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Supporting Information

2 Self-Assembled Monolayer Boosts the Air-Stability and

3 Electrochemical Reversibility of O3-Type Layered Oxides for Sodium-

4 Ion Batteries

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11 Experimental Section

12 1. Preparation of NFM and NFM@DHP

13 O3-NaNi_{0.33}Fe_{0.33}Mn_{0.33}O₂ (NFM) powders were synthesized by a traditional solgel route.1, ² The stoichiometric amounts of CH₃COONa (5% excess), 14 Mn(CH₃COO)₂·4H₂O, Ni(CH₃COO)₂·4H₂O, and Fe(NO₃)₃·9H₂O were fully dissolved 15 in deionized water and then mixed with aqueous citric acid under vigorously stirred till 16 a clear solution was obtained. The resulting solution was heated and stirred in a 90 °C 17 oil bath, with excess aqueous solution evaporated off until a clear and viscous gel 18 formed. The as-formed gel was dried at 125 °C overnight to produce a loose and porous 19 precursor. Then, the precursor was ground, calcined in air at 500 °C for 6 h, reground, 20 pressed into pellets, and calcined again at 900 °C for 10 h. Finally, the pellets were 21 ground to obtain the final NFM powders. 22

The NFM@DHP composite was prepared using a straightforward infiltration approach. An appropriate amount of dicetyl phosphate ($C_{32}H_{67}O_4P$) was fully dissolved in 20 mL of tetrahydrofuran (THF) to form a 0.01 wt% solution. Then, 0.2 g of NFM powders were immersed in the above solution. After vigorous stirring for 2 h at room temperature, the dispersion was centrifuged, followed by thorough washing with THF to remove residual DHP molecules. The precipitates were then vacuum-dried overnight at 60 °C to give the target NFM@DHP composite.

To assess the air stability of NFM and NFM@DHP, the samples were exposed to an environment with a relative humidity (RH) of 45-50% for various durations. This was achieved by placing the materials on a watch glass in a sealed container with a saturated solution of K_2CO_3 . The container was placed in a controlled environment at $30^{\circ}C$, with the RH regulated at 45-50% using a hygrometer. According to the experimental requirements, the materials were placed for certain days to obtain the target materials.

37 2. Characterizations.

The crystalline structures of the samples were identified using X-ray diffraction (XRD, Bruker D8 ADVANCE) with Cu Kα radiation. The GSAS-II software was used to obtain the Rietveld refinement data. The morphologies of the samples were observed by scanning electron microscopy (SEM, TESCAN MIRA3) equipped with energydispersive X-ray spectroscopy (EDS). Transmission electron microscopy (TEM, Talos f200i) was manipulated to acquire images of the refined structures of the samples. 44 Fourier transform infrared spectroscopy (FT-IR, BRUKER TENSOR II (FTS6000)) 45 was recorded using KBr pellets in a unit of cm⁻¹. The chemical compositions were 46 detected on an X-ray photoelectron spectrometer (XPS, Thermo Fischer, ESCALAB 47 Xi+) under a vacuum of 8×10^{-10} Pa (the C 1s peak at 284.8 eV as reference). Contact 48 angle analysis was recorded on a SINDIN SDC-200S instrument at room temperature.

49 3. Electrochemical measurements.

The electrochemical performance of the samples was evaluated using CR2032 50 coin-type cells assembled in an argon-filled glove box. The cathodes were prepared by 51 mixing the active materials with super-P carbon black and polyvinylidene fluoride 52 (PVDF) binder with a mass ratio of 8:1:1 in N-methyl pyrrolidone (NMP) to obtain a 53 stable homogeneous suspension, which was then coated on aluminum foil. The mass 54 loading of the active material on the electrode was 1-1.5 mg cm⁻¹. Prior to slurry 55 preparation, all exposed samples were dried under vacuum conditions at 80 °C. Sodium 56 metal foil was used as then anode and glass fiber (GF/D, Whatman, Φ 19 mm) was used 57 as the separator. The electrolyte was 1 M NaClO₄ in a mixture of propylene 58 carbonate/ethylene carbonate (PC/EC = 1:1 in volume) with fluoroethylene carbonate 59 (5% volume), with about 100 µL of electrolyte added into each coin-type cell using a 60 pipette. All batteries were rested overnight before testing. The cells were initially 61 activated for 3 cycles at 0.1 C before undergoing long-cycle testing. The cycled 62 cathodes (NFM and NFM@DHP) for XRD measurement were rinsed three times with 63 propene carbonate and subsequently blow-dried under argon gas. Electrochemical 64 impedance spectra (EIS) were acquired by an Admiral Instruments electrochemical 65 workstation. Charge and discharge measurements were carried out using multichannel 66 battery testing systems (CT2001A Wuhan LAND Electronics Co., Ltd., China) at room 67 68 temperature.



70 Figure S1 EDS mappings for Na, Mn, Fe, Ni, and P elements of a) NFM and b) NFM@DHP.







Figure S3 SEM image of NFM-7D.





Figure S4 FTIR spectra of NFM-7D and NFM@DHP-7D.



82 Figure S5 C 1 s XPS spectra of a) NFM and NFM-7D, and b) NFM@DHP and NFM@DHP-7D.



85 Figure S6 Nyquist plots of NFM and NFM@DHP before and after being exposed for 7 days. The

- 86 increased interfacial resistance of NFM@DHP should be ascribed to the presence of the DHP
- 87 coating on the electrode surface.
- 88



91 Figure S7 Charge/discharge curves of a) NFM and b) NFM@DHP at the 4th, 150th, and 330th

92 cycles.



Figure S8 a) XRD spectrum, b) the first charge/discharge curve at 0.1C, and c) cycling stability at
1C of NFM@DHP after 21-days of air exposure.



99 Figure S9 a) XRD spectrum, b) the first charge/discharge curve at 0.1C, and c) cycling stability at

100 1C of NFM@DHP after 28-days of air exposure.



103 Figure S10 Photographs of the slurries of a, b) NFM and c, d) NFM@DHP under a, c) humid and

- 104 b,d) dry air.
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- 106







Figure	S12 SEM image	of NFM after	200 cycles at 1C
riguit	or other intege		200 cycles at 10.

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