# Novel phosphorus-doped lead oxide electrode for

# oxygen evolution reaction

# **Supporting Information**

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#### Part I. Experimental details

The electrodeposition of P-PbO<sub>2</sub> electrode was achieved through an anodic oxidation technique. The preliminary composition of the deposition solution was 35 g L<sup>-1</sup> Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> + 16 g L<sup>-1</sup> Pb(NO<sub>3</sub>)<sub>2</sub>. The electrodeposition were carried out in a two-electrode system under constant current, where a 1 cm<sup>2</sup> pure lead substrate and a 4 cm<sup>2</sup> platinum plate was used as the anode and cathode, respectively. The deposition temperature and agitation rate were controlled by the DF-101S constant temperature magnetic agitator. Parameters for optimizing the quality and structure of P-PbO<sub>2</sub> deposit were listed in Table S1 in the Supporting Information.

The morphology of  $P-PbO_2$  deposit was observed using JSM-6360 SEM. GENESIS60S EDS was further used for the chemical composition analysis. The mass fraction (wt.%) of phosphorus in the deposit was approximately calculated with equation (1):

$$wt.\%(P) = (1 + \frac{M_{Pb} + 2M_O}{M_P} \times \frac{N_{Pb}}{N_p})^{-1} \times 100\%, \qquad (1)$$

where  $M_{Pb}$ ,  $M_P$  and  $M_O$  represent the relative atomic mass of lead, phosphorus and oxygen, respectively, and  $N_{Pb}/N_P$  is the atomic ratio of lead and phosphorus detected by EDS. TTR-III X-ray Diffratometer (XRD) with Cu Ka (k = 1.54056 Å) as radiation source was used to identify the phase composition and crystal structure. Analysis of XRD spectra were carried out in the MDI Jade 6.0 software.

Characterizations of electrocatalytic activity of the Pb/P-PbO<sub>2</sub> electrode for OER were performed on the PARSTAT 2273 Advanced Electrochemical System. The standard threeelectrode cell was used. The Pb/P-PbO<sub>2</sub> electrodes were used as anodes. A platinum plate of 4  $cm^2$  and the KCl saturated calomel electrode (SCE) were used as counter electrode and reference electrode, respectively. All potentials showed in the figures are against SCE. 160 g L<sup>-1</sup> H<sub>2</sub>SO<sub>4</sub> solution was used as the electrolyte. The characterization process include two steps: (1) preconditioning by galvanostatic polarization at 50 mA cm<sup>-2</sup> for 90 min, (2) electrochemical impedance spectra (EIS) were immediately collected at various anodic potentials. The frequency range was from 100 kHz to 10 mHz, the AC Amplitude was 5 mV. The solution temperature was controlled at 35 °C throughout the characterization. In addition, a plate of pure lead anode (1 cm<sup>2</sup>, Yuguang Gold & Lead, Henan, China) was galvanostatically polarized in 160 g L<sup>-1</sup> H<sub>2</sub>SO<sub>4</sub> solution for 72 h to form a Pb/PbO<sub>2</sub> anode. The same electrochemical measurements were carried out on this anode as comparison.

#### Part II. Deposition principle of P-PbO<sub>2</sub>

In order to study the formation mechanism of P-PbO<sub>2</sub> at the initial stage of the electrodeposition, both platinum and pure lead were used as the substrates. The cyclic voltammetry was carried out in Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> solution as well as the Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> + Pb(NO<sub>3</sub>)<sub>2</sub> solution. The scanning rate was 10 mV s<sup>-1</sup>. The potential cycle started from -1.0 to 2.5 V and then reversed to -1.0 V.

As showed in Fig. S1A, the voltammogram on Pt substrate in  $Na_4P_2O_7$  solution only shows an oxygen evolution branch (d) within 1.0~2.5 V, indicating  $Na_4P_2O_7$  solution is chemically stable in the potential region of -1.0 to 2.5 V. When the Pt substrate was cyclically polarized in  $Na_4P_2O_7 + Pb(NO_3)_2$  solution, the voltammogram presents three anodic peaks (a), (b), (c) and an oxygen evolution branch (d) in the positive scanning and two cathodic peaks (e) and (f) in the negative scanning. The appearance of (a) and (b) were considered due to the formation of lead phosphoric salt related intermediate passivation layer. The appearance of peak (c) and (e) represents the formation and reduction of PbO<sub>2</sub>, respectively.

In Fig. S1B, peak (a) and (b) on Pb substrate in  $Na_4P_2O_7$  solution are obviously enlarged, and thus one can predict that lead substrate reacted with the pyrophosphate to form the passivation layer. Also, the PbO<sub>2</sub> formation peak (c) and O<sub>2</sub> evolution branch (d) can be considered to combine to one branch. The onset potential (1.6 V) of the combined branch is higher than that of Pt substrate (1.0 V) since the over-potential for oxygen evolution on Pt is much lower than that on Pb. The appearance of anodic peak (a') in the negative scanning further indicates the partially oxidation of lead substrate. With the presence of  $Pb(NO_3)_2$  in the deposition solution, the voltammogram was compressed. One can predict that the lead substrate is prone to form a more compact passivation layer due to the appearance of anodic reaction (b'). This was also confirm by the compression of peak (a') since the contact of deposition solution with the lead substrate become more difficult due to the passivation layer. The layer formed at this case shows lower oxygen evolution activity, rendering the onset potential increase to the positive direction.

To confirm the intermediate product corresponding to anodic peak (a) and (b), constant potential polarization was carried out on the lead substrate  $(1 \text{ cm}^2)$  at -0.5 V and -0.1 V in a solution containing 35 g L<sup>-1</sup> Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> + 16 g L<sup>-1</sup> Pb(NO<sub>3</sub>)<sub>2</sub>. SEM and XRD were used to characterize the corresponding product, as shown in Fig. S2. The intermediate products present similar morphology while overall the later are much more flat and compact. XRD analysis indicates the product at -0.5 V corresponds to Pb<sub>2</sub>P<sub>2</sub>O<sub>7</sub> and that obtained at -0.1 V corresponds to Pb<sub>2</sub>P<sub>2</sub>O<sub>7</sub> +

 $PbP_2O_7$ . It is worthy to mention that the peak (a) and (b) appeared on Pt substrate is due to the deposited lead in the lower potential region corresponding to peak (f). Thus, we propose that at the initial stage of P-PbO<sub>2</sub> deposition, the electrochemical reactions can be expressed as follows:

Peak (a):  $2Pb + P_2O_7^{4-} - 4e^- \rightarrow Pb_2P_2O_7$ 

Peak (b):  $Pb_2P_2O_7 + P_2O_7^{4-} - 4e^- \rightarrow 2PbP_2O_7$ 

Peak (b'):  $Pb^{2+} + P_2O_7^{4-} - 2e^- \rightarrow PbP_2O_7$ 

Peak (c):  $Pb_2P_2O_7 + 8OH^- - 4e^- \rightarrow 2PbO_2 + P_2O_7^{4-} + 4H_2O_7^{4-}$ 



**Fig. S1.** (A, B) CV of (A) Pt and (B) Pb substrates in solution containing 35 g  $L^{-1}$  Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> and 35 g  $L^{-1}$  Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> + 16 g  $L^{-1}$  Pb(NO<sub>3</sub>)<sub>2</sub>, respectively. (C) Influence of additives on CV of lead substrate in the 35 g  $L^{-1}$  Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> + 16 g  $L^{-1}$  Pb(NO<sub>3</sub>)<sub>2</sub> solution.

Despite electrodeposition of PbO<sub>2</sub> in the Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> + Pb(NO<sub>3</sub>)<sub>2</sub> solution is theoretically possible according to above proposed principle, it is difficult to obtain a high quality deposit with controlled composition in practice: (1) the deposition potential is high (1.5 V~ 2.5 V), as a result the formation of PbO<sub>2</sub> is largely retarded by the evolution of O<sub>2</sub>; (2) the basic Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> +

Pb(NO<sub>3</sub>)<sub>2</sub> solution will become turbid during long time deposition, rending the deposit to be very rough or present a granular shape. Accordingly, several additives were added into the solution to optimize the deposition conditions. Fig. S3 shows the influence of  $Cu^{2+}$  (2 g L<sup>-1</sup>  $Cu(NO_3)_2$ ), F<sup>-</sup> (1 g L<sup>-1</sup> NaF) and peptone (0.2 g L<sup>-1</sup>) on the cyclic voltammograms on lead substrate. With the addition of  $Cu^{2+}$  in the deposition solution, the onset potential of branch (d) shifted to the negative direction and the peak height was obviously increased.  $Cu^{2+}$  partially deposited on the substrate through the replacement reaction, which will inhibit the formation of passivation layer and render the oxygen evolution and PbO<sub>2</sub> deposition become easier. F<sup>-</sup>, as an inhibitor for oxygen evolution [S 1] rendered oxygen evolution potential shift to positive direction, which is prone to the formation of PbO<sub>2</sub>. Peptone can favorably dissolved in the deposition solution and increase its wettability. As a result, the dissolution of evolved O<sub>2</sub> become easier and the branch (d) is heightened. It is worthy to point out that with the addition of these additives, the stability of deposition was largely improved. Experimentally we observed that even during constant current deposition of 120 min, the solution kept clear throughout and no precipitate formed.



Fig. S2. (A, B) SEM images showing the morphology of intermediate products obtained at -0.5

V (A) and -0.1 V (B). (C, D) XRD patterns corresponding to A and B, respectively.

#### Part III. Parameter study for the electrodeposition of P-PbO<sub>2</sub>

As shown in Table 1, the influence of temperature, additives, current density, agitation and deposition duration on the surface morphology and coating structure was studied in detail. The basic deposition solution was  $35 \text{ g L}^{-1} \text{ Na}_4 \text{P}_2 \text{O}_7 + 16 \text{ g L}^{-1} \text{ Pb}(\text{NO}_3)_2$  throughout the study.

	Temperature	Additives (g $L^{-1}$ )			Current	Agitation	Time
(°C)		$Cu(NO_3)_2$	NaF	Peptone	$(mA cm^2)$	(rpm)	(min)
1	30	2	0	0	10	400	60
2	40	2	0	0	10	400	60
3	50	2	0	0	10	400	60
4	60	2	0	0	10	400	60
5	70	2	0	0	10	400	60
6	70	2	0	0.2	10	400	60
7	70	2	0.5	0	10	400	60
8	70	2	0.5	0.2	10	400	60
9	70	2	1	0.2	10	400	60
10	70	2	1	0.4	10	400	60
10	70	2	1	0.4	10	400	60
11	70	2	1	0.4	15	400	60
12	70	2	1	0.4	20	400	60
13	70	2	1	0.4	10	300	60
10	70	2	1	0.4	10	400	60
14	70	2	1	0.4	10	500	60
15	70	2	1	0.4	10	500	60
16	70	2	1	0.4	10	500	90

Table S1. Experimental parameters used for the electrodeposition of P-PbO<sub>2</sub>.

# **2.1. Influence of temperature**

Fig. S3 shows the SEM images of P-PbO<sub>2</sub> obtained at different temperature by applying constant current density of 10 mA cm<sup>-2</sup> and agitation rate of 400 rpm. The deposition time was 60 min. At low temperature, the deposit surface was relatively rough, presenting a porous structure with

cauliflower structure. With the increase of temperature, the morphologies become flat and dense, uniform crystal grains with clear crystal edge were observed. This phenomenon can be explained as follows. On the one hand, the ionic migration became easier and the electrochemical activity of electrode was improved with the increase of temperature. As a result, the driving force of PbO<sub>2</sub> nucleation and growth was increased. On the other hand, the formation of PbO<sub>2</sub> is an exothermic process [S2]. With temperature increased, the equilibrium of the reaction shifts negatively, rendering the nucleation process much more difficult. Thus, high temperature presents a function of grain refinement, which makes the surface of PbO<sub>2</sub> more flat and dense.



**Fig. S3.** Influence of temperature on the morphology (×100) of P-PbO<sub>2</sub> deposited on lead substrate: (A) 30 °C, (B) 40 °C, (C) 50 °C, (D) 60 °C.

# 2.2. Influence of additives

The influence of  $F^-$  and peptone on the surface morphology of P-PbO<sub>2</sub> was shown in Fig. S4. For this group of experiments, the temperature is 70 °C, the current density is 10 mA cm<sup>-2</sup>, the agitation rate is 400 rpm and the deposition time is 60 min. Generally, the addition of peptone makes the deposits more uniform and denser while  $F^-$  makes them more flat [S 3]. The collaborative addition of peptone and  $F^-$  will further improve the quality of P-PbO<sub>2</sub> deposit. As shown in Fig. S4E-F, the deposit surface becomes smooth, compact and highly ordered. It is

worthy to mention that peptone is an effective surfactant in this deposition system [S4]. On the one hand, the solution became chemically more stable with the addition of peptone. There are neither any precipitates formed nor any color-change during the electrodeposition for even more than 90 min. On the other hand, peptone can be absorbed on the electrode surface and inhibits the electron transfer process, which will refine the  $PbO_2$  grains and render the surface of deposit smooth and dense.



**Fig. S4.** SEM (×1000) of P-PbO<sub>2</sub> prepared on Pb substrate with addition of (A) 0 NaF and 0 peptone, (B) 0 NaF and 0.2 g L<sup>-1</sup> peptone, (C) 0.5 g L<sup>-1</sup> NaF and 0 peptone, (D) 0.5 g L<sup>-1</sup> NaF and 0.2 g L<sup>-1</sup> peptone, (E) 1 g L<sup>-1</sup> NaF and 0.2 g L<sup>-1</sup> peptone, (F) 1 g L<sup>-1</sup> NaF and 0.4 g L<sup>-1</sup> peptone to the basic deposition solution.

# 2.3. Influence of current density and agitation

The influence of current density the morphology of P-PbO<sub>2</sub> coatings was shown in Fig. S5A-B and Fig. S4E. With the increase of current density, the deposit surface becomes rough. Threadlike cracks along the grain boundaries can also be observed. The electrodeposition of PbO<sub>2</sub> has been reported to be very sensitive to current density [S5]. On the one hand, the formation of PbO<sub>2</sub> is a process coupled with the evolution of O<sub>2</sub>. At low current density, most of the charges are used in the formation of PbO<sub>2</sub> and thus the crystal grains grow up uniformly. As current density increase, the charges used for oxygen evolution also increased, rendering the deposit a relative rough surface. On the other hand, PbO<sub>2</sub> electrodeposition is a process

controlled by the migration of ions. The previously adsorbed oxygen-containing species are very important for the nucleation and growth of PbO<sub>2</sub> [S6]. At low current density, mass transport process takes a major part compared with the electron transfer and the deposit structure can be uniformly formed. At higher current densities, concentration polarization becomes obvious and Pb-containing ions can't migrate to the electrode surface timely as needed. Hence, the deposit formed will be less ordered and the outward growth of PbO<sub>2</sub> is aggravated. Based on this principle, the deposition process should also have a large dependence on the agitation rate. As shown in Fig. S5C-D and Fig. S4E (400 rpm), the P-PbO<sub>2</sub> deposit presented a rough surface with visible threadlike cracks at agitation of 300 rpm. As the agitation increase to 400 and to 500 rpm, the deposits became smooth and orderly.



**Fig. S5.** SEM ( $\times$ 1000) of P-PbO<sub>2</sub> obtained at current of (A) 15 mA cm<sup>-2</sup>, (B) 20 mA cm<sup>-2</sup>, and obtained at agitation rate of (C) 300 rpm, (D) 500 rpm.

# 2.4. Influence of deposition time

It is well-defined that the electrodeposition processes present an electrochemical leveling function when organic surfactants are used [S7]. In this paper, when the deposition time was increased to 90 min, a highly smooth, uniform and reflective deposit was obtained, as shown in Fig. S6.



Fig. S6. SEM of P-PbO<sub>2</sub> deposit obtained after electrodeposition for 90 min: (A)  $\times 1000$ , (B)  $\times 100$ .

# Part IV. Pb/PbO<sub>2</sub> electrode as comparison

The control sample Pb/PbO<sub>2</sub> electrode was obtained by galvanostatically polarizing a 1 cm<sup>2</sup> pure lead plate at 50 mA cm<sup>-2</sup> for 72 h in 160 g  $L^{-1}$  H<sub>2</sub>SO<sub>4</sub> solution. The surface morphology was shown in Fig. S7A-B. Chemical and phase composition were shown in Fig. S7C-D.



Fig. S7. (A-B) SEM images showing the surface morphology of Pb/PbO<sub>2</sub> electrode: (A) ×1000,
(B) ×100. (C) EDS spectrum, (D) XRD pattern for the Pb/PbO<sub>2</sub> electrode.

### References

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