eV and orange to metallic tin at 484.8 eV.



**Figure S13:** S 2p XPS spectra of an as prepared  $Ag_3SnS_4$  GDE surface (Panel A) and Ar sputtering of the surface layer for 100s (B) after electrolysis for 1h at 100 mA cm<sup>-2</sup> and 300 mA cm<sup>-2</sup> each. Panel C corresponds to the  $Ag_3SnS_4$  GDE after electrolysis including a sonication step in 20ml HPLC water for 30s, together with Ar sputtering for 100s before analysis. Panel D corresponds to an  $Ag_3SnS_4$  GDE before and after Ar sputtering, as well as a sonication step for the removal of the top layer before sputtering the electrode surface after electrolysis. The red peak corresponds to the sum of fits, green to the metal sulfide precursor at 161.2 eV, blue to polysulfidic species at 163.6 eV.



**Figure S14**: Ag MNN Auger transitions of an as prepared  $Ag_3SnS_4$  GDE surface (Panel A), after Ar sputtering of the surface layer for 100s (B), after electrolysis for 1h at 100 mA cm<sup>-2</sup> and 300 mA cm<sup>-2</sup> each. Panel C corresponds to the  $Ag_3SnS_4$  GDE after electrolysis including a sonication step in 20 ml HPLC water for 30s, together with Ar sputtering for 100s before analysis. Panel D corresponds to an  $Ag_3SnS_4$  GDE before and after Ar sputtering, as well as a sonication step for the removal of the top layer before sputtering the electrode surface after electrolysis. The red peak corresponds to the sum of fits, green to the  $M_4N_{4,5}N_{4,5}$  transition used for determination of the Auger parameter together with the Ag 3d 5/2 transition.

	Ag 3d 5/2 B.E. / eV	AgM4N45N45 K.E. / eV	Auger parameter
$Ag_3SnS_4$	367.0	357.7	724.7
Ag <sub>3</sub> SnS <sub>4 post electrolysis</sub>	367.2	357.7	724.9
Ag <sub>3</sub> SnS <sub>4 post elec.,</sub>	367.1	358.8	725.9
$Ag_8SnS_6$	367.1	357.6	724.7
$Ag_8SnS_{6 post electrolysis}$	367.5	358.5	725.5
Ag <sub>8</sub> SnS <sub>6 post elec.,</sub> sonicated	367.2	358.9	726.1
Metallic silver	368.8	357.1	725.9
Ag <sub>2</sub> S	368.2	356.4	725.6

**Table S1.** Ag3d and  $AgM_4N_{4,5}N_{4,5}$  peak positions and Auger parameters for  $Ag_3SnS_4$ ,  $Ag_8SnS_6$  electrodes before and after electrolysis.



**Figure S15.** Linear sweep voltammetry of  $Cu_3SnS_4GDE$  recorded in 1 M KOH with a  $CO_2$  supply of 20 ml min<sup>-1</sup>. The average of 3 LSV is displayed as a solid line and the standard error is shown as a shaded area, before chronopotentiometry at -100 mA cm<sup>-2</sup> in teal and after in black. The potentials were recorded using a Gaskatel RHE and are IR corrected based on the measured internal resistance of the electrochemical cell.



**Figure S16.** Linear Sweep Voltammetry of  $Ag_3SnS_4$  GDE recorded in 1 M KOH with a  $CO_2$  supply of 20 ml min<sup>-1</sup>. The average of 3 LSV is displayed as a solid line and the standard error is shown as a shaded area, before chronopotentiometry at -100 mA cm<sup>-2</sup> in teal and after in black using IR-corrected electrode potentials.



**Figure S17.** Linear Sweep Voltammetry of SnS GDE recorded in 1 M KOH with a  $CO_2$  supply of 20 ml min<sup>-1</sup>. The average of 3 LSV is displayed as a solid line and the standard error is shown as a shaded area, before chronopotentiometry at -100 mA cm<sup>-2</sup> in teal and after in black using IR-corrected electrode potentials.



**Figure S18.** Linear Sweep Voltammetry of CuS GDE recorded in 1 M KOH with a  $CO_2$  supply of 20 ml min<sup>-1</sup>. The average of 3 LSV is displayed as a solid line and the standard error is shown as a shaded area, before chronopotentiometry at -100 mA cm<sup>-2</sup> in teal and after in black using IR-corrected electrode potentials.

## **Construction of surface slabs**

To explain the electrocatalytic behavior of Ag and Cu, Sn sulfides, a theoretical approach was undertaken. As copper sulfide undergoes an activation under working conditions, the alloying of Cu and Sn is believed to be the main active species in the conversion of  $CO_2$  to formate. (1) Thus, supercells and consequently slab models of M<sub>3</sub>Sn (M = Ag, Cu), characterized by an orthorhombic Pmmn structure were considered. To describe the enhanced selectivity towards different  $CO_2RR$  products, the CO and HCOO molecules were chosen as adsorbates. Additionally, proton adsorption was also considered due to HER being a competing process. The calculations were performed within the DFT formalism using VASP (*2*, *3*) code and Perdew-Burke-Ernzerhof (PBE) (*4*) potential with the DFT-D2 method to correct for van der Waals interactions. The plane-wave energy cut-off was set to 520 eV and the k-point mesh were generated using Monkhorst-Pack scheme with distances of 0.025 Å.

The 2x2x2 Me<sub>3</sub>Sn supercells were modelled in VESTA software. (5) The structures were relaxed using convergence criteria of  $1 \cdot 10^{-5}$  eV and  $2 \cdot 10^{-2}$  eV/Å<sup>2</sup> for electronic and ionic relaxation, respectively. After the relaxation procedure, the energy of the systems was evaluated again, using stricter electronic convergence criterion of  $1 \cdot 10^{-6}$  eV.

Subsequently, slab models of (100), (110) and (010) planes were created with a vacuum level of 15 Å (Fig. S1, Supplementary Information). To assess the stability of the modelled surfaces, the surface energy  $E_{surf}$  of the slabs was evaluated using eq. (1):

$$E_{surf} = \frac{E_{slab} - nE_{cell,avg}}{2A},\tag{1}$$

where  $E_{slab}$  – total energy of the slab,  $E_{cell,avg}$  – total energy of the supercell divided by the number of atoms in the supercell, n – number of atoms in the slab; A – surface area of the slab, calculated as the product of a and b lattice parameters. Numerical data used in the calculation of surface energies is available in Tables S2-3.

The slabs were again relaxed with two bottom layers of atoms being fixed in the initial positions. The (010) surface was the focus of the investigation due to presence of multiple non-equivalent adsorption sites, as well as lowest surface energy (described in the next section). Placing the adsorbate on the surface resulted in surface coverage of 0.125 ML (1x1). The same numerical parameters and convergence criteria were used as for supercells and pristine slabs, except for systems with OCHO adsorbate for which the ionic convergence criterium was increased to  $5 \cdot 10^{-2} \text{ eV/Å}^2$ . For OCHO, two binding modes were analyzed, namely, \*OCHO and \*O\*OCH. The free-enthalpy change  $\Delta G$  was calculated using eq. (2):

$$\Delta G = E_{ads} - E_{slab} - E_{molecule} - \Delta E_{ZPE} - T\Delta S, \qquad (2)$$

where  $E_{ads}$  and  $E_{slab}$  – total energies of the systems with and without adsorbates, respectively;  $E_{molecule}$  – the calculated energy of either ½H<sub>2</sub>, CO, and HCOO molecules; The entropic term  $T\Delta S$  was approximated using experimental data for H<sub>2</sub>, CO and HCOOH molecules and applying Hess law.  $\Delta E_{ZPE}$  – difference in zero-point energies taken from VASP and experimental data for gas-phase molecules. Numerical data is available in Table S4.

## Surface energy

The surface energies of (010), (100) and (110) Me<sub>3</sub>Sn planes are presented in Figure S19. For both Cu- and Ag-based alloys, the same trends are observable: the (010) plane is the most stable, the (100) plane is the least stable, and the (110) plane exhibits intermediate values of  $E_{surf}$ . However, in all cases, Ag-rich systems exhibit lower plane energies, suggesting greater thermodynamic stability. Incorporating Cu into the structure raises this energy by about 30%, likely due to the mismatch of ionic radii of respective elements. The presence of 4d electrons causes Ag ions to have larger radii in every coordination compared to Cu ions <sup>6</sup>. This also affects the crystallographic cell parameters and volumes. The volume of the Ag-based alloy is 30% higher than that of Cu-based one, which correlates with the higher surface energies. Higher surface energy typically indicates greater surface tension and a higher degree of disorder, leading to increased chemical reactivity. This, in turn, means that for catalytic reactions requiring active sites with higher energy from external sources to drive the reduction reactions. In other words, the  $\Delta G$  of adsorption should be more spontaneous on Cu<sub>3</sub>Sn as opposed to Ag<sub>3</sub>Sn.



Fig S19. Surface energy of different M<sub>3</sub>Sn planes.



Fig S20. The  $Cu_3Sn$  supercell and slab models of (100), (010), (110) planes.

Lattice parameter	Ag₃Sn	Cu₃Sn	
<b>a</b> [Å]	12.123	11.012	
<b>b</b> [Å]	9.541	8.470	
<i>c</i> [Å]	10.626	9.647	
$\alpha = \beta = \gamma [\circ]$	90		
<i>V</i> [Å <sup>3</sup> ]	1229.092	899.742	

**Table S2.** Unit cell parameters of  $2x2x2 M_3$ Sn supercells (M = Ag, Cu).

Model	Number of atoms in the cell	$E_{DFT}$ [eV]	<b>A</b> [Å <sup>2</sup> ]	$E_{surf} [eV/Å^2]$					
Ag₃Sn									
2x2x2 supercell	64	-229.37	-	-					
(010) slab	64	-211.52	128.91	0.138					
(100) slab	32	-98.64	101.38	0.158					
(110) slab	32	-90.49	164.00	0.148					
Cu₃Sn									
2x2x2 supercell	64	-268.95	-	-					
(010) slab	64	-249.29	106.23	0.185					
(100) slab	32	-118.13	81.71	0.200					
(110) slab	32	-108.72	134.02	0.192					

**Table S3.** Total energies, surface areas and surface energies of pristine Me<sub>3</sub>Sn supercells and slabs.

	$r_1$ [Å]	$r_2$ [Å]	$r_{3}$ [Å]	$E_{DFT}$ [eV]	$E_{ZPE}$ [eV]	<i>TS</i> [eV]	Δ <i>G</i> [eV]		
Н									
*Ag	1.65			-214.166	0.101	0.000	0.775		
'Ag'Ag'Ag	1.91	1.91	1.92	-214.922	0.146	0.009	0.055		
'Ag'Ag	1.81	1.81		-214.791	0.126	0.003	0.172		
'Ag'Ag (Sn)	1.90	1.90	2.17	-214.663	0.124	0.016	0.285		
*Sn	1.77			-214.124	0.123	0.061	0.778		
*Ag (110)	1.81	2.31	2.54	-93.2494	0.093	0.010	0.637		
*Cu	1.52			-252.218	0.108	0.000	0.495		
'Cu'Cu (Sn)	1.66	1.67	2.38	-252.271	0.140	0.024	0.451		
'Cu'Cu'Cu	1.74	1.74	1.80	-252.993	0.167	0.006	-0.227		
'Cu'Cu (SnSn)	1.67	1.67		-252.494	0.159	0.017	0.254		
'Cu'Sn	1.60	2.07		-252.285	0.125	0.004	0.441		
*Sn	1.76			-251.841	0.127	0.053	0.838		
*Cu (110)	1.62	2.39	2.38	-111.590	0.078	0.000	0.528		
			CO ar	nd OC					
*Ag (OC)	<u>3.20</u>			-226.371	0.135	0.057	0.402		
*Ag (Sn)	2.11	3.50		-226.727	0.166	0.146	-0.013		
'Ag'Ag'Ag	2.38	2.30	2.30	-226.645	0.161	0.176	0.034		
'Ag'Ag	2.26	2.26		-226.680	0.169	0.178	0.005		
*Cu	1.86			-264.934	0.188	0.104	-0.391		
'Cu'Cu	1.98	1.98		-265.093	0.178	0.160	-0.616		
'Cu'Cu (SnSn)	2.01	2.01		-264.672	0.174	0.157	-0.196		
*Sn	<u>3.41</u>			-264.165	0.143	0.231	0.207		
			OCHO ai	nd OOCH					
'Ag'Ag'Ag	2.35	2.36	2.36	-237.858	0.590	0.253	-2.238		
'Ag'Ag	2.29	2.27		-237.810	0.586	0.270	-2.211		
*Ag*Sn	2.29	2.24		-238.163	0.605	0.227	-2.501		
**Sn	2.34	2.41		-237.857	0.595	0.274	-2.253		
**Cu	2.07	2.71		-275.618	0.582	0.218	-2.255		
*Cu*Cu	2.00	2.01		-276.274	0.609	0.203	-2.869		
*Cu'Cu'Cu	2.05	2.18	2.18	-276.247	0.617	0.196	-2.824		
'Cu'Cu'Cu	2.17	2.16	2.17	-275.872	0.584	0.192	-2.481		
'Cu'Cu (Sn)	2.17	2.17	2.42	-275.422	0.585	0.255	-2.093		
*Cu*Sn	2.02	2.28		-275.949	0.612	0.209	-2.547		

**Table S4.** Binding (interactions) lengths, total energies, zero-point energies, adsorption energies and free-enthalpy change for different adsorption sites on  $Me_3Sn$  slab models.  $r_i$  correspond to the distance between the adsorbate and individual atoms.

	$r_1$ [Å]	$r_2$ [Å]	$r_{3}$ [Å]	$E_{DFT}$ [eV]	$E_{ZPE}$ [eV]	<i>TS</i> [eV]	Δ <i>G</i> [eV]
**Sn	2.51	2.30		-275.363	0.589	0.211	-1.986
			Gas-phase	molecules			
1⁄2 H <sub>2</sub>				-3.386	0.273	0.205	
HCOO				-23.917	0.528	0.555	
СО				-14.776	0.216	0.611	



**Fig S21.** \*H sites on (010) Me<sub>3</sub>Sn planes after the final relaxation process: **a**) \*Ag; **b**) 'Ag'Ag'Ag; **c**) 'Ag'Ag'; **d**) 'Ag'Ag (Sn); **e**) \*Sn; **f**) \*Cu; **g**) 'Cu'Cu (Sn); **h**) 'Cu'Cu'Cu; **i**) 'Cu'Cu (SnSn); **j**) 'Cu'Sn; **k**) \*Sn. The numbers correspond to the atom in the site and  $r_i$  in Table S4.



**Fig S22.** \*CO sites on (010)  $Me_3Sn$  planes after the final relaxation process: **a)** \*Ag (OC); **b)** \*Ag **c)** 'Ag'Ag'Ag'; **d)** 'Ag'Ag; **e)** \*Cu; **f)** \*'Cu'Cu; **g)** 'Cu'Cu (SnSn); **h)** \*Sn. The numbers correspond to the atom in the site and  $r_i$  in Table S4.



**Fig S23.** \*OCHO and \*O\*OCH sites on (010)  $Me_3Sn$  planes after final relaxation process: **a)** 'Ag'Ag'Ag; **b)** 'Ag'Ag (Sn); **c)** \*Ag\*Sn; **d)** \*\*Sn; **e)** \*\*Cu; **f)** \*Cu\*Cu (Sn); **g)** \*Cu'Cu'Cu; **h)** 'Cu'Cu'Cu; **i)** 'Cu'Cu (Sn); **j)** \*Cu\*Sn; **k)** \*\*Sn. The numbers correspond to the atom in the site and  $r_i$  in Table S4.



**Fig S24.** Distribution of the computed Ag residual Bader charge in Ag,  $Ag_3Sn$  supercell, (010), (110) and (100) facets of  $Ag_3Sn$ . Each dot represents an atom in the respective model using the above-described optimized geometries.

Cell type	Catalyst	Electrolyte	Potential vs. RHE (V)	FE (%)	Ј <sub>нсоон</sub> (mA cm <sup>-2</sup> )	Reference
Flow Cell	Ag₃SnS₄	1М КОН	-1.1	54	162	This work
Flow Cell	Cu₃SnS₄	1М КОН	-1.0	57	57	This work
Flow cell	Cu <sub>2</sub> Sn <sub>s</sub>	0.5 М КНСО <sub>3</sub>	-2.2	96	241	(1)
H-type cell	Ag₃Sn / SnO₂	0.5 M NaHCO₃	-0.8	87	25	(6)
H-type cell	Pd₄Ag	0.1 М КНСО <sub>3</sub>	-0.2	100.0	2.1	(7)
H-type cell	N-SnO₂	0.1 М КНСО <sub>3</sub>	-1.2	89.1	15.5	(8)
H-type cell	3 at.% Cu- SnS <sub>2</sub>	0.5 М КНСО <sub>3</sub>	-1.0	90.9	23.8	(9)
Flow cell	S-Bi- NSs	1 М КОН	-0.9	96	192	(10)
Flow- cell	SnS	1 М КОН	-1.0	97.4	88.6	(11)

Table S5 Comparison of CO<sub>2</sub>RR activity in respect to HCOOH production of this work and published catalysts in literature.

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